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December 14, 1983 L-83-583

Office of Nuclear Reactor Regulation Attention: Mr. Darrell G. Eisenhut, Director Division of Licensing U. S. Nuclear Regulatory Commission Washington, D.C. 20555

Dear Mr. Eisenhut:

Re: St. Lucie Unit I Docket Nos. 50-335 Reactor Vessel Surveillance Specimen Capsule W-97

Our letter L-83-316, dated May 20, 1983, as supplemented by L-83-538, dated October 31, 1983, committed to providing a final report summarizing the test results obtained from the first reactor vessel material irradiation surveillance specimen.

Please find attached the evaluation of irradiated capsule W-97.

Very truly yours,

Earon For

J. W. Williams, Jr. Vice President Nuclear Energy

JWW/JEM/DAC/cab

Attachment

8312200339 831

cc: Mr. James P. O'Reilly, Region II Harold F. Reis, Esquire PNS-LI-83-732

FLORIDA POWER & LIGHT COMPANY ST. LUCIE UNIT NO. 1

POST-IRRADIATION EVALUATION OF REACTOR VESSEL SURVEILLANCE V CAPSULE W-97 V

December 1983

Prepared by:

S.T. Byme S. T. Byrne, Cognizant Engineer

Date: 11-16-83

Approve by:

S. M. Schloss, Supervisor Primary Boundary Materials

Approved by:

J. J. Koziol, Man Systems Materials Manager

Date: // -17-83

Date: 11.17.83

Date: u/17/83

Approved by:

831,2200339

WR Moran For T. P. Gates T. P. Gates, St. Lucie Unit #1 Project Manager

Combustion Engineering, Inc. Nuclear Power Systems Windsor, Connecticut

RECILATORY DECKET FILE COPY

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TABLE OF CONTENTS

Section	Title	Page No.
I	Summary	1
II	Introduction	3
III	Surveillance Program Description	4
IV	Capsule Withdrawal and Disassembly	16
٧	Test Results	18
VI	Data Analysis	77
VII	References	. 87
Appendix A	Tension Tests - Description and Equipment	A-1
Appendix B	Charpy Impact Tests - Description and Equipment	B-1
Appendix C	Instrumented Charpy V-Notch Data Analysis	C-1



ii

List of Tables

•

٠,

Table No.	Title	Page No.
III-1	Reactor Vessel Beltline Plates	5
III-2	Reactor Vessel Beltline Welds	6
III-3	Reactor Vessel Beltline Plates Chemical Analysis	. 7
III-4	Surveillance Plate and Weld Metal Chemical Analysis	8
III-5	St. Lucie Unit 1 Reactor Vessel Surveillance Capsule Removal Schedule .	14
III-6	Type and Quantity of Specimens in W-97 Capsule	15
IV-1	Mechanical Test Specimens Removed from W-97 Capsule	17 -
V-1 **	Composition and Melting Points of Temperature	19
	Monitor Materials	
V-2	Neutron Flux Monitors	·23
V-3	St. Lucie Unit 1 Flux Spectrum Monitor Activities	26
V-4	Axial Variation in Surveillance Capsule Flux	36
V-5	Flux Monitor Saturated Activities	37
V-6	Average Fast Neutron Flux	40
V-7	Fast Neutron Fluence	40
V-8	Irradiated Plate and Weld Chemical Analysis	46
V-9	Post-Irradiation Tension Test Properties	49
V-10	Pre-Irradiation Tension Test Properties .	50
V-11	Charpy Impact Results, Base Métal (WR)	57 [°]
V-12	Charpy Impact Results, Base Metal (RW)	58
V-13	Charpy Impact Results, Weld Metal	59
V-14	Charpy Impact Results, HAZ	60



(E

....

iii

List of Tables (cont'd)

Table No.	Title	Page No.
VI-1	Summary of Toughness Property Changes	80
VI-2	Projected NDTT Shift and Adjusted RTNDT	84
' 4	for Surveillance Material	
VI-3	Proposed New Capsule Removal Schedule	86
C-1	Instrumented Charpy Test, Base Metal (WR)	C-3
C-2	Instrumented Charpy Test, Base Metal (RW)	C-4
C-3	Instrumented Charpy Test, Weld Metal	C-5
C-4	Instrumented Charpy Test, HAZ	C-6
C-5	Toughness Property Changes Based on Instrumented Charpy Impact Test	C-7 -



iv



•

۲ ۲ ۲ 'n

r

,

List of Figures

. .

Figure No.	IITIE	Page No.
III-l	Surveillance Capsule Assembly	10
III-2	Charpy Impact Compartment Assembly	* 11
III-3	Tensile-Monitor Compartment Assembly	12
III-4	Location of Surveillance Capsule Assemblies	13
V-1	Temperature Monitors, Compartment 7214	20
V-2	Temperature Monitors, Compartment 7241	20
V-3.	Temperature Monitors, Compartment 7273	21
V-4	Geometry Used in DOT Model	32
V-5	Power Distributions Used in DOT, Cycle 1-5	33 _
V-6	Power Distributions Used in DOT, Cycle 6	34
V-7	DOT Model of W-97 Surveillance Capsule	35
V-8	Azimuthal Flux Distribution, Cycles 1-5	38
V-9	Azimuthal Flux Distribution, Cycle 6	39
V-10	Axial Power Distribution, Cycles 1-5	41
V-11	Axial Power Distribution, Cycle 6	42
V-12	Axial Flux Distribution, Cycles 1-5	43
V-13	Axial Flux Distribution, Cycle 6	44
V-14	Location of Weld Metal Chemical	47
-	Analysis Specimens	
V-15	Stress-Strain Record, Base Metal, 72F	51
V-16	Stress-Strain Record, Base Metal, 250F	51
V-17	Stress-Strain Record, Base Metal, 550F	52
V-18	Stress-Strain Record, Weld Metal, 72F	5 2
V-19	Stress-Strain Record, Weld Metal, 250F	53
V-20	Stress-Strain Record, Weld Metal, 550F	53
V-21	Stress-Strain Record, HAZ Metal, 72F	54 ·
V-22	Stress-Strain Record, HAZ Metal, 250F	54
V-23	Stress-Strain Record, HAZ Metal, 550F	55
V-24	Irradiated Tension Specimens	56

۷

. ٢

)

List of Figures (cont'd)

Figure No.	Title	Page No.
V-25	Charpý Impact Energy, Base Metal (WR)	61
V-26	Charpy Lateral Expansion, Base Metal (WR)	62
V-27	Charpy Shear Fracture, Base Metal (WR)	63
V-28	Charpy Impact Energy, Base Metal (RW)	64
V-29	Charpy Lateral Expansion, Base Metal (RW)	65
V-30	Charpy Shear Fracture, Base Metal (RW)	66
V-31	Charpy Impact Energy, Weld Metal	67
V-32	Charpy Lateral Expansion, Weld Metal	68
V-33	Charpy Shear Fracture, Weld/Metal	69
V-34	Charpy Impact Energy, HAZ	70 ⁻
V-35	Charpy Lateral Expansion, HAZ	71
V-36	Charpy Shear Fracture, HAZ	72
V-37	Fracture Surfaces, Impact Specimens, Base Metal (WR)	73
V-38	Fracture Surfaces, Impact Specimens, Base Metal (RW)	74·
V-39	Fracture Surfaces, Impact Specimens, Weld Metal	75 .
V-40	Fracture Surfaces, Impact Specimens, HAZ	76
VI-1	Predicted NDTT Shift for the St. Lucie Unit 1 Reactor Vessel Surveillance Materials	81





,

)

List of Figures (cont'd)

Figure No.

٤.

<u>Title</u>

Page No.

	•	
A-1	Tension Test System	A-2
A-2	Typical Tension Specimen	A-3
A-3	Location of Tension Specimens in Base Metal	A-4
A-4	Location of Tension Specimens in Weld Metal	A-5
A-5	Location of Tension Specimens in HAZ	A-6
8-1	Charpy Impact Test System	8-4
8-2	Typical Charpy V-Notch Impact Specimen	B-5
B-3	Location of Charpy Specimens in Base Metal	B-6
B-4	Location of Charpy Specimens in Weld Metal	B-7_
B-5	Location of Charpy Specimens in HAZ	B-8
Ç-1	ICV Load vs. Temperature Diagram, Base Metal (WR)	C-8
C-2	ICV Load vs. Temperature Diagram, Base Metal (RW)	C-9
C-3	ICV Load vs. Temperature Diagram, Weld Metal	C-10
C-4	ICV Load vs. Temperature Diagram, HAZ	C-11



SUMMARY

I.

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The first surveillance wall capsule (W-97) was removed from the St. Lucie Unit 1 reactor vessel in May 1983 after 4.67 effective full power years of reactor operation. The surveillance test specimens and monitors were evaluated at C-E's Windsor, Connecticut laboratory facility.

Post-irradiation evaluation of the temperature monitors indicated that the irradiation temperature was between $536^{\circ}F$ and $558^{\circ}F$. Analysis of the neutron threshold detectors provided a capsule fluence of 5.5 x 10^{18} n/cm² (E>1 MeV), which corresponded to a maximum fluence at the inside surface of the reactor vessel of 3.9 x 10^{18} n/cm².

Radiation induced changes in the uniaxial tension and impact properties were determined for the base metal, weld metal and heat-affected. zone surveillance materials. Transition temperature shifts ranged from $68^{\circ}F$ to $74^{\circ}F$ for the base and weld metal, respectively. The upper shelf impact energy after irradiation was in excess of 75 ft-lb for each of the surveillance materials, ranging from 77.5 ft-lb for the base metal (transverse) to 106.5 ft-lb for the base metal (longitudinal).

Chemical analysis of the irradiated base and weld metal Charpy specimens was performed by X-ray fluorescence. The analysis confirmed that the chemistry of the irradiated materials was consistent with the chemistry originally reported for the surveillance materials.

The base metal, weld metal and HAZ exhibited similar changes in tensile properties after irradiation. The yield strength and ultimate strength increased approximately 14%, while total elongation and reduction in area decreased 2% to 3% (on average) after irradiation.

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The NDTT shift prediction method from the St. Lucie Unit 1 Technical Specifications was found to be conservative by a factor of 90% for the base metal and 85% for the weld metal based on the W-97 surveillance capsule measurements. In contrast, shifts predicted using Regulatory Guide 1.99 were 115% higher than measured for the weld metal and 17% higher than measured for the base metal. The radiation resistance of the weld is attributed to the low nickel content (.11%). Based upon experimental data, the weld metal shift will continue to be similar to that of the plate, despite the difference in copper content (0.23 w/o for the weld and 0.15 w/o for the plate). Therefore, a more accurate shift prediction method was developed using the surveillance measurements and the slope of the Regulatory Guide 1.99 shift correlation. The predicted end-of-life (32EFPY) RTNDT shift for the surveillance plate and weld at the inside surface of the reactor vessel is 187°F using the revised prediction method and the projected EOL fluence of 3.5×10^{19} n/cm² (based on operation without the thermal shield subsequent to Cycle 5).

The predicted decrease in upper shelf energy at end-of-life based on the method given in Regulatory Guide 1.99 is 41% for the weld and 37% for the plate at the one-quarter thickness location in the vessel. Using this conservative prediction, the upper shelf energy of the plates will remain above 70 ft-lb during the design life of the vessel, and the weld shelf energy will remain above 85 ft-lb. These projected values are well in excess of the 50 ft-lb value currently considered to be a reasonable lower limit for continued safe operation.

The projected changes in upper shelf energy and NDTT shift are based on the post-irradiation test measurements from the W-97 encapsulated materials, and projected EOL fluence values for operation without the thermal shield following Cycle 5 using planned power distributions for Cycle 6 to extrapolate beyond 4.67 EFPY. These surveillance results are representative of the six beltline plates, the closing girth seam weld (9-203), and the intermediate shell course longitudi-



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nal seam welds (2-203). The lower shell course longitudinal seam welds (3-203) and the upper to intermediate shell girth weld (8-203), because of their higher nickel content, are expected to be more sensitive to radiation than the surveillance weld, so an alternate shift prediction methodology should be used. (Further information concerning shift prediction methodology for these welds is addressed in a separate report⁽¹¹⁾.)

Recommended changes to the surveillance capsule withdrawal schedule were provided to meet the requirements of 10CFR50, Appendix H. If the proposed schedule is implemented, the next capsule will be withdrawn after 7 to 9 effective full power years of operation.

II. Introduction

The purpose of the St. Lucie Unit 1 surveillance program is to monitor the radiation induced changes in the mechanical properties of ferritic materials in the reactor vessel beltline during the operating lifetime of the reactor vessel. The surveillance program includes the determination of the pre-irradiation (baseline) strength and toughness properties and periodic determinations of the property changes following neutron irradiation. These property changes are used to verify and update the operating limits (heat-up and cool down pressure/temperature limit curves) for the primary system.

The St. Lucie Unit 1 Surveillance $Program^{(1)}$ is based upon ASTM E185-70, "Recommended Practice for Surveillance Tests for Nuclear Reactor Vessels". The pre-irradiation (baseline) evaluation results from the St. Lucie Unit 1 reactor vessel surveillance materials are described in C-E report TR-F-MCM-005.⁽²⁾ This report describes the results obtained from evaluation of irradiated materials from capsule W-97 which was removed from the reactor in May 1983.

III. Surveillance Program Description

The St. Lucie Unit 1 reactor pressure vessel was designed and fabricated by Combustion Engineering, Inc. The reactor vessel beltline, as defined by 10CFR50, Appendix H, consists of the six plates used to form the lower and intermediate shell courses in the vessel, the included longitudinal seam welds and the lower to intermediate shell girth seam weld. The plates were manufactured from SA533 Grade B Class 1 quenched and tempered plate. The heat treatment consisted of austenization at 1600 + 50F for four hours, water quenching and tempering at 1225 + 25F for four hours. The ASME Code qualification test plates were stress relieved at 1150 + 25F for forty hours, and furnace cooled to 600F. The longitudinal and girth seam welds were fabricated using E8018-C3 manual arc electrodes and Mil B-4 submerged arc weld wire with Linde 124, 0091 and 1092 flux. The post weld heat treatment consisted of a forty hour* 1150 + 25F stress relief heat treatment followed by furnace cooling to 600F. The beltline materials are identified in Tables III-1 and III-2. The chemical analyses of the six beltline plates are given in Table III-3. The materials included in the surveillance program were obtained from the actual reactor vessel beltline materials. The base metal surveillance material, a section from plate C-8-2, was selected from the six beltline plates on the basis of the highest initial Charpy 30 ft-1b index temperature. The heat treatment of the surveillance plate duplicated that of the reactor vessel ASME Code.qualification test plates. The surveillance weld material was fabricated by welding together sections of plates C-8-1 and C-8-3 using the same weld procedure and wire-flux combination used for the intermediate to lower shell girth seam weld (9-203). Mil B-4 submerged arc filler wire and Linde 0091 flux were used. The postweld heat treatment consisted of a forty hour stress relief at 1125 ± 25F followed by furnace cooling to 600F. The surveillance heat-affected zone material was fabricated by welding together sections of plates C-8-2 and C-8-3 in the same manner as the surveillance weld material with the same post weld heat treatment. The chemical analyses of the surveillance plate and weld (2) is given in Table III-4.

*Eighteen hours for the closing girth seam weld.

TABLE III-1 REACTOR VESSEL BELTLINE PLATES

Location	<u>Piece Number</u>	<u>Code Number</u>	<u>Heat Number</u>	Supplier
Intermediate Shell	215-02A	C-7-1	A-4567-1	Lukens
Intermediate Shell	215-02B	C-7-2	8-9427-1	Lukens
Intermediate Shell	215-02C	C-7-3	A-4567-2	Lukens
Lower Shell	215-03B	C-8-1	C-5935-1	Lukens
Lower Shell	215-03C	C-8-2	C-5935-2	Lukens
Lower Shell	215-03A	C-8-3	C-5935-3	Lukens



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TABLE III-2 REACTOR VESSEL BELTLINE WELDS

Location	Weld Seam No.	<u>Wire Heat No.</u>	Flux Type	Flux Batch
Intermediate Shell Longi- tudinal Seam	2-203A	A-8746/34B009 M/A IAGI*	Linde 124	3878 & 3688
Intermediate Shell Longi- tudinal Seam	2-203B	A-8746/34B009 M/A IAGI* M/A CBBF*	Linde 124	3878 & 3688
Intermediate Shell Longi- tudinal Seam	2-203C _.	A-8746/34B009 M/A IAGI* M/A CBBF*	Linde 124	3878 & 3688
Lower Shell Longitudinal Seam	3-203A	348009 M/A KBEJ*	Linde 1092	3889
Lower Shell Longitudinal Seam	3-203B	34B009 M/A KBEJ* M/A JACJ*	Linde 1092	3889
Lower Shell Longitudinal Seam	3-203C	34B009 M/A KBEJ*	Linde 1092	3889
Intermediate to Lower Girth Seam	9 - 203 1	90136 M/A ABEA* M/A FOAA*	Linde 0091	3999

*Manual shielded metal arc electrode (all others automatic submerged arc wire).

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-6-

TABLE III-3 REACTOR VESSEL BELTLINE PLATES CHEMICAL ANALYSIS (WEIGHT PERCENT)

Element	<u>C-7-1</u>	<u>C-7-2</u>	<u>C-7-3</u>	<u>C-8-1</u>	<u>C-8-2</u>	<u>C-8-3</u>
¢;	17	20	20	17	17.	10
51	•17	.20	.20	•17	•17	.13
3	.013	.010	.012	.010	.010	.010
P 	• 004	.004	.004	.006	.006	•004
Mn	1.28	1.28	1.33	1.28	1.29	1.22
C	.24	.23	.21	.28	.28	.22
Cr	.03	.03	.06	.07	.07	.06
Ni	.64	.64	.58	.56	.57	.58
Мо	.60	.59	.58	.65	.66	.59
۷	.003	.003	.003	.002	.002	.002
СЬ	<.01	<.01	<.01	<.01	<.01	<.01
В	.0003	.0002	.0002	.0003	.0003	.0001
Со	.005	.006	.005	.007	.007	.006
Cu	.11	.11	.11	.15	.15	.12
A1	.018	.020	.020	.027	.025	.022
W	<.01	<.01	<.01	<.01	<.01	<.01
Ti	<.01	<.01	<.01	<.01	<.01	<.01
As	.015	.015	.010	.013	.014	.011
Sn	.006	.006	.006	.009	.010	.006
Zr	.002	.002	.002	.002	.002	.002
N ₂	.006	.007	.007	.008	.009	.006
-						







TABLE III-4 SURVEILLANCE PLATE AND WELD METAL CHEMICAL ANALYSIS

	Weight Percent		
Element	Plate $C-8-2^{(a)}$	Weld <u>C-8-1/C-8-3</u> (b)	
Si	.17	.20	
S	.010	.012	
Р	.006	.013	
Mn	1.29	1.02	
C	.28	.12	
Cr	.07	.06	
Ni	.57	.11	
Мо	.66	.55	
V	.002	.006	
СЬ	<.01	<.01	
В	.0003	.0001	
Со	.007	.004	
Cu	.15	.23	
A1	.025	.021	
W	<.01	<.01	
Ti	<.01	<.01	
As	.014	.014	
Sn	.010		
Zr	.002	.001	
N2	.004	.008	

a - Heat C-5935-2

b

Mil B-4 wire heat 90136, Linde 0091 Flux lot 3999



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-8-



Drop weight, Charpy impact and tension test specimens were machined from the surveillance materials as described in reference 1. In addition to the surveillance material specimens, Charpy impact specimens were machined from a section of plate Ol from the Heavy Section Steel Technology (HSST) program to serve as standard reference material (SRM).

The surveillance and SRM test specimens were enclosed in six capsules for irradiation in the St. Lucie Unit 1 reactor vessel. The surveillance capsule assembly is shown in Figure III-1. Each assembly consists of four compartments containing Charpy impact specimens (Figure III-2) and three compartments (Figure III-3) containing tension specimens and monitors (flux and temperture). Each capsule is positioned in a holder tube attached to the reactor vessel cladding to irradiate the specimens in an environment which duplicates as closely as possible that experienced by the reactor vessel. Capsule locations are shown in Figure III-4. The axial portion of each capsule is bisected. by the midplane of the core. The circumferential locations were selected to coincide with the peak flux regions of the reactor vessel.

The existing Technical Specification withdrawal schedule for the surveillance capsules is given in Table III-5.

The type and quantity of test specimens contained in the W-97 capsule are given in Table III-6.



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POWER & LIGHT CO. St. Lucie Unit 1

TYPICAL CHARPY IMPACT COMPARTMENT ASSEMBLY





TABLE III-5 EXISTING TECHNICAL SPECIFICATION SCHEDULE FOR ST. LUCIE UNIT 1 REACTOR VESSEL SURVEILLANCE CAPSULE REMOVAL

Removal Sequence	Azimuthal <u>Location</u>	Approximate Removal Time (Years)	Target <u>Fluence (n/cm²)</u>
1	, 97 ⁰	8	3.2×10^{18}
2	104 ⁰	16	5.7 × 10^{18}
3	284 ⁰	23	8.3×10^{18}
4	263 ⁰	30	1.2×10^{19}
- 5	277 ⁰	35	1.4×10^{19}
-6	830	40	1.7 x 10 ¹⁹





TABLE III-6 TYPE AND QUANTITY OF SPECIMENS IN W-97 CAPSULE

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<u>Material</u>	Charpy Impact	<u>Tension</u>
Base Metal (Transverse)	12	-
Base Metal (Longitudinal)	12	3
Weld Metal	12	3
Heat-Affected Zone	12	3
Total	48	9

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.CAPSULE WITHDRAWAL AND DISASSEMBLY

The St. Lucie Unit 1 reactor vessel was shut down for refueling on February 26, 1983. A special* retrieval tool was attached to the W-97 capsule adapter to disengage the latches, and the capsule was withdrawn from its holder and out of the reactor vessel. The W-97 capsule was transferred to the spent fuel pool where it was sectioned into lengths for insertion into a shipping cask. Sectioning was accomplished by drilling to separate the individual specimen compartments.

The surveillance capsule was shipped to Neutron Products, Inc. in Dickerson, Maryland, for inspection, disassembly and specimen removal in the hot cell facility. No unusual features or damage were revealed by visual inspection. An inventory of the mechanical test specimens removed from the W-97 capsule is given in Table IV-1.

*A special tool was used since the lock assembly adapter (ACME threaded nose cone which normally serves as the point of attachment for the retrieval tool) had previously disengaged from the assembly and was removed from the reactor vessel.



IV.

TABLE IV-1 MECHANICAL TEST SPECIMENS REMOVED FROM W-97 CAPSULE

Compartment	Material and	
Number	Specimen Type	Specimen Identification
7214	HAZ Tension	4JC, 4JP, 4J5
7224	HAZ Charpy	42C, 425, 41U 42T, 42B, 427 42D, 42A, 42P 42E, 426, 41Y
7232	Base Metal Charpy (Transverse)	232, 231, 22D 23A, 22B, 22A 22E, 227, 237 23K, 23L, 22J
7241	Base Metal Tension	1JY, 1KB, 1KC
7251	Base Metal Charpy (Longitudinal)	131, 12T, 114, 12L, 12U, 127 116, 12A, 134 132, 113, 115
7263	Weld Metal Charpy	36P, 323, 34C, 35P, 347, 334 341, 35B, 31A 33E, 36E, 371
7273	• Weld Metal Tension	3KC, 3L3, 3JT



-17-



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TEST RESULTS

A. Irradiation Environment

1. Temperature Monitors

Each Tensile-Monitor Compartment (Figures III-1 and III-3) in the capsule assembly contained a set of four temperature monitors to provide an indication of the maximum temperature in the capsule during irradiation. The composition and melting point of each eutectic alloy monitor are given in Table V-1. Each monitor consisted of a helix of the eutectic alloy and a stainless steel weight encapsulated in a quartz tube. Each set of four temperature monitors was inserted into a stainless steel housing, and the temperature monitors were irradiated in the top, middle and bottom surveillance compartments.

Post-irradiation examination of the temperature monitors was performed at C-E's Windsor, Connecticut facility. Each temperature monitor was identified by length, photographed, and inspected to determine whether the eutectic alloy helix had melted and been crushed by the weight. Photographs of the three sets of monitors. at 5X are shown in Figures V-1 through V-3. The 536° F monitors (80% Au - 20% Sn) were completely melted. The 558° F monitor (90% Pb - 5% Sn - 5% Ag) helices were distorted but exhibited no melting. Each set of monitors exhibited similar features, indicating that the maximum irradiation temperature was in the 550° F - 558° F range and uniform along the length of the surveillance capsule.

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-18-

TABLE V-1 COMPOSITION AND MELTING POINTS OF. TEMPERATURE MONITOR MATERIALS

Composition (Weight %)	Melting Temperature	
80 Au, 20 Sn	536	
90 Pb, 5 Sn, 5 Ag	558	
97.5 Pb, 2.5 Ag	580	
97.5 Pb, 0.75 Sn, 1.75 Ag	590	





FIGURE V-1

TEMPERATURE MONITORS COMPARTMENT 7214 (5x)



FIGURE V-2

TEMPERATURE MONITORS COMPARTMENT 7241 (5x)


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536F

580F

558F

590F

2. Neutron Dosimetry

Each Tensile-Monitor compartment (Figures III-1 and III-3) in the capsule assembly contained two sets of neutron flux monitors as described in Table V-2. Each flux monitor was encapsulated in a stainless steel sheath (except for the sulfur which had a quartz sheath); in addition, cadmium covers were placed around the uranium, cobalt, nickel and copper monitors which have competing thermal activities. Each set of nine flux monitors was inserted into two stainless steel housings, one set for each of the top, middle and bottom surveillance capsule compartments.

The flux monitors were removed from the capsule compartments in the hot cell. Each monitor was inspected and its position in the housing verified by the number of grooves in the stainless steel sheath. The monitors were then repackaged and shipped to C-E's Windsor, Connecticut facility for radiochemical analysis. The identity of individual monitors to their originating compartments was lost because of an error in packaging at the hot cell.

a. Radiochemical Analysis

Radiochemical analysis of the flux monitors was performed in accordance with C-E Procedure 00000-FMD-401, Rev. O, November 1, 1978 ("Standard Method for the Analysis of Radioisotopes in Reactor Irradiation Surveillance Detectors and Flux Distribution Monitors"). The samples were prepared for radiochemical analysis using standard methods. Atomic absorption spectroscopy was used for the uranium, copper and nickel monitors to establish the amount of monitor material recovered. Simple gravimetric methods were used for the remaining monitors.

-22-

TABLE V-2 NEUTRON FLUX MONITORS

<u>Monitor</u>	<u>Material</u>	<u>Reaction</u>	Threshold Energy (MeV)	<u>Half-Life</u>
1	Cobalt (Cadmium Shielded)	Co ⁵⁹ (n, _Y)Co ⁶⁰	Thermal	5.27 years
2	Uranium*	U ²³⁸ (n,f)Cs ¹³⁷	1.2	30.2 years
3	Titanium	Ti ⁴⁶ (n,p)Sc ⁴⁶	8.0	65 days
4	Iron	Fe ⁵⁴ (n,p)Mn ⁵⁴	4.0	312.5 days
5	Cobalt	Co ⁵⁹ (n, _Y)Co ⁶⁰	Thermal	5.27 years
6	Uranium* (Cadmium Shielded)	U ²³⁸ (n,f)Cs ¹³⁷	1.2	30.2 years
7.	Nickel (Cadmium Shielded)	Ni ⁵⁸ (n,p)Co ⁵⁸	3.5	70.8 days .
8	Copper (Cadmium Shielded)	Cu ⁶³ (n,α)Co ⁶⁰	7.0	5.27 years
9	Sulfur	S ³² (n,p)P ³²	2.9	14.3 days

*U-238 foil depleted in U-235 to .05 w/o





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Gamma counting was performed with a 4096 channel gamma spectrometer system coupled with a hyperpure germanium detector. The system was calibrated at 0.5 Kev per channel to span the gamma energy range from 0.05 to 2 Mev. Efficiency calibration was performed using eight (8) gamma energies emitted from a mixed isotope standard traceable to the National Bureau of Standards. Phosphorus-32 beta activity was measured on a gas flow proportional counter which was calibrated with NBS traceable beta standards.

Physical constants used in the calculation of radioisotope activity levels are as follows:

Isotope	<u>Half-Life</u>	Gamma Energy (MeV)	<u>Intensity</u> (a)
Cobalt-58	70.8 days	0.8105	0.99
Cobalt-60	5.27 years	1.3325	1.00
Cesium-137	30.2 years	0.6616	0.85
Manganese-54	312.5 days	0.8347	1.00
Scandium-46	65 days	0.8892	1.00
Phosphorus-32	14.3 days	(b)	1.00

- a Intensity is branching ratio (gamma rays or beta particles per disintegration).
- b Decays by beta only.



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Flux spectrum monitor activity levels are presented in Table V-3. All values are decay corrected to the time of reactor shutdown, February 26, 1983. The uncertainty listed with each result is the 2-sigma counting error only. An additional error of $\pm 20\%$ for uranium monitors, $\pm 12\%$ for copper and nickel monitors and $\pm 7\%$ for all other metal monitors is estimated from volumetric and gravimetric operations and from the stated uncertainties of calibration isotopes. An additional error associated with the sulfur monitor results is $\pm 8\%$.

The shutdown activities determined from gamma ray emission rates were calculated as follows:

$$A = \frac{N_p}{EWBC \ (e^{-\lambda t})}$$

where: A =

= shutdown activity in disintergrations per minute
per milligram of material (dpm/mg)

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- Np = radioisotope net counts per minute
- E = full energy peak efficiency (counts per gamma ray emitted)
- W = weight of monitor sample (milligrams)
- B = radioisotope gamma ray branching ratio (gamma rays per disintegration)
- C = correction for coincident or random summing
- λ = radioisotope decay constant
- t = elapsed time between plant shutdown and counting

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TABLE V-3 ST. LUCIE UNIT 1 FLUX SPECTRUM MONITOR ACTIVITIES

		Monitor	Number of	Measured	Weight	Activity at End of
		<u>Material</u>	Grooves	<u>Isotope</u>	<u>(mg)</u>	Irradiation (dmp/mg)
	(1)	Cobalt (shielded)	0	Co-60	7.8 7.6 7.6	2.75+0.04x10 ⁵ 2.78+0.04x10 ⁵ 2.76+0.04x10 ⁵
	(2)	Uranium	I	Cs -137	24.6 24.7 36.9	 3.66+0.09x104 4.96+0.11x104 4.30+0.09x104
	(3)	Titanium ,	2	Sc-46 '	13.0 12.9 13.0	3.52 <u>+</u> 0.20x10 ⁴ 3.78 <u>+</u> 0.20x10 ⁴ 3.96 <u>+</u> 0.21x10 ⁴
	(4)	Iron ·	3	Mn-54	23.8 23.9 23.6	
ſ	(5)	Cobalt	. 4 ·	Co-60	8.0 7.7 7.4	1.83 <u>+</u> 0.01x106 1.32 <u>+</u> 0.01x106) 1.80 <u>+</u> 0.01x106
	(6)	Uranium (shielded)	[·] 5	Cs - 137	12.4 10.1 11.2	1.49+0.09x10 ⁴ 2.01+0.11x10 ⁴ 2.30+0.12x10 ⁴
	(7)	Nickel (shielded)	6	[.] .Co−58	22.3 19.6 •21.1	1.95+0.01×106 2.07+0.01×106 1.74+0.01×106
	(8)	Copper (shielded)	7	Co-60	17.5 21.5 20.9	8.58+0.19x103 8.28+0.17x103 9.09+0.39x103
	(9)	Sulfur	-	P-32	8.4 19.4 17.1	3.72+0.03x105 4.17+0.03x105 1.19+0.01x106



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b. Threshold Detector Analysis

The SAND-II⁽³⁾ and DOT IV version $4.3^{(4)}$ computer codes were used to calculate the fast flux and fluence at the surveillance capsule assembly location and at the reactor vessel.

The DOT IV neutron transport code was used in order to determine the spatial distribution of the neutron flux and the spectrum associated with it within the St. Lucie 1 vessel. The St. Lucie l geometry was modelled in the DOT calculation using R0 coordinates and octant symmetry as shown in Figure V-4. The DOT calculations used a source distribution deduced from pinwise PDQ power distributions. Figures V-5 and V-6 show assembly average power distributions for Cycles 1 through 5 and the planned distributions for Cycle 6, respectively, which were deduced from the PDQ results. A major difference which is evident in these two power distributions is the change from an out-in to a lowleakage fuel management for Cycles 5 and 6. For extrapolation to future cycles, the Cycle 6 fast fluxes were used. The removal of the St. Lucie 1 thermal shield after Cycle 5 contributed the only change to the physical geometry of St. Lucie 1. The DLC-23E CASK, 40 group neutron and gamma ray cross section library was used in the DOT calculation. A representation of the surveillance capsule detail, as used in the DOT model, is shown in Figure V-7.

The SANB II code which was used in the analysis of dosimeter activation requires, as input, saturated dosimeter activities (A_{sat}) . These saturated activities are deduced from the operating history and the decay constants for the various dosimeters. In order to obtain the saturated activities, equation 1 is used to convert the measured (unsaturated) activities to saturation in units of (dps/a).

(1)
$$A_{sat} = \frac{A_{unsat} * A_{W}}{N_a * I * S} * 16.67 \frac{mg.min}{g.sec}$$

-27-

where: $A_{sat} = saturated activity (dps/a)$ Aunsat = measured activity (dpm/mg) from Table V-3 $A_w = atomic weight (g/mole)$ N_{Δ} = Avogadros number (atoms/mole) I = isotopic abundance of target isotope S = saturation factor (see below)

In order to determine the saturation factor(s) for equation 1, the operating history of St. Lucie 1 was modelled by intervals of constant power (P_i) relative to a full power operation (P_{o}) for Cycles 1-5. Hence the saturation factor for each isotope may be given as:

(2)
$$S = \sum_{i}^{\Sigma} \left(\frac{P_{i}}{P_{o}}\right) f_{i} e^{-\lambda T} i \left[1 - e^{-\lambda t} i\right]$$

where: P_i = power of the ith time interval. $P_0 = full power level$ T; = time from end of interval i to reactor shutdown t_i = length of time interval i f_i = radial power distribution adjustment factor λ = decay constant

The activity for the short-lived isotopes ($t_{1/2}$ < cycle length) in the St. Lucie 1 dosimeters is indicative of the leakage flux for Cycle 5 and, therefore, in this case where the peripheral assembly powers decreased in the fifth cycle relative to the first 4 cycles, the monitor activity for the short lived isotopes underestimates the leakage flux relative to the average leakage

-28-

over the five cycles. For very long-lived isotopes, the saturation factor builds up to a steady rate through all five cycles. As a first order correction, the short-lived saturation factor was adjusted by setting f_i equal to the ratio of the weighted assembly average power level factor for the surveillance capsule position during Cycle 5 to the average over Cycles 1 to 5. For the long lived isotopes, f_i is approximately 1 as the average power distribution through Cycles 1 to 5 is seen by these isotopes. This correction is somewhat approximate due to the fact that the power distribution within the cycle is changing relative to the cycle average. Once again, this has the greatest effect on the short-lived isotopes.

Using the DOT flux spectrum at the surveillance capsule as input, the SAND code was used in the activity mode to calculate saturated activities relative to the normalized DOT spectrum. From this, an effective cross section $(\overline{\sigma})$ is obtained as shown in. equation 3. These $\overline{\sigma}$'s are based on the ENDF/B-V dosimetry cross section library, DOSCROS81, Reference 5.

(3)
$$\overline{\sigma} = \frac{A_{sat} \text{ (calculated)}}{\int_{1.0}^{\infty} \phi(E) dE}$$

From this effective $\overline{\sigma}$, the fast neutron flux (ϕ) above 1 MeV is found by:

(4)
$$\phi(>1 \text{ MeV}) = \frac{A_{sat}}{\sigma}$$
 (measured)

Using this method, an inferred flux is obtained for each monitor material.

The dosimeters from the three separate flux monitor compartments were mixed when shipped from the hot cell and, therefore, individual monitors were not identified as to their originating compartments. The activity levels of a given dosimeter type were •

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pooled as one set for the SAND analysis, and the compartment specific results were obtained by using axial factors at the capsule compartment elevations derived from an axial RZ DOT calculation. Table V-4 shows the results of the analysis of the axial variation in the surveillance capsule fast neutron flux.

Tables V-2 and V-3 summarize the flux monitors used and their measured activities. Table V-5 shows the saturated activity in units of disintegrations per second per atom (dps/a) calculated using Equation 1, the correction factors discussed above, and as used in the SAND analysis. The uranium fission monitor activities were also corrected for the thermal and epithermal fission of the U-235 impurities (350 ppm). The resulting correction factor (0.935) was applied to the measured fission product activity to obtain the activity due to U-238 fissions. The U-238 fission product activities were input to the SAND analysis in terms of fissions per second per U-238 atom (fps/a). This value was obtained by dividing the saturated activity (A_{sat}) by the fission yield for U-238 of the fission product being analyzed. The fission yields were obtained from ASTM E704-79 and give a fission yield of 5.99% for Cs-137. No correction has been included for photofission yield of U-238. Neglect of the photofission correction provides a conservative (higher) result for the fast neutron flux.

Figures V-8 and V-9 show the fast neutron flux azimuthal distribution at the vessel/clad interface for Cycles 1 to 5 and for Cycle 6 normalized to their respective peak azimuthal values as given in Table V-6.

The average axial power distribution for Cycles 1 to 5 and for Cycle 6 are shown in Figures V-10 and V-11. The axial peak for Cycles 1 to 5 is 1.18 and 1.11 for Cycle 6. These source distributions were used in RZ DOT calculations to obtain the axial fast flux distributions for BOC to EOC5 and Cycle 6 at the vessel clad interface as shown in Figures V-12 and V-13, respectively.





The fluence values in Table V-7 were based on the average flux values in Table V-6. For the extrapolation to end of life (EOL), Cycle 6 values were used for operation beyond Cycle 5. This extrapolation is based on the assumption of a uniformly thick core support barrel.*

Reference 3 states that the SAND code will give fluxes that are accurate to within \pm 10% to \pm 30% if the errors in the measured activities are within similar limits. In Table V-3, the quoted uncertainties are at a 2-Sigma level and are determined from counting statistics. An additional error of \pm 20% for uranium monitors, \pm 12% for copper and nickel monitors, and \pm 7% for all other metal monitors is estimated from volumetric and gravimetric operations and from the stated uncertainties of calibration isotopes.

Taking into account the adjustments used in the saturation factor correction and in the use of pooled data from different capsule locations, an estimated \pm 35% uncertainty should be used for the measured flux at the surveillance capsule. This uncertainty does not include variations that may occur in the Lead Factor due to changes in the azimuthal distributions which could result from different fuel management strategies.

* The repaired core support barrel (CSB) will contain holes of various sizes that are filled with water and covered on one end by 1/4 to 1/2 inch thick stainless steel plugs. These nonuniformities will cause neutron streaming at these locations and hence higher flux and fluence values at the corresponding vessel locations. These hole effects are treated in a separate report⁽¹¹⁾.

















Figure V-7 DOT MODEL OF W-97 SURVEILLANCE CAPSULE



-35-

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Axial Variation in Surveillance Capsule Fast Flux

Surv Axt	veillance Capsule ial Compartment	Axial Position Relative to Core Midplane (CM)	Fast Neutron (E > 1 Mev) Flux (n/cm ² .s)
	-,		
	7214	+89.6	3.6 + 10
	7241	+ 2.9	3.7 + 10
	7273	-83.8	3.5 + 10

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TABLE V-5

Flux Monitor Type	Measured Activity (dpm/mg)	Saturated Activity (dps/a)
1 1 2 (Cs)* 2 (Cs) 2 (Cs) 2 (Cs) 3 3 3 4 4 4 4 4 5 5 6 (Cs) 6 (Cs) 6 (Cs) 7 7 7	(dpm/mg) 2.75+5 2.78+5 2.76+5 3.66+4 4.96+4 4.30+4 3.52+4 3.78+4 3.96+4 1.09+5 1.18+5 1.16+5 1.83+6 1.32+6 1.80+6 1.49+4 2.01+4 2.30+4 1.95+6 2.07+6 1.74+6 8.58+3	(dps/a) 6.16-13 6.25-13 6.21-13 2.09-14** 2.83-14** 2.46-14** 6.33-16 6.80-16 7.12-16 3.49-15 3.78-15 3.78-15 3.71-15 4.11-12 2.97-12 4.05-12 1.53-14 2.06-14 2.36-14 5.04-15 5.34-15 4.49-15 5.12-17
8 8 9 9 9	8.28+3 9.09+3 3.72+5 4.17+5 1.19+6	4.94-17 5.42-17 3.84-16** 4.31-16** 1.23-15**

Flux Monitor Saturated Activities

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*Fission Yield of Cs-137 is 5.99% . **These monitors were not used in the SAND analysis

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-38-





Figure V-9 ST. LUCIE I CYCLE 6 AVERAGE AZIMUTHAL FAST NEUTRON (E_n > 1 MeV) FLUX DISTRIBUTION AT VESSEL/CLAD INTERFACE



-39-

TABLE V-6

St. Lucie 1 Average Fast Neutron Flux (E_n>1 MeV) (n/cm².s) (At Peak Axial and Azimuthal Location)

	BOL to EOC5 Actual	Cycle 6 Average (Without Thermal Shield)
Surveillance Capsule	3.7+10(1)	4.4+10
Vessel/Clad Interface	2.7+10	3.7+10
1/4 Thru Vessel	1.4+10	2.0+10
1/2 Thru Vessel	6.6+9	9.3+9
3/4 Thru Vessel	2.9+9	4.1+9

TABLE V-7 .

Fast Neutron Fluence $(E_n > 1 \text{ MeV})$ (n/cm^2) (At Peak Axial and Azimuthal Location)

•	BOL to EOC5 ² (Actual)	BOL TO EOL ³ (Without Thermal Shield)
Surveillance Capsule	5.5+18	4.3+19
Vessel/Clad Interface	3.9+18	3.5+19
1/4 Thru Vessel	2.1+18	1.9+19
1/2 Thru Vessel	9.8+17	9.0+18
3/4 Thru Vessel	4.3+17	3.9+18

¹ Numbers shown as 3.7+10 are to be interpreted as 3.7×10^{10} .

² 4.67 EFPY (1.473+8s) @ 2700 Mwt.

³ EOL corresponding to 32 EFPY (1.009+9s) @ 2700 Mwt; extrapolation based on planned Cycle 6 operating characteristics.









-41-





















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B. Chemical Analysis

Six Charpy specimens each of the base metal (transverse orientation) and weld metal were chemically analyzed by X-ray fluorescence for molybdenum, copper, nickel, manganese, silicon, sulfur and phosphorus content. Each Charpy specimen was placed in a carrier with a graphite mask for analysis. Calibration curves were initially established for the seven elements using nine plate and weld specimens with a known chemical content. One of these specimens (11E) was used to check for reproducibility with copper and phosphorus as the selected elements. Twelve separate measurements yielded a copper reproducibility of \pm 1% and a phosphorus reproducibility of \pm 7% at one standard deviation. Specimen 11E was also used as a control for irradiated specimen measurements.

Results of the analysis of the irradiated specimens and the control specimen are given in Table V-8. (Results for the unirradiated specimens are also included.) The base metal specimens represent three different sections of the surveillance plate at the 1/4 and 3/4 thickness locations. The weld metal specimens represent six separate regions through the thickness of the weld as shown in Figure V-14, which was based on the weld metal specimen layout drawing.⁽⁶⁾

C. Strength and Toughness Properties

1. <u>Tension Tests</u>

Tension tests were conducted in accordance with applicable ASTM standards and C-E laboratory procedures. The test method and equipment are described in Appendix A.

The three irradiated specimens from each material (base metal, weld metal and heat-affected zone) were tested at room tempera-





TABLE V-8

IRRADIATED PLATE AND WELD MATERIAL CHEMICAL ANALYSIS

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Specimen	Lab			Chemi	cal Cont	tent (We	eight P	<u>ercent)</u>		
ID	<u>Number</u>	<u>Material</u>	Мо	Cu	Ni	Mn	Si	S	ρ	
227 232 23L 132 113 12A	2220 2223 2226 2222 2224 2225	Plate(WR) Plate(WR) Plate(WR) Plate(RW) Plate(RW) Plate(RW)	0.62 0.61 0.62 0.62 0.62 0.63	0.14 0.13 0.14 0.14 0.14 0.14	0.57 0.57 0.58 0.57 0.56 0.57	1.26 1.28 1.29 1.28 1.26 1.29	0.21 0.21 0.20 0.20 0.20 0.20 0.20	0.018 0.018 0.018 0.018 0.018 0.018	0.007 0.007 0.007 0.008 0.006 0.009	
371 36P 323 31A 341 36E	2215 2216 2217 2218 2219 2221	Weld Weld Weld Weld Weld Weld	0.53 0.53 0.53 0.53 0.52 0.53	0.22 0.22 0.32 0.32 0.22 0.22	0.05 0.05 0.06 0.06 0.06 0.06	1.07 1.06 1.04 1.05 1.07 1.07	0.20 0.21 0.20 0.20 0.20 0.20	0.012 0.012 0.012 0.012 0.012 0.012 0.012	0.014 0.016 0.015 0.018 0.016 0.016	
11E 11E 11E	#1 #2 #3	Control Control Control	0.56 0.56 0.55	0.15 0.15 0.15	0.54 0.53 0.54	1.24 1.23 1.22	0.21 0.21 0.20	0.014 0.014 0.014	0.009 0.010 0.009	
Unirradia	ted Data									
23C 25K 216 221 233 235	1978 1979 1980 1981 1982 1983	Plate(WR) Plate(WR) Plate(WR) Plate(WR) Plate(WR) Plate(WR)	0.63 0.63 0.62 0.63 0.62 0.63	0.14 0.14 0.14 0.14 0.14 0.14	0.58 0.59 0.58 0.58 0.58 0.58	1.32 1.33 1.33 1.32 1.33 1.33	0.21 0.21 0.21 0.21 0.21 0.21 0.21	0.018 0.018 0.017 0.018 0.017 0.018	0.007 0.007 0.006 0.007 0.007 0.007	
32L 34T 333 356 36C 362	1984 1985 1986 1987 1988 1989	Weld Weld Weld Weld Weld Weld	0.53 0.53 0.53 0.53 0.53 0.52 0.53	0.21 0.20 0.44 0.20 0.20 0.21	0.06 0.05 0.06 0.05 0.05 0.05	1.09 1.07 1.08 1.07 1.10 1.11	0.21 0.21 0.21 0.21 0.21 0.21 0.21	0.012 0.012 0.011 0.012 0.011 0.012	0.014 0.013 0.013 0.014 0.014 0.013	

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FIGURE V-14 LOCATION OF WELD METAL CHEMICAL ANALYSIS SPECIMENS (COPPER CONTENT IN PARENTHESES)

$ \begin{array}{ c c c c c c c c c c c c c c c c c c c$	٢
32L ^b 36E ^a (.21) 362 ^b 362 ^b (.21)	
32L 36E ^a (.21) 362 ^b 362 ^b (.21)	
362 ^b (.21)	
36P ^a	
(.22)	
31A ^a	
(.32)	
356 ^b	
(.20)	
333 ^b	
· (.44)	
36C ^b 371 ^a	
(.20)(.22) вот	том

SIDE VIEW

------ WELDING DIRECTION ------>



a Irradiated specimens

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b Baseline specimens

ture, $250^{\circ}F$ and $550^{\circ}F$. The tensile properties are listed in Table V-9, and the stress-strain curves are shown in Figure V-15 through V-23. The pre-irradiation tensile properties⁽²⁾ are summarized in Table V-10 (each value average of three tests). Photographs of the broken irradiated specimens are shown in Figure V-24.

2. Charpy V-Notch Impact Tests

Charpy V-notch impact tests were conducted in accordance with applicable ASTM standards and C-E laboratory procedures. The test method and equipment are described in Appendix B.

Twelve irradiated specimens from each material (transverse and longitudinal base metal, weld metal and heat-affected zone) were tested at a series of temperatures to establish the transition temperature behavior. The impact data (impact energy, lateral expansion and fracture appearance as a function of test temperature) are shown in Tables V-11 through V-14 and Figures V-25 through V-36. (Also shown in each of the figures is the unirradiated transition temperature curve from the baseline evaluation.⁽²⁾) Fracture surface photographs of the broken irradiated specimens are shown in Figures V-37 through V-40.

Each impact test was instrumented. Additional data related to instrumented impact testing are presented in Appendix C.





TABLE V-9 POST-IRRADIATION TENSION TEST PROPERTIES

<u>Material</u>	Specimen Code	Test Temp. _(⁰ F)	Yield Strength 0.2% Offset (ksi)	Ultimate Tensile Strength .(ksi)	Fracture Load (1b)	Fracture(a) Strength -(ksi)	Fracture) Stress ·(ksi)	Reduction of Area (%)	Elongation (1-inch gage) TE/UE(d) (%)
Base Metal	1KB	72	80.6	103.0	3240	65.5	201.7	67.5	26/9.7
	1JY	250	76.4	96.9	3240	66.0	190.9	65.4	24/8.8
	1KC	550	71.5	98.4	3270	66.6	177.9	62.5	23/8.1
Weld Metal	ЗКС	72	84.9	98.0	2970	60.0	198.6	69.8	28/12.2
	3JT	250	79.5	91.6	2940	59.9	191.0	69.1	25/11.1
	3L3	550	75.2	94.4	3370	68.7	165.5	58.5	26/8.6
	•					•			
HAZ	4J5	72	78.2	98.6	3018	61.5	187.9	67.3	27/8.3
	4JC	250	75.2	94.7	2986	60.8	196.8	69.1	(c) /7.5
	4JP	550	75.2	95.6	3192	65.0	158.8	59.0	24/9.1

a - Fracture strength is the fracture load divided by initial cross sectional area

b - Fracture stress is the fracture load divided by final cross sectional area

c - Not determined; specimen broke outside gauge length

d - TE = total elongation, UE = uniform elongation

-49-


PRE-IRRADIATION TENSION TEST PROPERTIES

Material	Test Temp. (^O F)	Yield Strength 0.2% Offset (ksi)	Ultimate Tensile Strength _(ksi)	Fracture Load (1b)	· Fracture(a) Strength _(ksi)	Fracture Stress (ksi)	Reduction of Area (%)	Elongation (1-inch gage) TE/UE(C) (%)
Base Metal	72	71:4	92.3	2720	56.8	207	73	27/10.3
	250	66.8	86.2	2640	53.9	186	71	23/8.1
	550	63.9	90.1	2940	61.0	182	67	25/8.9
Weld Metal	72	71.9	84.7	2300	46.9	186	75	30/11.1
	250	69.0	79.0	2400	48.9	180	73	26/9.5
	550	65.9	84.6	2760	56.2	175	64	26/9.9
HAZ	72	66.8	86.0	2580	52.6	204	52	27/10.0
	250	66.2	80.8	2640	53.8	. 179	69	23/8.0
	550	62.4	84.4	2730	55.0	178	69	24/8.5

a - Fracture strength is the Fracture Load divided by initial cross sectional area

b - Fracture stress is the Fracture Load divided by final cross sectional area

c - TE = total elongation, UE = uniform elongation

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Figure V-16 STRESS-STRAIN RECORD, BASE METAL PLATE C-8-2 SPECIMEN No. 1JY, TEST TEMPERATURE 250° F



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FIGURE V-24 IRRADIATED TENSION SPECIMENS

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TABLE V-11 CHARPY V-NOTCH IMPACT RESULTS IRRADIATED BASE METAL (TRANSVERSE) (PLATE C-8-2).

Specimen Test Identification Temperatur <u>Number (^OF)</u>		Impact Energy (Ft-1bs.)	Lateral Expansion (mils)	Fracture Appearance (% Shear)	
22A	0	5.5	0	0	
23A	40	22.5	13	0	
22D	60	19.5	17	10	
237	60	27	19	10 _	
232	80	34.5	29	20	
22J	80	35	26	10	
23L	120	38	32	20	
227	160	52	40	30	
23K	160	54.5	44	40	
22B	210	, 7 6	56	80	
22E	250	67	56 .	70	
231	250	77.5	66	100	

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TABLE V-12 CHARPY V-NOTCH IMPACT RESULTS IRRADIATED BASE METAL (LONGITUDINAL) (PLATE C-8-2)

Specimen Identification	Test Temperature	Impact Energy	Lateral Expansion	Fracture Appearance
Number	()	(Ft-105.)	(m115)	(% Silear)
12T	0	6	1	0
12A	40	23	16	0
114	60	25	. 20	10
131	60	25.5	17	10
113	80	32.5	23	10
· 134	80	33	22	10
116	120,	44	35	30 .
132	160	69	52	40
127	200	107.5	77	100
120	250	87	66	80
115	250	110	85	100
12L	300	102	76	ِ 100 _.

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-58-

TABLE V-13							
CHARPY	V-NOTCH	IMPACT	RESULTS				
IRF	RADIATED	WELD ME	ETAL				

Specimen Identification Number	Test Temperature (^O F)	Impact Energy (Ft-1bs.)	Lateral Expansion (mils)	Fracture Appearance (% Shear)
34C	-40	6.5	0	0
36E	0	15	11	10
35B	20	55	28	30
33E	- 40	28.5	23	20
347	40	51	37	20 _
371	· 60 *	61	51	60
36P	60 .	62.5	50	50
. 35P	80	[,] 82	63	70 •
341	120	89	70	80
31A -	160	102.5	81	100
. 323	200 •	95	72	100
33Y	200	102.5	77	100

-59-

TABLE V-14 CHARPY V-NOTCH IMPACT RESULTS IRRADIATED HAZ METAL (BASE METAL PLATE C-8-2)

Specimen Identification Number	Test Temperature (^O F)	Impact Energy (Ft-1bs.)	Lateral Expansion (mils)	Fracture Appearance (% Shear)	
42D	-40	25	15	10	
42E	-20	38.5	25 ·	30	
42B	-20	41 .	39	40 ·	
426	0	61	38	40	
41Y	40	68	42	50 -	
42P	40	72.5	. 48	70	
42A	80	48	34	30	
425	100	90	66 .	80	
427	120	94	66	80	
42C	160	72.5	49	70	
410	210	109	76 [.]	100	
42T	210	118.5	72	100	

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Figure V-25 CHARPY IMPACT ENERGY, BASE METAL PLATE C-8-2 (TRANSVERSE)







Figure V-26 CHARPY LATERAL EXPANSION, BASE METAL PLATE C-8-2 (TRANSVERSE)









Figure V-27 CHARPY SHEAR FRACTURE, BASE METAL PLATE C-8-2 (TRANSVERSE)











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-65-



Figure V-30 CHARPY SHEAR FRACTURE, BASE METAL PLATE C-8-2 (LONGITUDINAL) ń.

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-66-



Figure V-31 CHARPY IMPACT ENERGY, WELD METAL

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Figure V-33 CHARPY SHEAR FRACTURE, WELD METAL





Figure V-34 CHARPY IMPACT ENERGY, HEAT-AFFECTED ZONE



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-70-



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Figure V-35 CHARPY LATERAL EXPANSION, HEAT-AFFECTED ZONE



-71-



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Figure V-36 CHARPY SHEAR FRACTURE, HEAT-AFFECTED ZONE





-73-



FIGURE V-38

FRACTURE SURFACES OF CHARPY V-NOTCH IMPACT TEST SPECIMENS BASE METAL (LONGITUDINAL)

12A 40⁰F

113 80⁰F







Specimen No.: Test Temperature:



131 60⁰F

Specimen No.: Test Temperature:





Specimen No.: Test Temperature: 12U 250⁰F



115. 250⁰F















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-75-





FIGURE V-40





-76-

A. Irradiation Exposure -

The W-97 surveillance capsule was removed from the St. Lucie Unit 1 reactor vessel following plant shutdown on February 26, 1983. The surveillance specimens were irradiated at approximately $550^{\circ}F$; the temperature variation along the length of the capsule was negligible based on the similarity in appearance of the three $558^{\circ}F$ monitors.

The surveillance capsule was exposed to a maximum neutron flux of 3.7×10^{10} n/cm²-s (E>1.0 MeV). The highest flux was calculated for the middle compartment (7241). The neutron flux in the top compartment (7214) and bottom compartment (7273) was approximately 3.6×10^{10} n/cm²-s and 3.5×10^{10} n/cm²-s, respectively, indicating a total axial variation of 6%. The maximum surveillance capsule fluence after 4.67 effective full power years (EFPY) operation at 2700 Mwt was, therefore, 5.5×10^{18} n/cm² (E>1.0 MeV).

For the first five cycles, the average lead factors between the neutron fluence at the surveillance capsule and the maximum azimuthal position on the reactor vessel were calculated to be 1.41 at the vessel-clad interface, 2.62 at the quarter thickness location in the vessel, and 12.8 at the three-quarter thickness location.* The maximum neutron flux at the vessel-clad interface was, therefore, 2.7×10^{10} n/cm²-s. The predicted end-of-life fluence at the vessel-clad interface is 3.5×10^{19} n/cm² ($\pm 30\%$); the Cycle 6 fuel management strategy was used as the basis for extrapolation beyond Cycle 5 to end-of-life for operation without the thermal shield.

*Lead factors subsequent to Cycle 5 will be lower due to a difference in fuel management strategy.



VI.

B. Chemical Analysis

X-ray fluorescence analysis of the irradiated base metal and weld metal Charpy specimens (Table V-8) demonstrated that the encapsulated surveillance specimens had the same chemical composition as that originally reported (Table III-4). The base metal chemistry was uniform throughout the surveillance plate. The weld metal chemistry, however, ranged from 0.22 to 0.32 w/o copper for the irradiated specimens and 0.20 to 0.44 w/o copper for the unirradiated specimens. From Figure V-14, the specimens exhibiting 0.32 w/o copper or greater (specimen numbers 31A and 323) came from one end of the surveillance weld, whereas the balance of the weld exhibited lower copper (0.20 to 0.22 w/o). None of the remaining six elements analyzed showed a significant variation indicating that the differences in copper content were a function of the weld wire copper coating thickness.

C. Uniaxial Tension Properties

Radiation induced changes in uniaxial tension properties of the St. Lucie Unit 1 surveillance materials were determined from a comparison of Tables V-9 and V-10. The yield strength and ultimate strength increased an average of 14% following irradiation to 5.5×10^{18} n/cm². Uniform and total elongation changed very little following irradiation, while reduction in area decreased about 7% (excluding HAZ material). Fracture stress (fracture load divided by final cross sectional area) was not changed significantly by irradiation. In general, property changes were similar for each of the materials despite the large difference in copper content between base metal and weld metal.

Post-irradiation room temperature yield strength values ranged from 78Ksi for the HAZ to 85Ksi for the weld metal. Total elongation of the irradiated materials ranged from 23% to 28%.

-78-

D. Charpy Impact Toughness Properties

The radiation induced changes in toughness of the St. Lucie Unit I surveillance materials are summarized in Table VI-1. Index temperature shifts were measured using the average curves at the 30 ft-1b level (Cv30), 50 ft-1b level (Cv50), and 35 mils lateral expansion level (Cv35). Upper shelf energy changes were based on the average impact energy for each set of test specimens exhibiting 100% shear fracture measured before and after irradiation. The unirradiated Charpy impact data were obtained from the baseline evaluation report.⁽²⁾

The base metal and weld metal exhibited similar shifts in the 30 ft-lb index temperature ($68-74^{\circ}F$). The shift for the heat-affected-zone material was significantly less (28F). The decrease in upper shelf energy was similar for the irradiated base and weld materials, ranging from 23 to 31%.

The St. Lucie Unit 1 design curve for prediction of transition temperature shift as a function of neutron fluence is given in Figure B3/4.4-1 of the Technical Specifications.⁽⁷⁾ The design curve prediction for the surveillance capsule exposure of 5.5×10^{18} n/cm² is 137° F, or 85% to 90% higher than the measured shifts for the base metal and the weld metal. In contrast, shifts predicted using NRC Regulatory Guide $1.99^{(8)}$ based on measured copper and phosphorus content are 82° F for the base metal (17% more than measured) and 159° F for the weld metal (115% more than measured).

In order to provide an improved means of predicting transition temperature shift of the surveillance material for St. Lucie Unit 1, Figure VI-1 was developed. Figure VI-1 was developed based on the methodology of Regulatory Guide $1.99^{(8)}$ using the measured value of shift at the 30 ft-1b level for the weld metal $(74^{\circ}F)$ at 5.5×10^{18} n/cm². The shift versus neutron fluence relationship is as follows:

-79-



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TABLE VI-1 SUMMARY OF TOUGHNESS PROPERTY CHANGES FOR ST. LUCIE UNIT 1 SURVEILLANCE MATERIALS (550F IRRADIATION, 5.5 $\times 10^{18}$ n/cm²)

<u>Material</u>		Index Temperatures (⁰ F)				Upper Shelf Energy	Upper Shelf Energy	
	<u>Cv_30^a</u>	<u>ACv 30</u>	<u>Cv 50^b</u>	<u>∆Cv50</u>	<u>Cv 35^C</u>	<u>ACv 35</u>	Energy (ft-1b)	Change (%)
Base Metal (WR)	16 ^d	-	40 ^d		36 ^d		103 ^d	4
	86	70	150	110	126	<u>)</u> 90	78	24
Base Metal (RW)	8 ^d	*	43 ^d		35 ^d		139 ^d	
•	76	68	126	83	117	82	107	23
Weld Metal	-53 ^d		-36 ^d		-35 ^d		144 ^d	
	21	74	44	80	37	72	100	31
Heat-Affected	^	·· ·	.d	-	b		d	
Zone	-60-		-4-		-16°	•	133 ⁴	
	-32	28	0	4	7 <u>,</u>	23	114	14

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- a 30 ft-1b Index Temperature
- **b** 50 ft-1b Index Temperature
- c 35 mils lateral expansion Index Temperature
- d Unirradiated values



FIGURE VI-1 PREDICTED TRANSITION TEMPERATURE SHIFT FOR THE ST. LUCIE UNIT 1 SURVEILLANCE MATERIALS



-81-

NDTT = 100
$$(\frac{\cancel{0}}{10^{19}})^{0.5}$$

where

This relationship should be evaluated again once results are available from surveillance materials irradiated to a higher fluence.

The measured shift for the weld material was only 6% higher than that for the base metal material even though the copper content of the weld was significantly higher (0.23% Cu for the weld versus 0.15% Cu for the plate). The greater irradiation resistance of the weld is a result of the low nickel content (0.11%). Based on an empirical evaluation of weld metal irradiation sensitivity⁽⁹⁾, low nickel weldments (e.g. less than 0.2% Ni) exhibited significantly smaller NDTT shifts than high nickel weld-Similar results were observed with the Millstone Unit 2 ments. surveillance weld.⁽¹⁰⁾ Therefore, the low weld metal shift determined from the St. Lucie Unit 1 W-97 surveillance capsule evaluation is consistent with surveillance and experimental results. Furthermore, it is reasonable to expect that both the base metal and the weld metal will continue to exhibit similar shifts in subsequent analyses.

As noted in Section VIB, chemical analysis of the baseline and irradiated weld metal Charpy specimens indicated that a region of the weld contained higher copper (0.32 to 0.44 w/o Cu) than the balance of the weld (0.20 to 0.22 w/o Cu). The two irradiated specimens (31A and 323) identified as containing 0.32 w/o Cu

-82-

exhibited Charpy impact properties consistent with the specimens containing lower copper. A third specimen (33E) which was not analyzed but which bordered on the higher copper region of the weld exhibited test results below the mean energy versus test temperature curve. However, the results were within the overall scatterband. Pending results from future surveillance capsules, it does not appear that the range of measured copper content has a significant effect on the radiation sensitivity of the surveillance weld.

Projected values of NDTT shift and adjusted RT_{NDT} are given in Table VI-2. Using the highest initial reference temperature (20⁰F) for the surveillance materials (base metal transverse) and the revised shift prediction method and vessel fluence, the predicted adjusted RT_{NDT} (vessel inside surface) after 32 EFPY is 207°F. Similarly, the 1/4 t adjusted RT_{NDT} is projected to be 158⁰F. (Note that the revised shift prediction method. applies to the specific plates and welds represented by the surveillance materials; i.e., it applies to the six beltline plates and to weld seams 9-203 and 2-203. It should not be applied to weld seams 3-203 or 8-203* because of the higher nickel content in these welds compared to the surveillance weld. Further information concerning shift prediction methodology for these welds is provided in separate report⁽¹¹⁾.)

*Weld seam 8-203 is the girth weld between the intermediate and upper shell course. It is located outside of the beltline as defined in lOCFR50, Appendix G.



-83-
TABLE VI-2 PROJECTED NDTT SHIFT AND ADJUSTED RTNDT FOR SURVEILLANCE MATERIAL

		END-OF-LIFE	····	
Vessel Location	<u>Fluence^a</u>	(32 EFPY) <u>NDTT SHIFT</u>	<u>Adj. RTNDT^b PLATE WELD</u>	
Inside Surface	3. <u>5</u> x 10 ¹⁹ n/cm ²	187 ⁰ F	207 ⁰ F	127 ⁰ F
1/4 t	1.9 x 10 ¹⁹ n/cm ²	138 ⁰ F	158 ⁰ F	78 ⁰ F
3/4 t	3.9 x 10 ¹⁸ n/cm ²	62 ⁰ F	82 ⁰ F	2 ⁰ F

- a Projected fluence assuming same power level, coolant inlet temperature and fuel management strategy as for Cycle 6.
- b Adjusted RT_{NDT} = Initial RT_{NDT} plus NDTT shift, where initial RT_{NDT} is 20F for surveillance plate and -60F for surveillance weld.





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The predicted decrease in upper shelf energy at end-of-life based on the method given in Regulatory Guide $1.99^{(8)}$ is 41% for the weld and 32% for the plate at the one-quarter thickness location in the vessel. Using this conservative prediction, the upper shelf energy of the plates will remain above 70 ft-lb during the design life of the vessel. The upper shelf energy of the weld will remain above 85 ft-lb throughout vessel life. These projected values of upper shelf energy are well in excess of the 50 ftlb value currently considered to provide a reasonable margin for continued safe operation.

Recommended changes to the surveillance capsule removal schedule are contained in Table VI-3. The schedule is designed based on 10CFR50, Appendix H and a time-averaged lead factor (capsule to vessel ID) of 1.25. (For example, the fourth capsule's exposure after 19 EFPY will be equivalent to that at the vessel inside surface after 24 EFPY). The 104° and 284° capsules are designated for standby purposes.

TABLE VI-3 PROPOSED NEW CAPSULE REMOVAL SCHEDULE FOR ST. LUCIE UNIT 1

Removal	Azimuthal	Alternate	Approximate
Sequence	Location	Location	Removal Time ^a
1	97 ⁰	-	4.67 EFPY ^b
2	. 263 ⁰	83 ⁰ , 277 ⁰	7-9 EFPY
3	830	277 ⁰	12-14 EFPY
4	277 ⁰	-	17-20 EFPY
5	104 ⁰ ·	-	Standby
6	284 ⁰	<u>ب</u>	- Standby

a) Time in effective full power years (EFPY) at 2700 Mwt.

b) Actual removal time.



-86-

REFERENCES

- "Program for Irradiation Surveillance of Hutchinson Island Plant Reactor Vessel Materials", Combustion Engineering, INC., F-NCM-007, September 15, 1970.
- "Florida Power & Light Company, St. Lucie Unit No. 1, Evaluation of Baseline Specimens, Reactor Vessel Materials Irradiation Surveillance Program", TR-F-MCM-005, to be issued.
- 3. "SAND Neutron Flux Spectra Determination by Multiple Foil Activation - Iterative Method", CCC-112, Oak Ridge National Laboratory, Oak Ridge, Tenn.
- 4. "DOT-IV Version 4.3 Two Dimensional Transport Code System", CCC-429, Oak Ridge National Laboratory, Oak Ridge, Tenn.
- "DOSDAM 81-82, Multigroup Cross-sections in SAND-II Format for Spectral, Integral and Damage Analysis", ORNL, DLC-97, Oak Ridge, Tenn.
- 6. C-E Drawing E-19367-165-111-02, "Weld Metal Test Specimens", January 28, 1972.
- 7. Florida Power & Light Company, St. Lucie Unit 1, Safety . Technical Specifications, Docket 50-335.
- 8. Regulatory Guide 1.99, Revision 1, "Effects of Residual Elements on Predicted Radiation Damage to Reactor Vessel Materials", April 1977.
- 9. J. D. Varsik and S. T. Byrne, "An Empirical Evaluation of the Irradiation Sensitivity of Reactor Pressure Vessel Materials", <u>Effects of Radiation on Structural Materials</u>, ASTM STP 683, 1979.
- "Northeast Utilities Service Company, Millstone Nuclear -Unit No. 2; Evaluation of Irradiation Capsule W-97", "TR-N-MCM-008, April 1982.
- Florida Power & Light Company, St. Lucie Unit No. 1 (Thermal Shield Removal and Core Support Barrel Repair Evaluation Report), to be issued.

-87-

VII.

APPENDIX A

TENSION TESTS - DESCRIPTION AND EQUIPMENT

The tension tests were performed using a Riehle universal screw testing machine with a maximum capacity of 30,000 lb and separate scale ranges between 50 lb and 30,000 lb. The machine, shown in Figure A-1, is capable of constant cross head rate or constant strain rate operation. The tension testing was covered by the certificate of calibration which is included at the end of the Appendix A.

Elevated temperature tests were performed in a 2-1/2" ID x 18" long high temperature tension testing furnace with a temperature limit of 1800F. A Riehle high temperature, dual range extensometer was used for monitoring specimen elongation.

The tension specimen is depicted in Figure A-2. Figures A-3 through A-5 are isometric drawings showing the orientation and location of the tension specimens in the base metal, weld metal and heat-affected-zone, respectively.

Tension testing was conducted in accordance with ASTM Method E 8-82, "Tension Testing of Metallic Materials" and/or Recommended Practice E 21-79, "Elevated Temperature Tension Tests of Metallic Materials," except as modified by Section 2.1 of Recommended Practice E 184-79, "Effects of High-Energy Neutron Radiation on the Mechanical Properties of Metallic Materials." Implementation of the ASTM Test Methods to the testing of irradiated tension specimens is described in C-E Laboratory Procedure 00000-MCM-041, Revision 0, "Procedure for Tension Testing of Irradiated Metallic Materials," August 16, 1978.

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FIGURE A-1 TENSION TEST SYSTEM WITH CONTROL CONSOLE AND ELEVATED TEMPERATURE TESTING EQUIPMENT



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FIGURE A-2 TYPICAL TENSION SPECIMEN







A-3 LOCATION OF TENSION SPECIMENS WITHIN BASE METAL TEST MATERIAL

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FIGURE A-5 LOCATION OF TENSION SPECIMENS WITHIN HEAT-AFFECTED-ZONE TEST MATERIAL



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MEASUREMENT SYSTEMS DIVISION 929 CONNECTICUT AVENUE, BOX 9021 BRIDGEPORT, CONNECTICUT 06602 (203) 335-2511 TELEX 96-4224

Certificate of Calibration

Riehle® Testing Machine-

Calibration Date

Machine Description

3-10-83 Company Serial No. Combustion Engineering STA 44372 Location Byte Windson Conn. Measurement Systems Division of Page-Wilson Corporation certifies that the machine described above has been calibrated to ASTM designation E4 using calibrated weights and/or elastic-calibration devices calibrated to National Bureau of Standards specification.

TENSION

Machine Range	3000	
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Machine Range

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Calibration Engineer بالمشارع المراجع المراجع Standards Manager

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PAGE-WILSON CORPORATION

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AEASUREMENT SYSTEMS DIVISION CONNECTICUT AVENUE, BOX 9021 RIDGEPORT, CONNECTICUT 06602 (203) 335-2511 TELEX 96-4224

Certificate of Calibration Riehle[®] Testing Instrument

Calibration Date

8-10-83 Сотралу

Combustion Engineering Location

Windsor Conn.

Measurement Systems Division of Page-Wilson Corporation verifies that the attached graph is certification of calibration of the instrument described above. This instrument was calibrated to ASTM designation E83.

Instrument Description

DN-120007

Serial No.

Glieh 6 DND 10-20

Richle RD53 5/ R-67338 <u>Equipment used in calibration</u>

Callbration Engineer

Standards Manager

A-8

APPENDIX B

CHARPY IMPACT TESTS - DESCRIPTION AND EQUIPMENT

The standard impact tests and instrumented tests were performed on a calibrated instrumented impact testing system, shown in Figure B-1. C-E's instrumented impact test equipment provides for signal retention and the subsequent data analysis. The output signal from the instrumented tup is recorded by an oscilloscope. Permanent records were made of the load signal and integrated energy as it was displayed on the oscilloscope screen with a poloroid camera and with a computer printout.

The system consists of the following elements:

- a. A Model SI-1 BLH Sonntag Universal Impact Machine with a specifically machined pendulum tup, instrumented with four resistance strain gages in full bridge circuit. This tup "load cell" is calibrated statically and dynamically to provide a given pounds/volt sensitivity for known settings of the balance and gain on the dynamic response system. The instrumented machine meets all impact test machine requirements of ASTM and is certified by AMMRC, the U.S. Army Materials and Mechanics Research Center (Watertown Arsenal): A copy of the certification papers is included in this Appendix.
- b. A model 500 Dynatup dynamic response system which supplies regulated and constant dc excitation to strain gages on the pendulum tup, provides balancing, variable load sensitivity and calibration functions, and amplifies load-time signal to a <u>+</u>10 volt, <u>+</u>100 milliampere level while preserving kHz frequency response and 0.05 percent accuracy while simultaneously recording the area beneath the load-time trace.



- c. A photoelectric triggering device and velocometer composed of a high intensity light directed through a grid mounted on the pendulum of the impact tester, and passed to a photosensor through fiber optics. A special circuit ensures accurate, reliable and fail safe triggering of the oscilloscope recorder plus an accurate display of the average velocity of the pendulum during impact.
- d. A 5113 Dual Beam Tektronix Storage Oscilloscope with a No. 5A18N dualtrace amplifier plug-in unit and a No. 5B12N dual time base plug-in unit. Also included is a C-58 camera with mounting adapter. This device gives a display of each test trace for visual analysis of the load-time impulse recorded by the instrument.

The standard Charpy specimen is described in Figure B-2. Figures B-3 through B-5 are isometric drawings showing the orientation and location of the Charpy impact specimens in the base metal, weld metal and heat-affected-zone, respectively.

All Charpy impact tests were conducted in accordance with ASTM Method E 23-82, "Notched Bar Impact Testing of Metallic Materials." Implementation of ASTM E23for the testing of irradiated Charpy specimens is described in C-E.Laboratory Procedure 00000-MCM-040, Revision 0, "Procedure for Instrumented Charpy Impact Testing of Irradiated Metallic Materials," July 31, 1978.

The constant temperature necessary for conducting the Charpy impact tests was obtained from a series of circulating liquid baths capable of maintaining stable temperature throughout the range of $-150^{\circ}F$ to $+250^{\circ}F$. For test above $250^{\circ}F$, specimens were heated in a controlled circulation furnace where temperature was maintained to an accuracy of $5^{\circ}F$. The temperature baths were composed of the following equipment:

Two Neslab Constant Temperature Circulating Baths - Model TEZ 10, with Model CT 150 Thermoregulators and Labline 11 inch diameter thermocups, Designated Baths 1 and 4.



B-2

Medium: Ethylene Glycol - room temperature to 250°F.

One Neslab Constant Temperature Circulating Bath - Model TEZ 10 with a Model CT 59 Thermoregulator and a Labline 11 inch diameter thermocup. Designated Bath 2.

Medium: Isopropanol - room temperature to -10⁰F. Neslab Portable Bath Cooler, Model PCB-2 connected.

One Low Temperature Stirred Bath, one 11 inch thermocup, one Honeywell Controller and Solenoid control valves to Flexi-Cool cooling system. Designated Bath 5.

Medium: Isopropanol - room temperature to -150⁰F. Coolant: Freon

One Grieve Industrial oven, controlled air circulation. Designated Bath 3.

Medium: Air, 100°F to 800°F.

All baths - Copper Constantan Thermocouple Honeywell Six Point Temperature Chart Recorder Digitec Themocouple Thermometer - Model 590 TF Standard Mercury Column Thermometer Bimetallic - spring Thermometer

The temperature instruments were calibrated in accordance with the ASME Boiler and Pressure Vessel Code, Section III, Paragraph 2360. Copies of the applicable certificates are provided at the end of this Appendix.





FIGURE B-1

CHARPY IMPACT TEST SYSTEM, ASSOCIATED CONSTANT TEMPERATURE BATHS AND INSTRUMENTED CHARPY IMPACT DATA PROCESSING EQUIPMENT











B-6



FIGURE B-4

LOCATION OF CHARPY IMPACT SPECIMENS WITHIN WELD METAL TEST MATERIAL







DEPARTMENT OF THE ARMY A ARMY MATERIALS AND MECHANICS RESEARCH CENTER WATERTOWN, MASSACHUSETTS 02172

REPLY TO ATTENTION OF

DRXMR-STM

5 July 1983

Combustion Engineering, Inc. ATTN: Mr. R. J. Hurlburt 1000 Prospect Hill Road P.O. Box 500 Windsor, CT 06095-0500

Dear Mr. Hurlburt:

A set of Charpy impact test specimens broken on the 240 ft-1b capacity Satec machine, Serial No. 1366 has been received for evaluation along with the ______ completed questionnaire.

The results of the tests indicate the machine to be producing acceptable energy values at both energy levels (see inclosed table).

It is noted that one specimen has unusally sharp anvil marks. This could be caused by sharp radius or by not properly positioning the specimen against the anvil supports. The anvils' radii should be checked to assure that the radii are 0.039'' + 0.002''. If the anvils are within limits, proper care should be taken when positioning the specimens against the anvils.

This machine satisfies the proof-test requirements of ASTM Standard E-23.

If this machine is moved or undergoes any major repairs or adjustments, this certification becomes invalid and the machine must be rechecked. Removal of the pendulum, replacement of anvils or adjusting the height of drop are examples of such major repairs or adjustments. It should be noted that if a . specimen requires over 80% of the machine capacity to fracture, the machine should be checked to assure that the pendulum is straight, the anvils or striker have not been damaged and that all bolts are still tight. This certification is valid for one year from the date of the test.

Sincerely yours,

B-9

l Encl Table

ROGER M. LAMOTHE Chief, Mechanical Behavior & Testing Branch



ARMY MATERIALS AND MECHANICS RESEARCH CENTER

Watertown, Massachusetts 02172

Date of Test: 4 June 1983

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TABLE

COMPARISON TESTS ON CHARPY IMPACT MACHINES

Combustion Engineering, Inc.Facility1000 Prospect Hill Rd., P.O. Box 500, Windsor, CT 06095Make of MachineSatec SystemsSerial No.1366

	AMMRC		Variation	
	(ft-1b)	(ft-1b)	Actual	Allowed
High Energy	69.9	70.7	+1.1 %	<u>+</u> 5.0%
Low Energy	11.6	11.9	+0.3 ft-1b	<u>+</u> 1.0 ft-1b

XMR Form 105 Rev 1 Apr 81



COMBUSTION LNGHTCHING, INC. Nuclear Laboratories INSTRUMENT CALIBRATION REQUIREMENT SHEET

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14

DATE: ______

INSTRU	MENT	READA	BILITY	CALIB	RATION	CHECKED
FUNCTION	Туре	RANGE	MIN READABILITY	ACCURACY	FREQUENCY	ВҮ
Thermometer	Digital	-313 ⁰ F to +752 ⁰ F	1 ⁰ F	<u>+</u> 1 ⁰ F	3 mos.	
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APPENDIX C

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INSTRUMENTED CHARPY V-NOTCH DATA ANALYSIS

All baseline and irradiated Charpy impact tests in this program were performed on an instrumented test system. Instrumented impact testing provides more quantitative data from a Charpy specimen which enable a more detailed analysis of the surveillance material toughness behavior.

Photographs of the oscilloscope traces and computer printouts of load and energy versus time were taken for each test of the base plate (transverse and longitudinal orientation), weld, and heat-affected-zone. From each trace, the general yield load (PGY), maximum load (PM), and fracture load (PF) were determined, as shown in Tables C-1 through C-4. For each material, the loads were plotted against the corresponding test temperature to generate the irradiated load/temperature diagrams. The post-irradiation load/temperature results are shown in Figures C-1 to C-4.

Three index temperatures are of interest. T_B , the brittle transition temperature, corresponds to the onset of ductile fracture; below T_B the fracture is completely brittle. T_N , the ductility transition temperature, corresponds to the mid-transition region where the fracture has become predominantly ductile. T_D , the ductility temperature, corresponds to the onset of the upper shelf energy where fracture is completely ductile.

The radiation-induced toughness property changes of the surveillance materials are summarized in Table C-5. Standard Charpy impact data are included with the instrumented data since each method represents a unique material property. The standard Charpy test provides a measurement of the total energy to initiate and propagate a crack through to failure of the material. In contrast, analysis of the instrumented data enables characterization of the components of the dynamic load behavior prior to material failure. The shift in the brittle transition temperature, T_B , and the ductility transition temperature, T_N , are comparable to the shift in the 30 ft-lb Charpy index temperature, Cv_{30} . The radiation-induced changes in the instrumented data therefore tend to corroborate the changes determined from the standard Charpy impact data.

C-1

The third-parameter obtainable from the instrumented data is T_D , the ductility temperature, which is given in Table C-5. T_D corresponds closely with the onset of the upper shelf energy (minimum temperature for the material to exhibit 100% shear fracture). The agreement is seen to hold for both the unirradiated and irradiated data.

The instrumented Charpy analysis substantiates the results from the standard impact tests. In particular, this approach provides a more quantitative means of measuring radiation-induced property changes by analysis of the entire load record rather than using the single measurement of impact energy. As more experience is gained with this technique, it offers the potential of providing a more quantitative measurement of toughness property changes than is possible with current impact testing.

TABLE C-1 INSTRUMENTED CHARPY IMPACT TEST, ST. LUCIE UNIT 1 IRRADIATED BASE METAL (TRANSVERSE)

.

	Test			Fast Fracture
Specimen	Temperature	Yield Load,	Maximum Load,	Load,
Identification	(⁰ F)	PGY (1b)	РМ (1Ь)	PF (1b)
22A	0			3800
23A	40	3400	-4300	* =
22D	60	3350	4000	
237	60	3100	4200	
232	80	3250	4300	
22J	80	3000	4200	
23T	120	3100	4300	,
227	160	3050	4250	4200
23K	160	2950	4200	4000
22B	210	2850	4100	3200
^ 22E	250	2800	4000	3500
231	250	2900	4000	



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TABLE C-2

INSTRUMENTED CHARPY IMPACT - TEST, ST. LUCIE UNIT 1 IRRADIATED - BASE METAL (LONGITUDINAL)

	Test			Fast Fracture
Specimen	Temperature ·	Yield Load,	Maximum Load,	Load,
Identification	(⁰ F)	PGY (1b)	РМ (1Ъ)	PF (1b)
12T	0	3700	3900	
12A	. 40	3350	4250	
114	Ġ0	´3200	• 4250	
131	60	3200	· 4200	
113	80	3150	4200	
134	80 .	3200	4250	
116	120	3100	4250	
132	160	3050	4200	3800
127	200	2950	4200	
120	250	- 2900	4150	3300
115	250	2900	4150	
12L	300	2850	4000	•



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TABLE C-3 INSTRUMENTED CHARPY IMPACT TEST, ST. LUCIE UNIT 1 IRRADIATED WELD METAL

,

	Test			Fast Fracture
Specimen	Temperature	Yield Load,	Maximum Load,	Load,
Identification	(PGT (1D)	PM (1D)	Pr (ID)
34C	-40	3750	4250	,
36E	0	3400	3950	
- 35B	20	3450	4300	
33E	. 40	3300	4100	
347	40	3400	4250	
371	60	3300 -	4150	
36P ⁻	60	3300	4100	3900
35P	80	3150	4050 -	3250
341	120	3050	4000	3250
31A	160	3050	4100	
323	200	2850	3900	
33Y	200	2900	3900	·

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TABLE C-4INSTRUMENTED CHARPY IMPACTTEST, ST. LUCIE UNIT 1 IRRADIATEDHEAT-AFFECTED-ZONE

.

	Test			Fast Fracture
Specimen	Temperature	Yield Load,	Maximum Load,	Load,
Identification	(⁰ F)	PGY (1b)	РМ (1Ь)	PF (1b)
42D	-40	4000	4400	. ==
42E	-20	3850	4350	
42B	-20	3800	4500	
426	0	3800	<i>,</i> 4600	
41Y	40	3750	4600	4400
42P	40	3700	4650	4000
42A	. 80	3500	4400	4100
425	100	3300	4400	3900
427	120	3500	4500	3500
42C	160	3200	4400	3900
410	210	3050	4200	,
42T	210	3000	4250	

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TABLE C-5 TOUGHNESS PROPERTY CHANGES BASED ON INSTRUMENTED CHARPY IMPACT TEST

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<u>Material</u>	T _B (⁰ F)	ΔT _B (⁰ F)	T _N (⁰ F)	∆T _N (^o f)	∆Cv ₃₀ (⁰ F)	<u>∆Cv₅₀</u>	T _D (^o f)	Min. Temperature for 100% Shear Fracture (^O F)
Base Metal (WR)						•		•
unirrad	-52		32				114	110
irrad	5	57	140	108	70	110	221	250
Base Metal (RW)			``					· .
unirrad	-52	 *	51				150	160
irrad	-4	48	130	· 79	68	83	280	260
Weld Metal			-					
unirrad	-96		-30				36 -	80
irrad	-59	43	40	70	74	80	135	160
Heat-Affected-Zon	ne					۰		
unirrad	-130		-8				62	120 ·
irrad	-60	70	6	14	28	4	130	210







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Figure C-3 INSTRUMENTED CHARPY LOAD - TEMPERATURE DIAGRAM, WELD METAL







Figure C-4 INSTRUMENTED CHARPY LOAD - TEMPERATURE DIAGRAM, HEAT-AFFECTED ZONE

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