

**EPRI**

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RESEARCH INSTITUTE

# **Welding Issues: Alloy 52 Weldability & Testing; Magnetic Stir Welding; Laser & Friction Stir Welding on Irradiated Material**

**NRC/Industry Technical Meeting**

**Rockville, MD**

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Charlotte, North Carolina, USA

# Presentation Roadmap

- **Alloy 52 / 52M Challenges**
- **Cracking Mechanisms**
- **Weldability Testing**
  - Description of Testing and Ranking of Filler Metals
  - Dilution Issues and Testing
  - High Chromium Nickel-Base Alloy Development
- **Magnetic Stir Welding**
- **Welding Irradiated Material**
  - Problem and Issue
  - Laser Welding
  - Friction Stir Welding

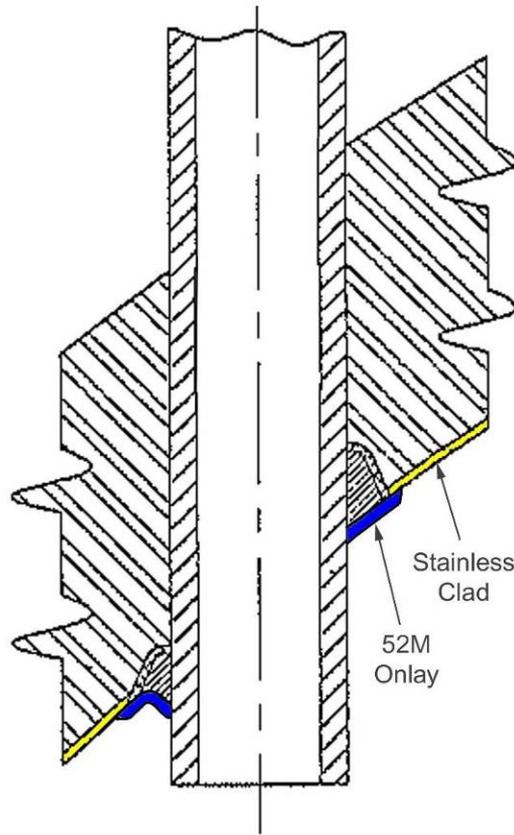
# Filler Metal 52 & 52M Welding Challenges

- Alloy 52 / 52M weldability issues
  - Sluggish weld puddle
  - Heat-to-heat variations can cause significant difference in weldability
- Ti & Al oxide buildup
- Tendency for lack of bond and/or lack of fusion
- Susceptible to various types of weld metal cracking
  - ***Ductility-dip cracking (DDC)***
  - Liquation cracking
  - ***Solidification (hot) cracking***



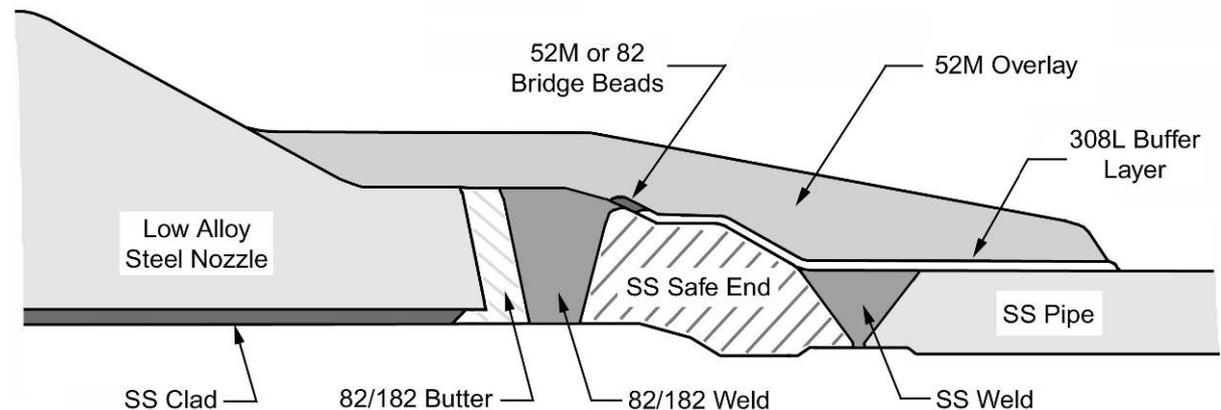
# 52 / 52M Must be Welded on Variety of Materials

- 52 / 52M must weld successfully over a variety of materials
  - Low alloy steel, 82/182 weld metal, SS weld metal, 304 & 316 base metals, CF8A & CF8M cast materials, etc.



**CRDM Schematic with 52M Overlay**

## Schematic of 52M Structural Weld Overlay



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# Cracking Mechanism No. 1

- **Solidification Cracking (type of hot cracking)**
  - Occurs in the weld fusion zone at the terminal stage of solidification in brittle temperature range (BTR)
  - Associated with liquid films along grain boundaries
  - Low melting constituents segregate to grain boundaries where they form liquid films that separate during thermal contraction of the weld
  - Controlled by volume fraction of low melting point liquid, grain boundary area, and wetting characteristics

# Cracking Mechanism No. 2

## • Ductility-Dip Cracking (DDC)

- Occurs in weld metal HAZ during multi-pass welding
- Associated with a sharp drop in ductility at temperatures slightly above the recrystallization temperature ( $\sim \frac{1}{2} T_L$  to  $\frac{3}{4} T_L$  range)
- One theory suggests that DDC occurs during rapid grain growth in the ductility-dip temperature range (DTR) along migrated grain boundaries
- Low impurity weld metals, such as filler metal 52, have low fraction of 2<sup>nd</sup> phase particles to control and obstruct grain growth

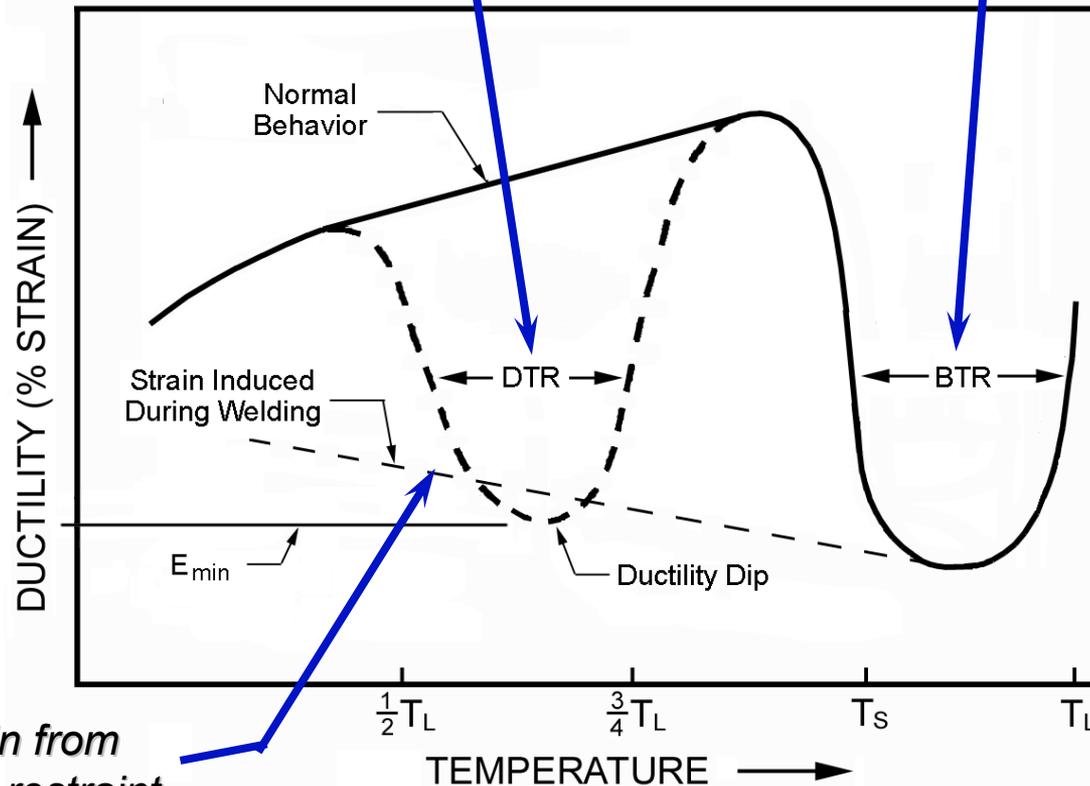
# Ductility-Dip and Brittle Temperature Ranges

## Ductility-dip Temperature Range (DTR)

range of ductility-dip envelope  
(~ 10% to 15% strain)

## Brittle Temperature Range (BTR)

~ liquidus to terminal solidus range



applied strain from  
shrinkage and restraint

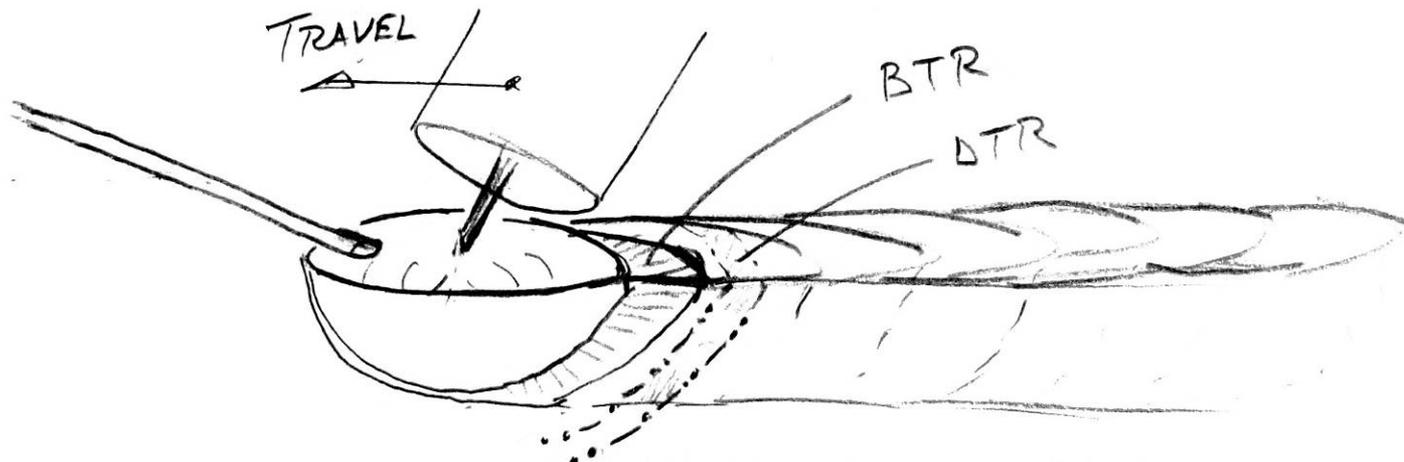
# Solidification & Ductility-Dip Crack Locations

- **Solidification Cracks**

- Initiate in the Brittle Temperature Range (BTR)
- Typically surface connected and sometimes subsurface in weld fusion zone where shrinkage strain is high enough to cause rupture

- **Ductility-Dip Cracks (DDC)**

- Initiate in the Ductility-dip Temperature Range (DTR)
- Typically subsurface in reheated weld metal where strain is high enough to cause rupture



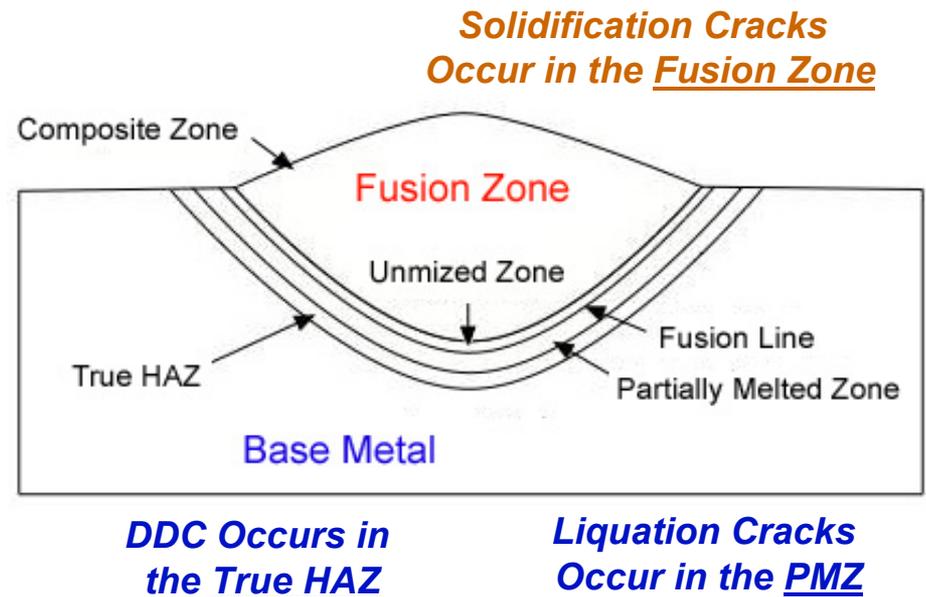
# Weld Zone Definitions and Crack Locations

- **Heat Affected Zone (HAZ) includes:**

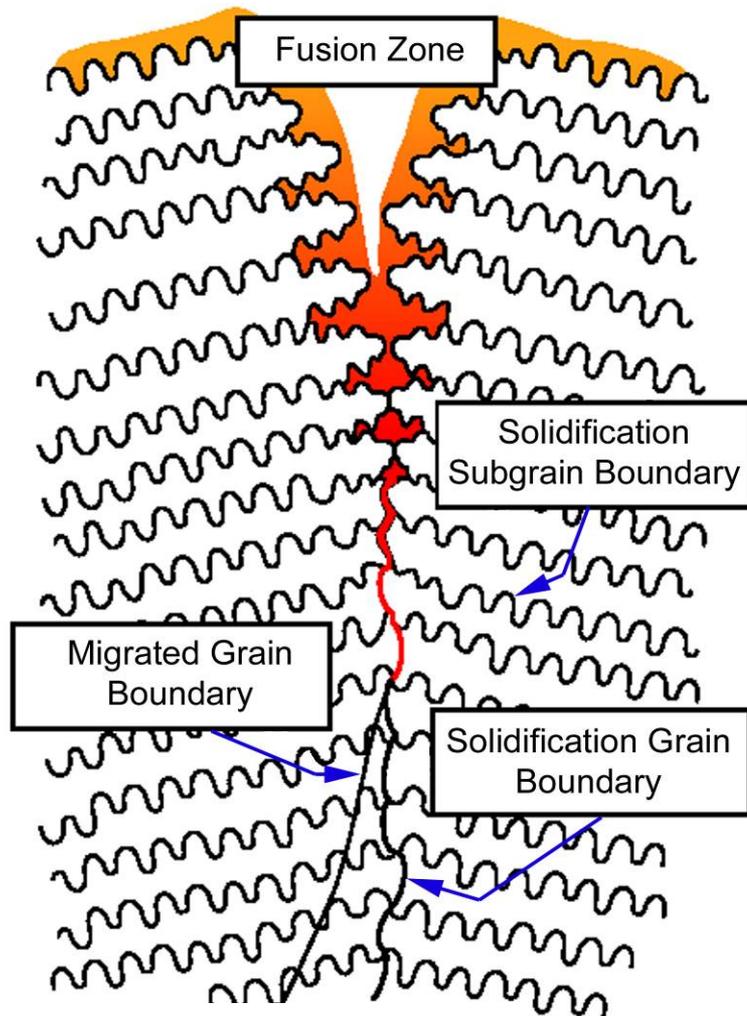
- **Partially Melted Zone (PMZ):** *base metal* that only partially melts and re-solidifies during welding where temperature is between the liquidus  $T_L$  and terminal solidus  $T_{TS}$  temperatures
- **True Heat Affected Zone (T-HAZ):** *base metal* or reheated *weld metal* where no melting occurs

- **Fusion Zone includes:**

- **Composite Zone (CZ):** mixture of *base metal* and *weld filler metal*
- **Un-mixed Zone (UMZ):** melted and re-solidified *base metal* that does not mix with the weld metal



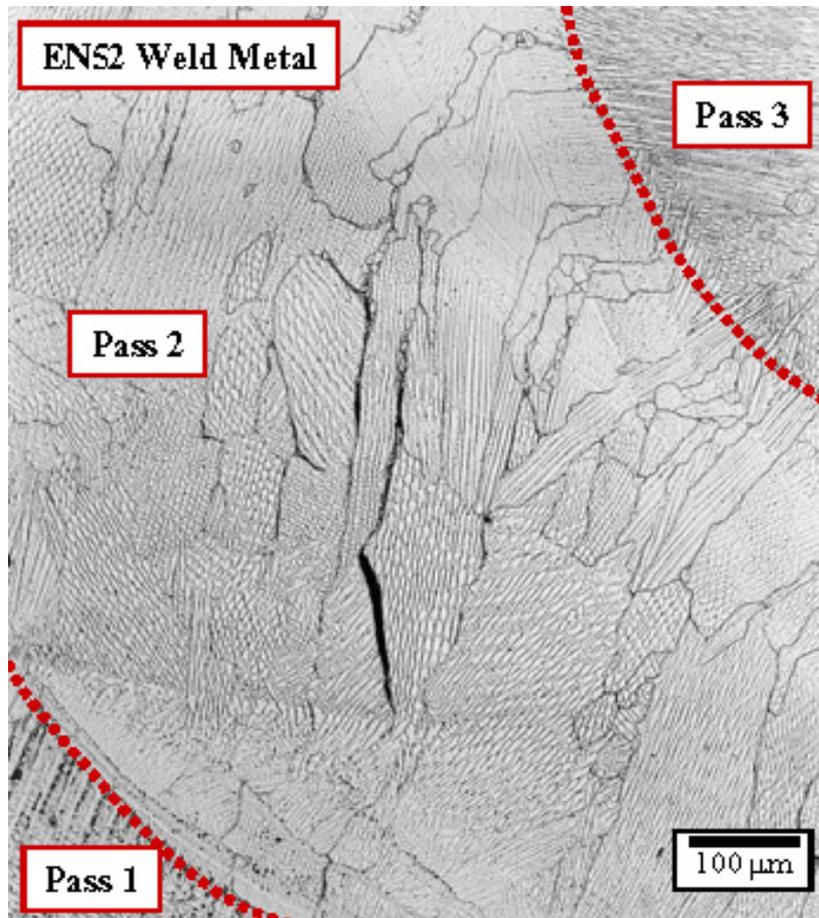
# Solidification & Ductility-dip Crack Morphology



Adapted from Lippold

- **Solidification Grain Boundary (SGB)**
  - High composition gradient
  - High angle misorientation
- **Solidification Subgrain Boundary (SSGB)**
  - High composition gradient
  - Low angle misorientation
- **Migrated Grain Boundary (MGB)**
  - Local variation in composition
  - High angle misorientation
- **Solidification cracks occur in SGBs & SSGBs**
- **Ductility-dip cracks occur along MGBs**

# Ductility-Dip Crack Location and Morphology



Courtesy Mark Cola

- Ductility-dip crack (DDC) in 2<sup>nd</sup> pass reheated in ductility-dip temperature range by 3<sup>rd</sup> weld pass
- DDC occurs along large and straight migrated grain boundaries
- Susceptible weld metals (i.e., 52 & 52M) have low impurities and few 2<sup>nd</sup> phases to pin migration (growth) of weld metal grains

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# Filler Metals in EPRI Test Matrix

- Special Metals (21% Cr)
    - **82 (ERNiCr-3)** \*\* *special heat with high hot crack resistance*
  - ThyssenKrupp (27% Cr)
    - **52i-A (ERNiCrFe-15)** \* *small experimental melt*
    - **52i-B (ERNiCrFe-15)** \*\* *large production melt*
  - Special Metals (30% Cr)
    - **52 (ERNiCrFe-7)** *not in test matrix*
    - **52M (ERNiCrFe-7A)** \*
    - **52MSS-A & B (ERNiCrFe-13)** \* *two small experimental melts*
    - **52MSS-C (ERNiCrFe-13)** \* *large production melt*
    - **52MSS-D (ERNiCrFe-13)** \*\* *large production melt*
    - **52MSS-E low Fe (ERNiCrFe-13)** \*\*\* *small experimental melt*
- \* *Testing complete*   \*\* *Testing in progress*   \*\*\* *Testing planned*

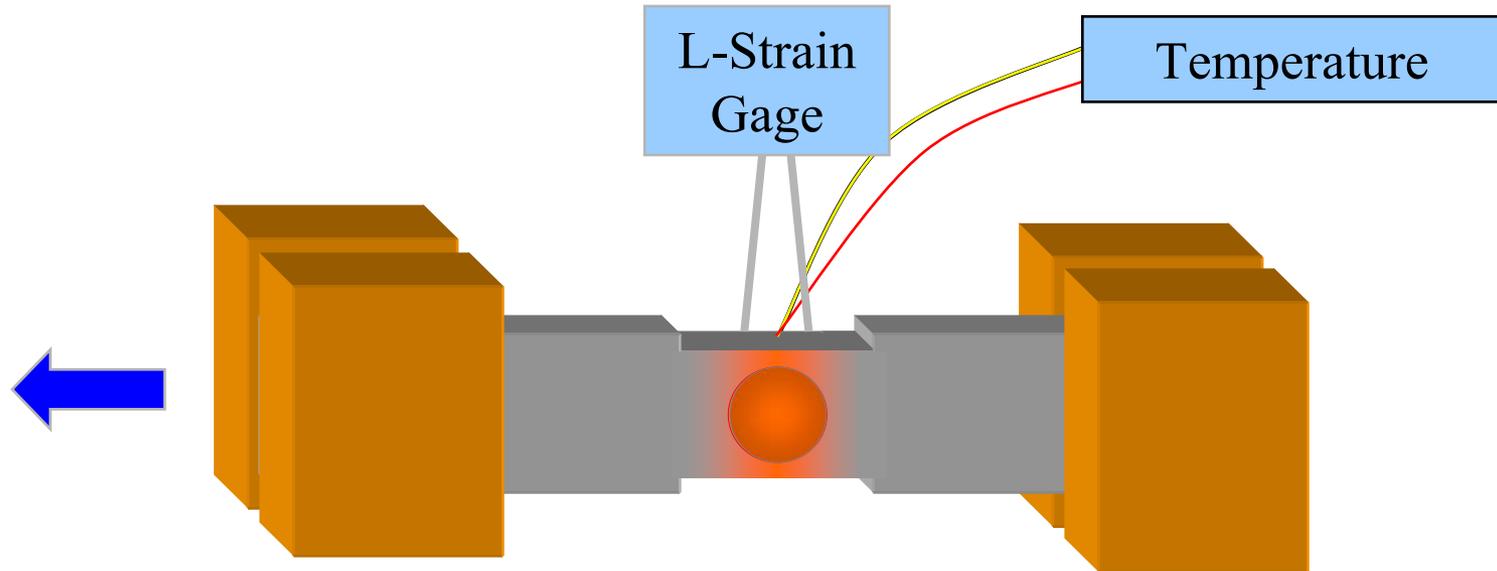
# Table of Filler Metal Compositions

	52M NX0T85TK	52MSS-A D5-8423	52MSS-B HV1224	52MSS-C NX77W3UK	52MSS-D NX79W1UK	52i-A HD52	52i-B 187775	82 6359DR	
Al	0.09	0.07	0.24	0.13	0.12	0.06	0.45	-	Al
B	0.0004	-	-	0.001	-	0.0002	<0.0010	-	B
C	0.02	0.03	0.018	0.023	0.03	0.031	0.040	0.033	C
Co	0.011	<0.001	0.003	<0.01	0.014	-	<0.02	0.03	Co
Cr	30.11	29.92	29.20	29.49	29.46	26.88	26.98	21.35	Cr
Cu	0.03	0.06	0.055	0.05	0.04	-	0.01	0.01	Cu
Fe	8.87	8.31	8.63	8.79	8.91	3.00	2.55	0.53	Fe
Mn	0.72	0.19	0.70	0.31	0.31	3.19	3.04	2.90	Mn
Mo	0.05	3.83	3.68	3.51	3.20	-	0.003	-	Mo
Nb	0.87	2.57	2.4	2.51	2.40	2.65	2.58	2.43	Nb
Ni	59.21	54.67	54.67	52.36	56.20	63.84	63.88	74.55	Ni
P	0.002	<0.001	0.016	0.004	0.005	0.003	0.002	0.003	P
S	0.0005	0.001	0.0006	<0.0005	0.00015	0.0006	0.001	0.001	S
Si	0.11	0.12	0.15	0.11	0.11	0.15	0.05	0.16	Si
Ta	<0.01	0.017	0.013	0.01	<0.01	-	0.004	<0.01	Ta
Ti	0.16	0.19	0.21	0.18	0.18	0.19	0.37	0.33	Ti
Mg	-	-	-	-	-	0.0003	0.002	-	Mg

(1) Composition from Certified Material Test Reports

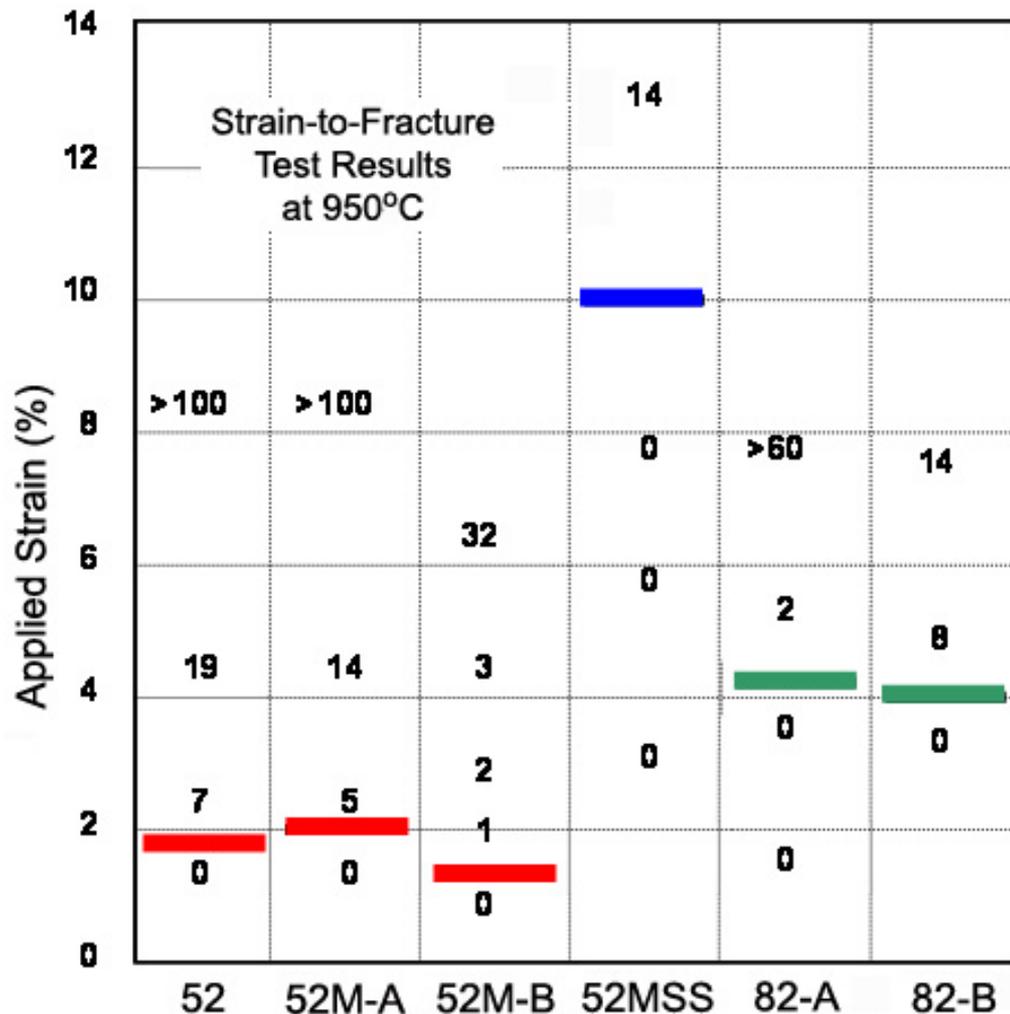
# Strain-to-Fracture (STF) Test Description

- STF test measures susceptibility to ductility-dip cracking (DDC)
- Specimens are prepared with weld metal in the gage area with a polished spot weld to provide consistent weld grain structure
- Gleeble™ tester is used to apply controlled heating and strain loading



*From John Lippold, OSU*

# Strain-to-Fracture (STF) Data

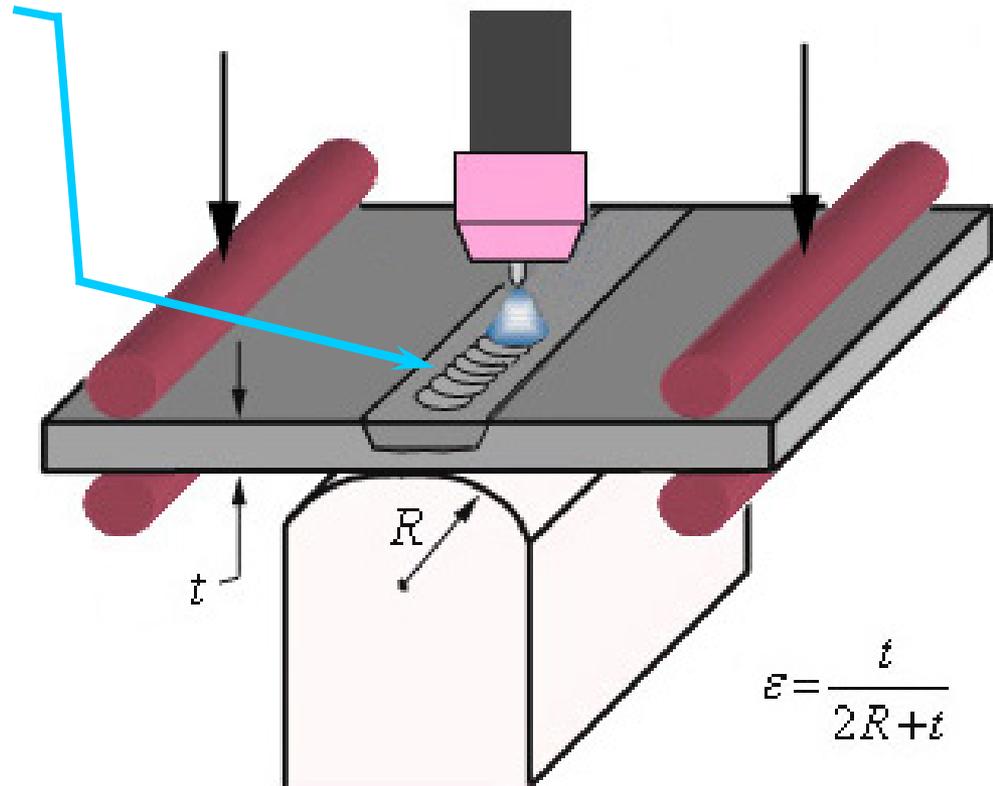


Increasing DDC Susceptibility

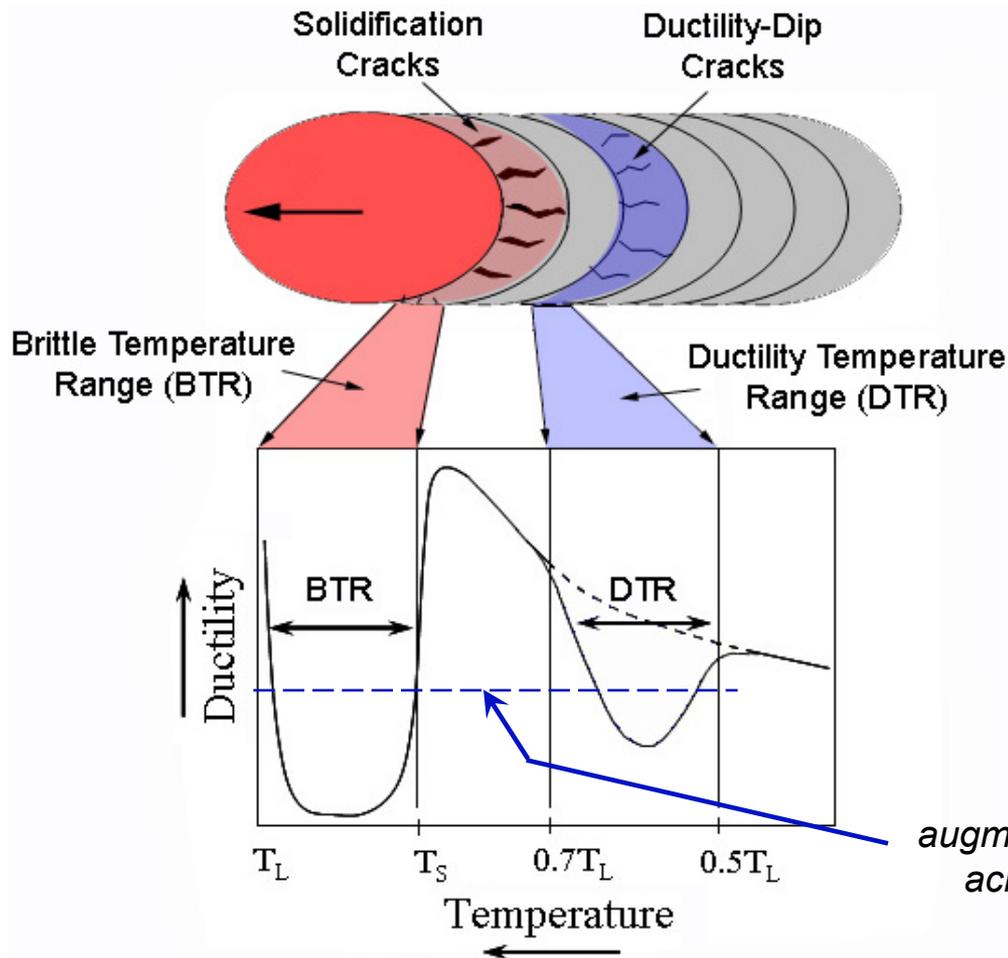
- Ductility-dip cracking (DDC) is a solid state 'reheat' type cracking mechanism
- 52 & 52M both have low resistance to DDC
- 82 is considered acceptable based on experience
- 52MSS shows superior resistance to DDC
- Recent new heat of 52MSS is off the chart with threshold between 19% and 21% applied strain
- No STF testing with 52i

# Transverse Vareststraint Test Description

- Autogenous weld bead (no filler metal) over all weld metal specimen
- Specimen is bent during welding to apply an augmented strain on the plate surface during weld solidification
- Testing is performed over range of strain values (*radius of die block determines strain*)



# Transverse Vareststraint Weld Details



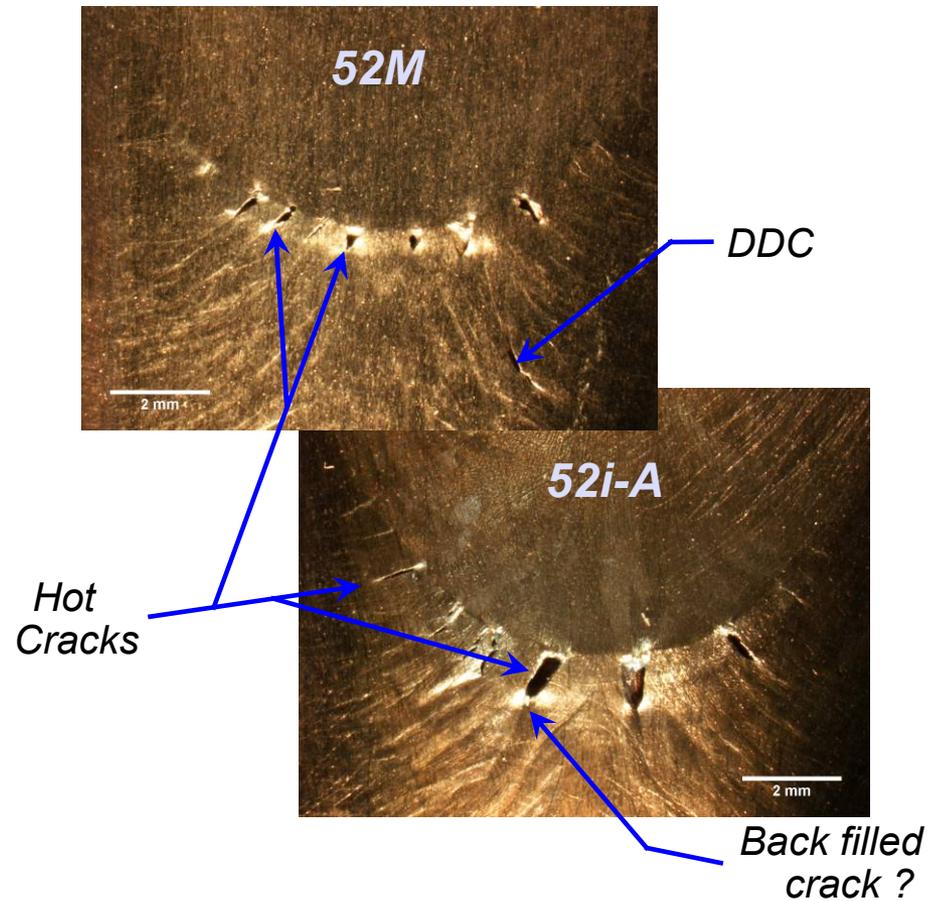
## Test Parameters

Current, A	180
Voltage, V	10
Arc Length, in	0.08
Travel Speed, ipm	5.0
Augmented Strain Range, %	0.25 – 10.0
Ram Travel Speed, in/min	6.0
Pre-Bend Weld Length, in	1.5
Total Weld Length, in	2.0

*augmented strain is constant across BTR & DTR troughs*

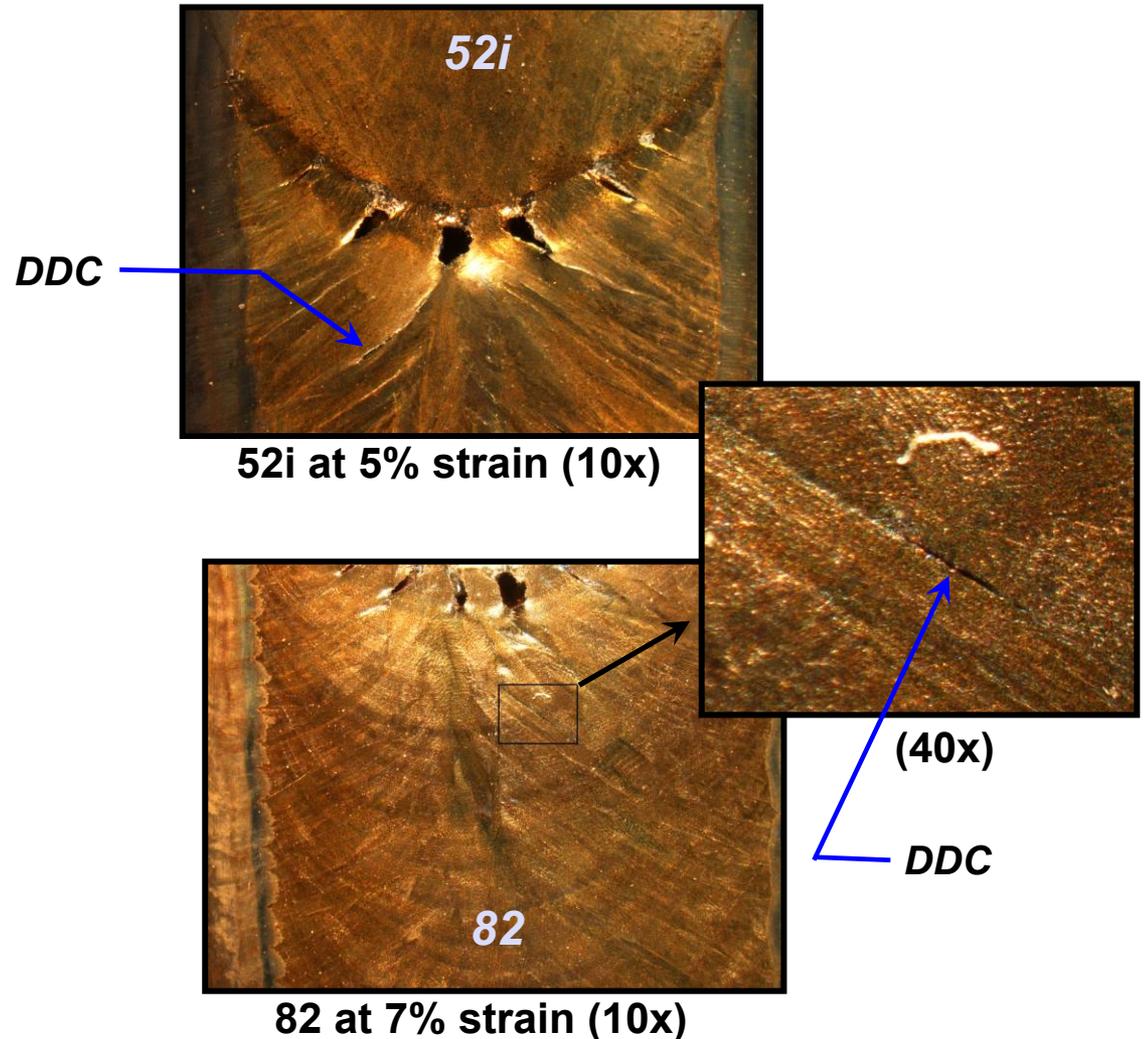
# Evaluation of Transverse Vareststraint Cracks

- Maximum crack length (MCL) is measured for each strain level tested
- Maximum crack distance (MCD) is longest crack measured at or above the saturation strain
- Above saturation strain threshold the MCL is essentially constant
- DDC can also be found by the transverse vareststraint test

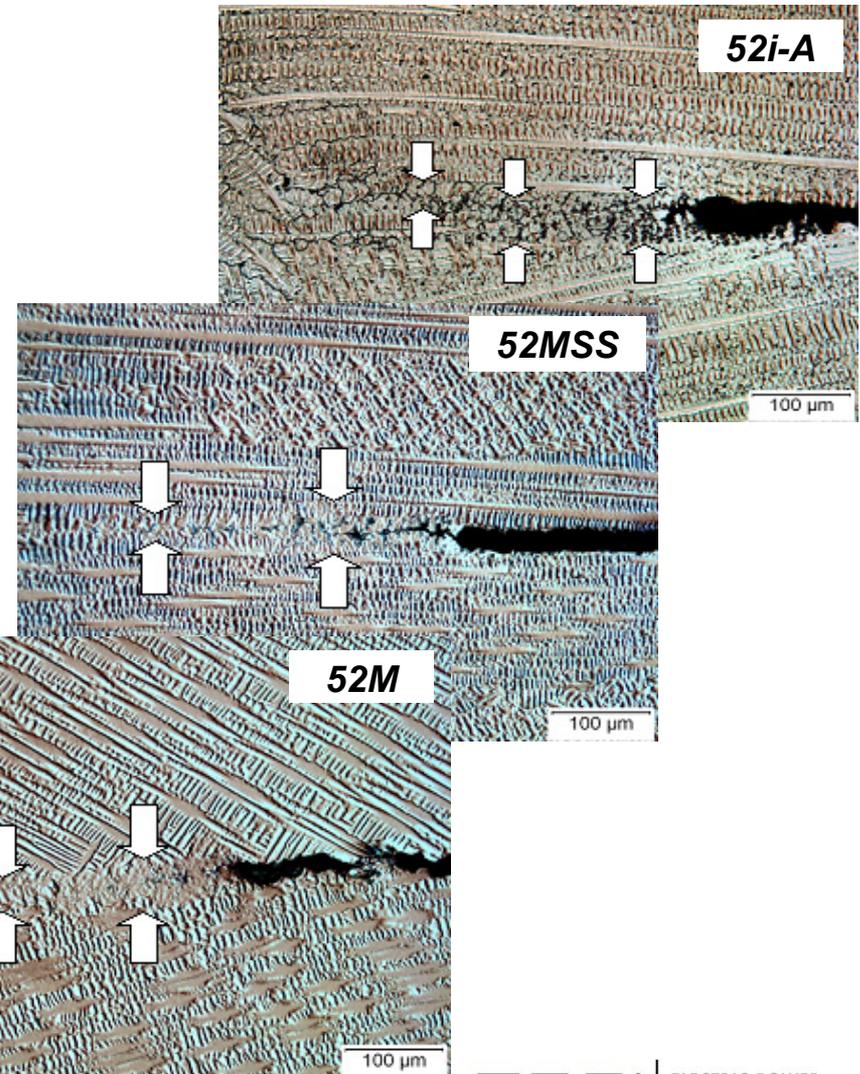
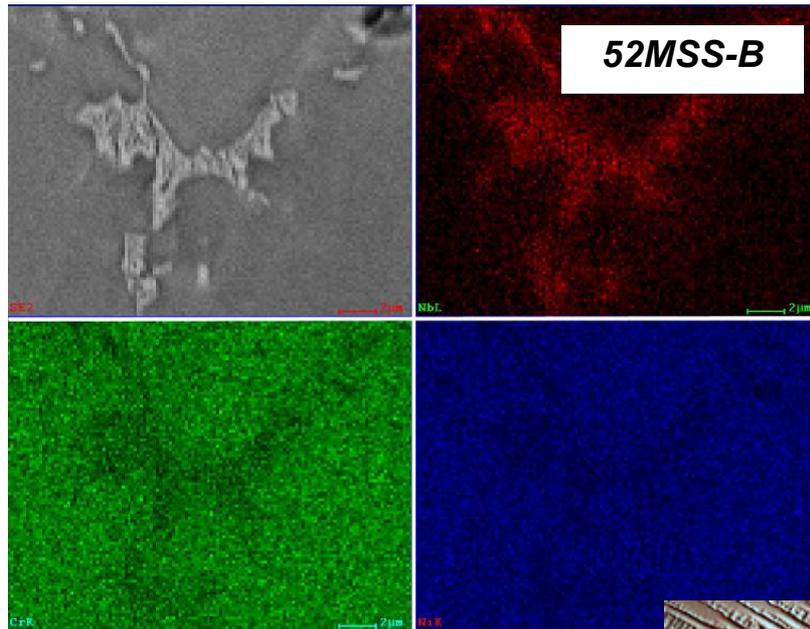


# Varestraint Test Results (DDC Resistance)

- Solidification cracking occurs between  $T_{liq}$  and  $T_{sol}$
- DDC occurs between  $\sim 0.75T_{liq}$  and  $0.5T_{liq}$
- DDC observed in 52M at 5% strain
- DDC observed in 82 at 7% strain
- DDC observed in 52i at 5% strain



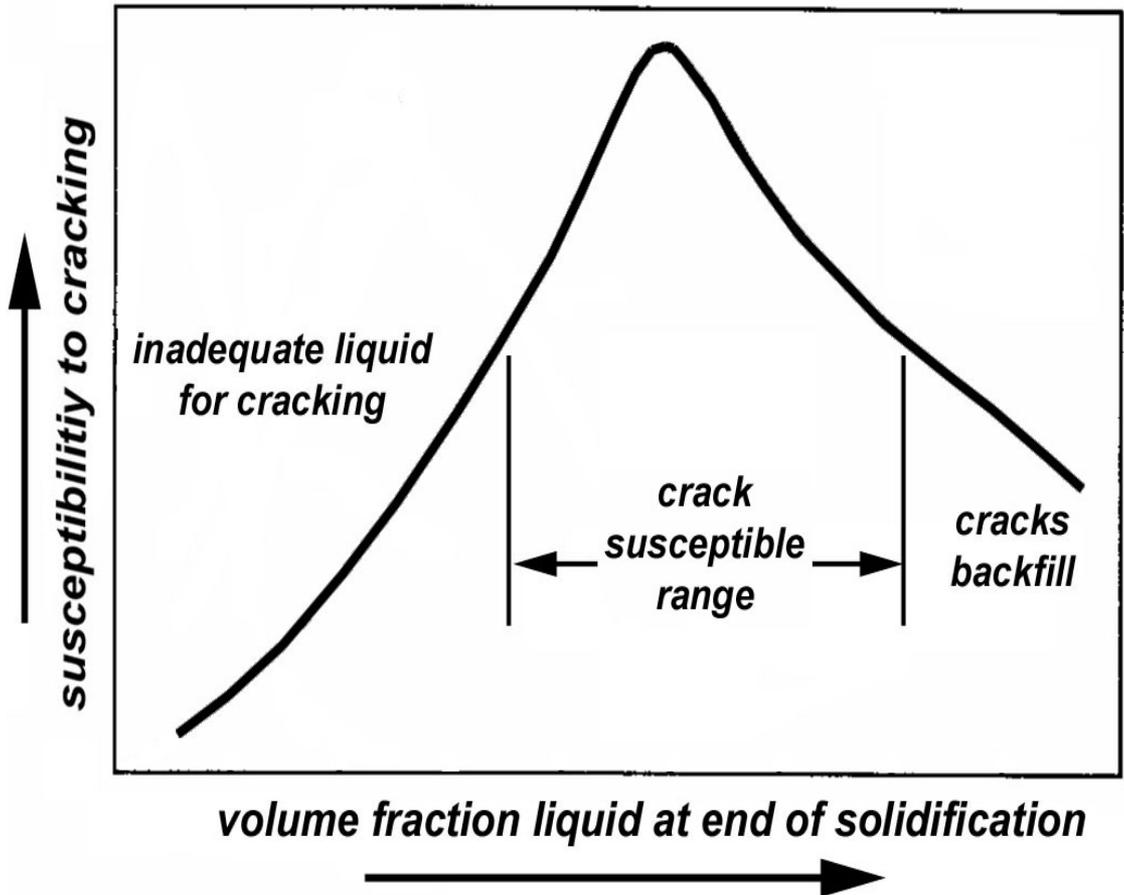
# Backfilling in Transverse Vareststraint Cracks



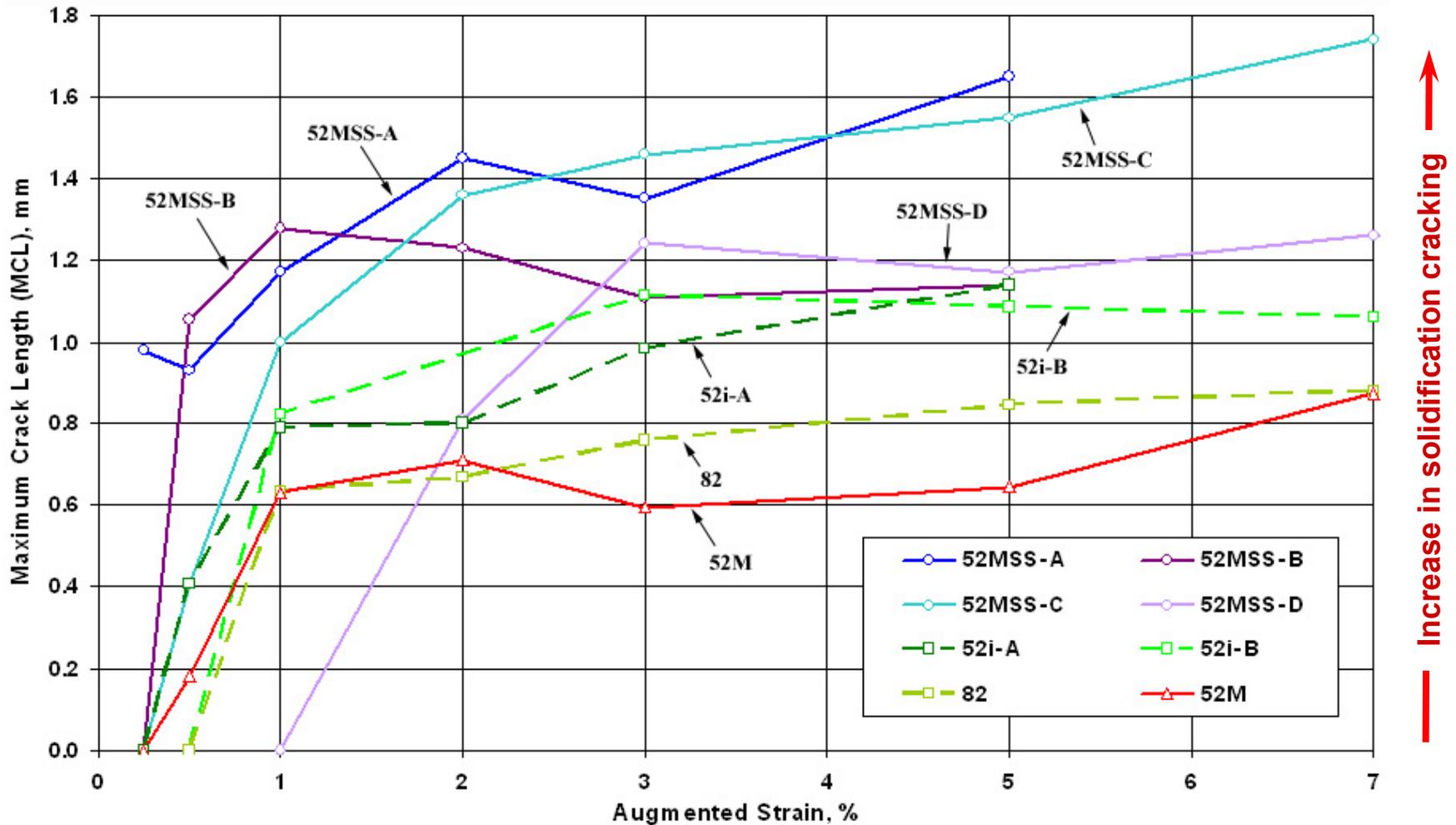
- Evidence of backfilling (crack healing) in 52i-A, 52MSS, and 52M
- EDS analyses shows NbC eutectic and Laves in solidification boundaries

# Crack Healing by Backfill Mechanism

- Wide solidification temperature range indicates higher volume fraction liquid at end of solidification
- 52i has widest solidification range but is less susceptible to cracking
- Reduced susceptibility to cracking is likely due to adequate volume fraction of liquid to cause crack 'healing' (back filling)



# Maximum Crack Length vs Augmented Strain



# Cast Pin Tear Test (CPTT) Description

- CPTT evaluates solidification crack susceptibility
- Alloy charge is cast into a 3/8" diameter mold
- Charge may be adjusted for weld metal dilution
- Longitudinal tensile strain occurs in pin as it solidifies and cools
- Strain increases as pin length increases



*OSU cast pin tear test apparatus*



*Set of buttons is prepared for each heat*

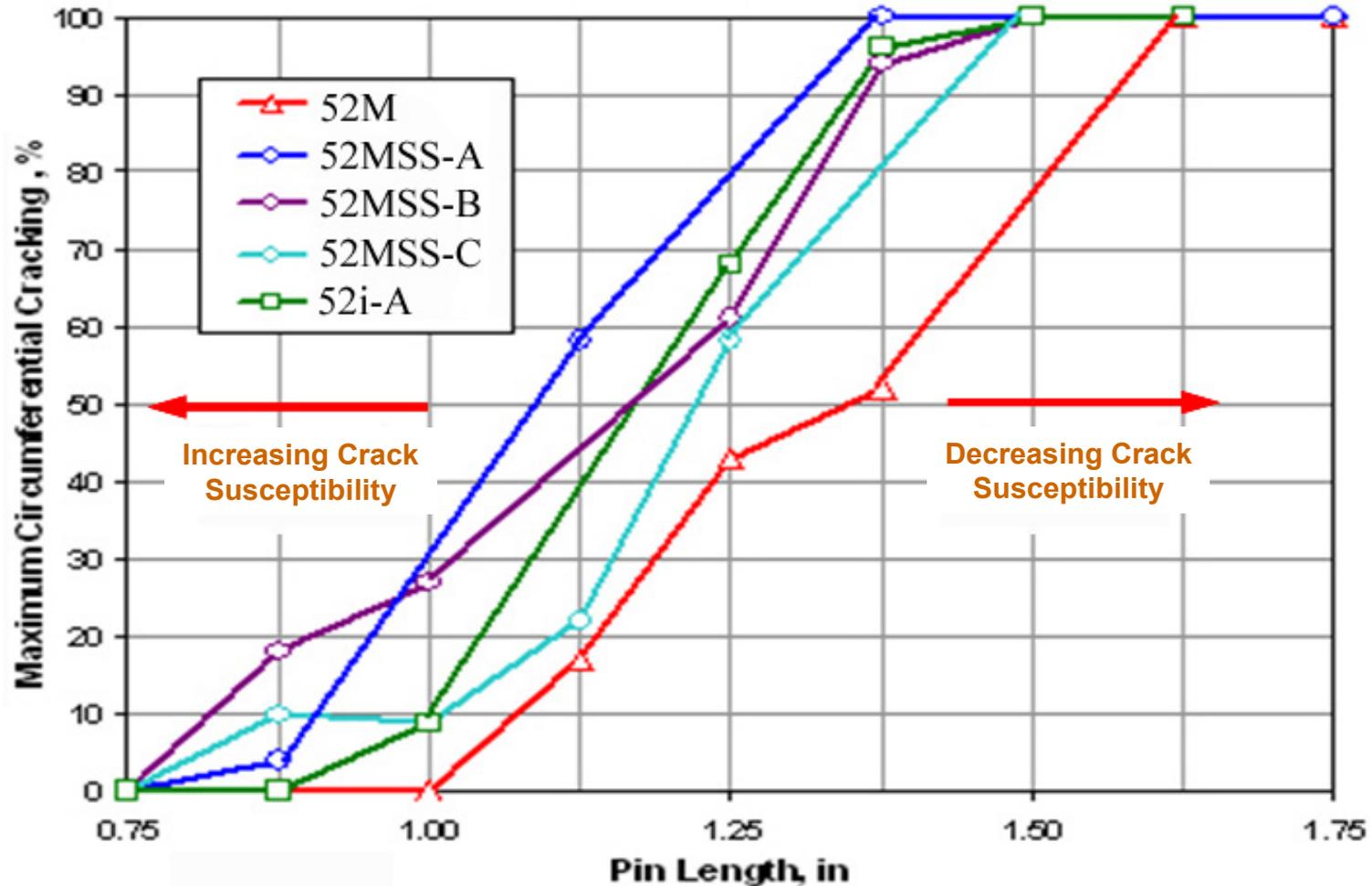


*Button is melted by electric arc and cast into pin mold*



*3/8" diameter pins are cast from 3/8" to 2-1/8" gauge length in 1/8" increments. Head and foot of pin restrain gauge length during cooling*

# Max Circumferential Cracking (MCC) vs Pin Length



# Comparison of Filler Metal Cracking Susceptibility

- Cast pin tear test results (solidification cracking at 40% MCC)

52MSS-A > 52MSS-B > 52i-A > 52MSS-C > 52M

- Transverse vareststraint results (solidification cracking in 2% - 5% strain range)

52MSS-A > 52MSS-C > 52MSS-B > 52MSS-D > 52i-B > 52i-A > 82 > 52M

← Increase in Solidification Cracking →

- Transverse vareststraint results (DDC)

52i-A > 52M > 82 > 52MSS (no DCC observed in 52MSS)

- Strain-to-fracture test results (DDC)

52 = 52M > 82 > 52MSS (no STF test data for 52i)

← Increase in Ductility-dip Cracking (DDC) →

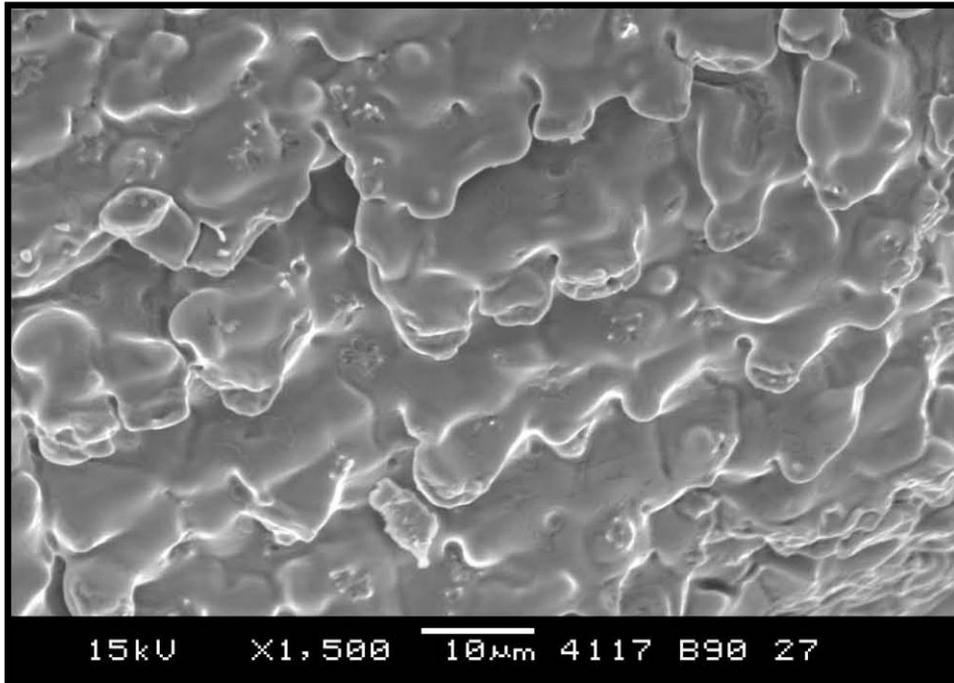
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# Dilution Testing Objectives

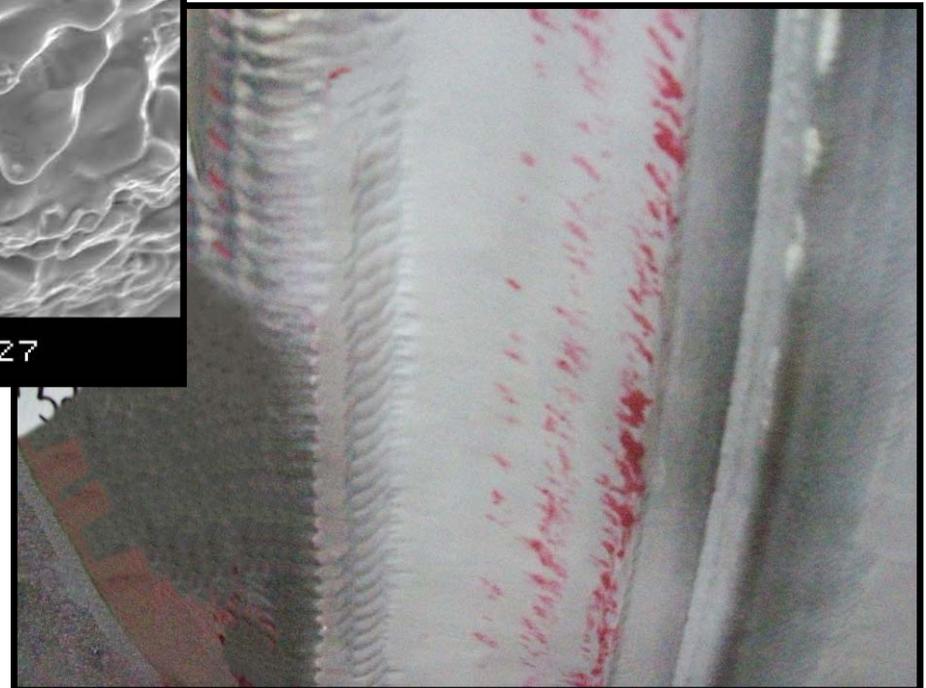
- Investigate and quantify influence of dissimilar metal dilution on heats and classifications of 52M
  - Determine level of dilution with stainless steel that causes solidification cracking
    - Testing to date shows 52M diluted with ~35% Fe increases susceptibility to hot cracking*
  - Establish S & P and Si threshold(s) that promote solidification cracking
  - Investigate influence of S + Si on dilution and potential for increasing risk for solidification cracking
  - Optimize Cast Pin Tear apparatus to improve resolution and sensitivity for dilution effects on solidification cracking

# Result of Excessive Dilution by Cast Stainless Steel



*Test Mockup (below) - 52M pad  
on ER308L buffer layer*

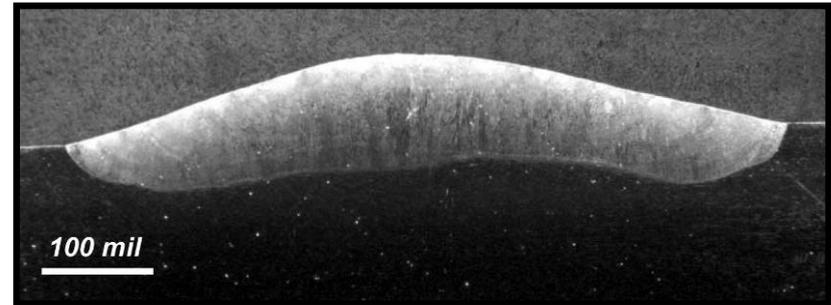
*Base metal is SA-351 CF8A  
0.019% S, 0.032% P, 0.72% Si*



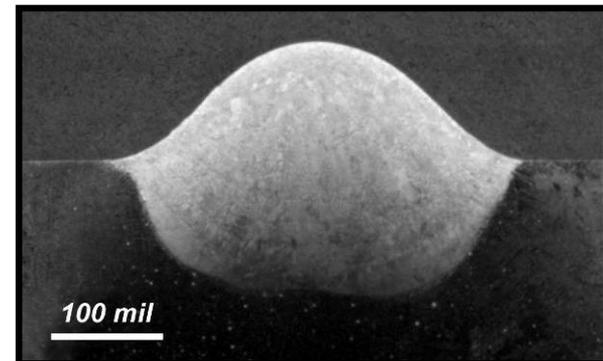
- *SEM of hot crack (above) in boat sample removed from 52M overlay*
- *52M layer (right) shows multiple liquid penetrant crack indications*

# Influence of CASS Piping on 52M Weld Bead Shape

- Influence of CASS trace elements on weldability:
  - Weld bead shape and penetration
  - Susceptibility to hot cracking
- Industry needs to:
  - Understand how CASS influences 52M welding
  - Identify deleterious trace elements in CASS
  - Define threshold values to protect against hot cracking



*Weld Shape on 304L Plate*

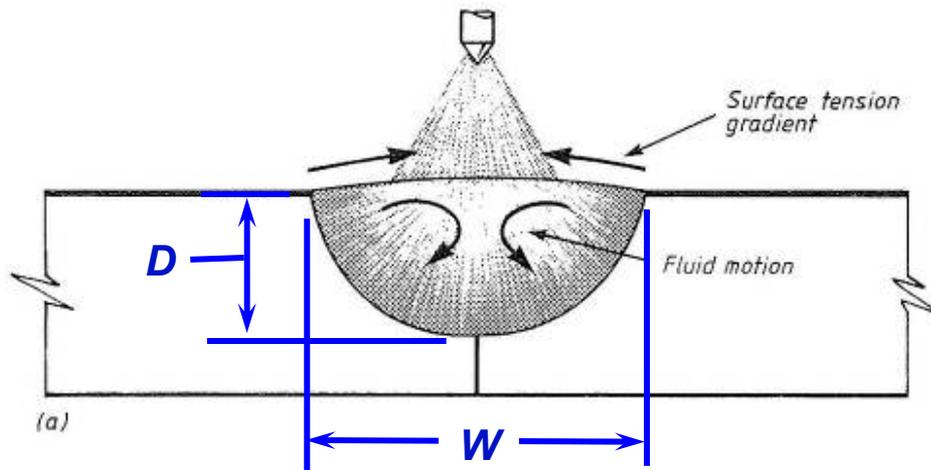
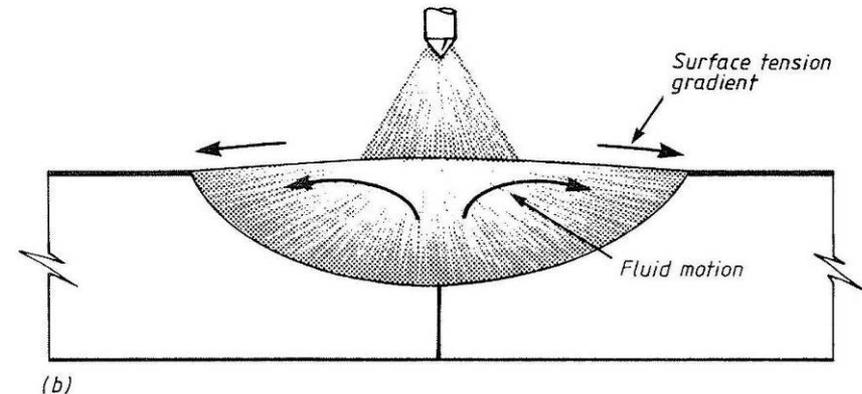


*Weld Shape on CF8A (CASS) Pipe*

# Effect of Sulfur and Silicon on Austenitic Welds

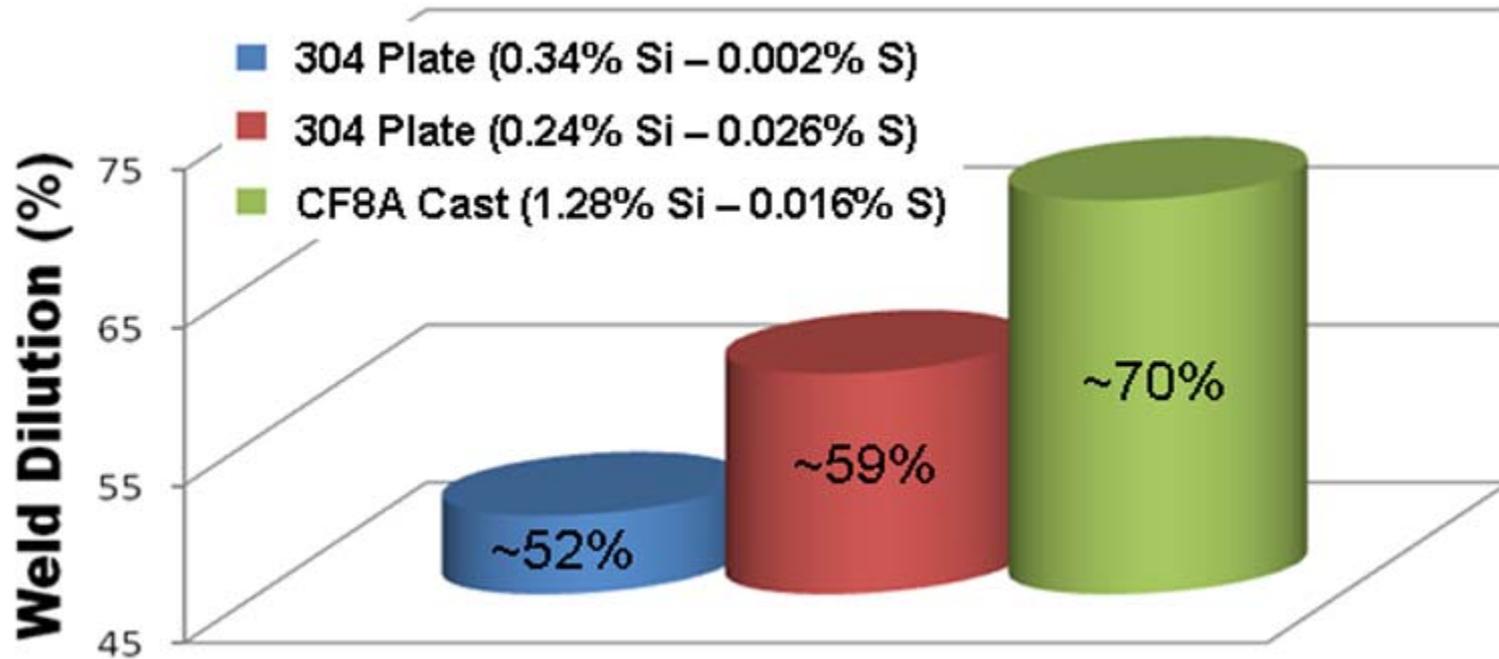
- Sulfur influences the weld pool surface tension gradient in austenitic welds
- Surface tension gradient drives molten metal flow (Marangoni flow)
- Silicon addition decreases viscosity which enhances flow

From Tinkler – London Conf Nov 1983



- Shallow & wide bead < 0.008% sulfur
- Deep & narrow bead > 0.015% sulfur
- High sulfur is known to cause hot cracking

# Influence of S & Si Composition on Dilution



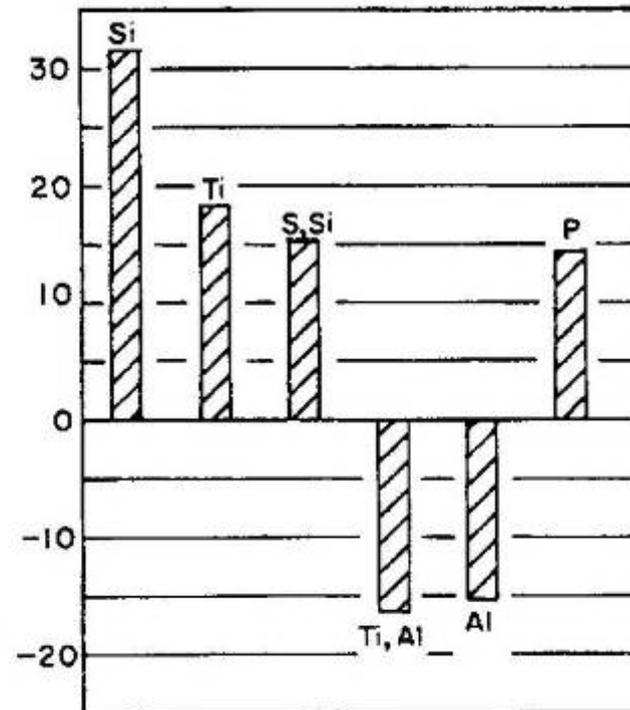
Single bead dilution ranged from 52% to 70% for low to high S & Si

# Minor Element Influence on Inconel 600

- Si, S-Si, and P increased weld cross-sectional area
  - Area increase corresponds to increase in dilution
- Si decreases viscosity and S increases penetration

## Cross-Sectional Area

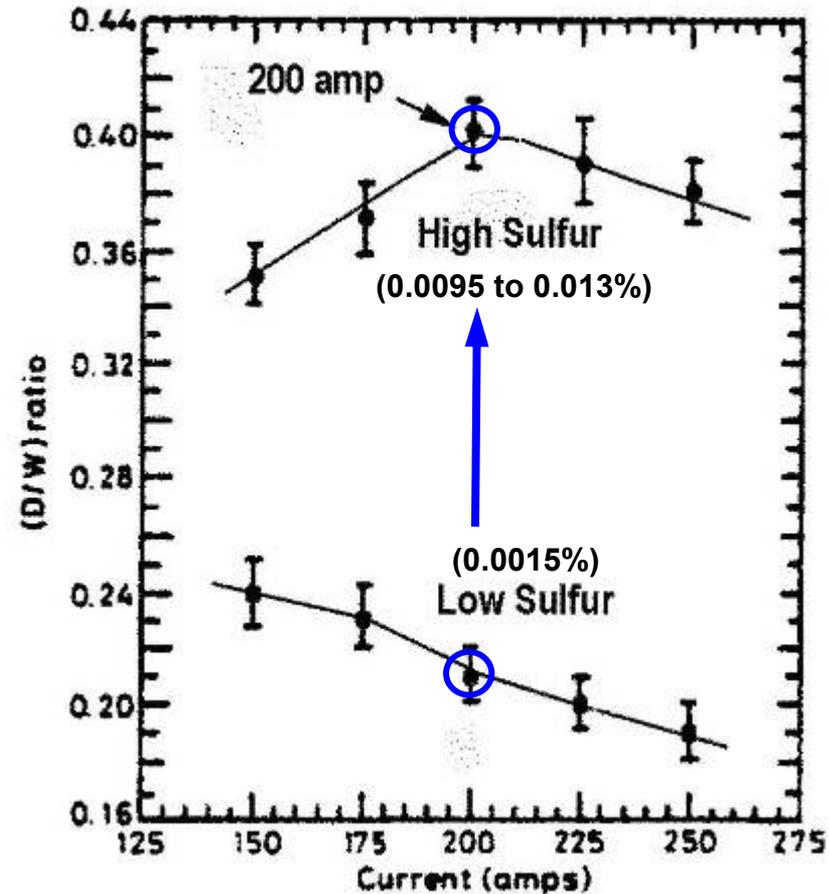
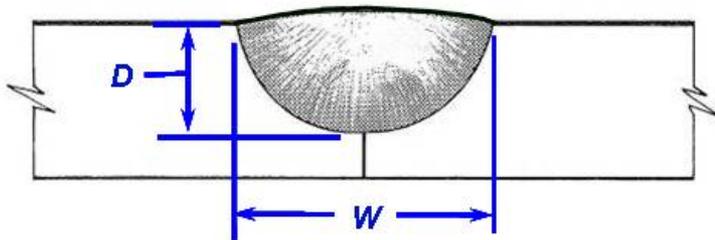
Treatment combination	Effect	Best value
Si	31.3	17.3
Ti	18.1	37.9
S-Si	17.7	35.0
Ti-Al	-16.2	6.4
Al	-15.4	4.4
P	14.1	17.7
Mean	-	27



*From Savage – WJ April 1977*

# Influence of Current and Sulfur on D/W

- Influence of current on austenitic stainless steels (304L & 316L) with
  - Low 0.0015% S
  - High 0.0095 to 0.013% S
- Low S – D/W ratio decreases as current increases
- High S – D/W ratio is higher and peaks at 200 amps

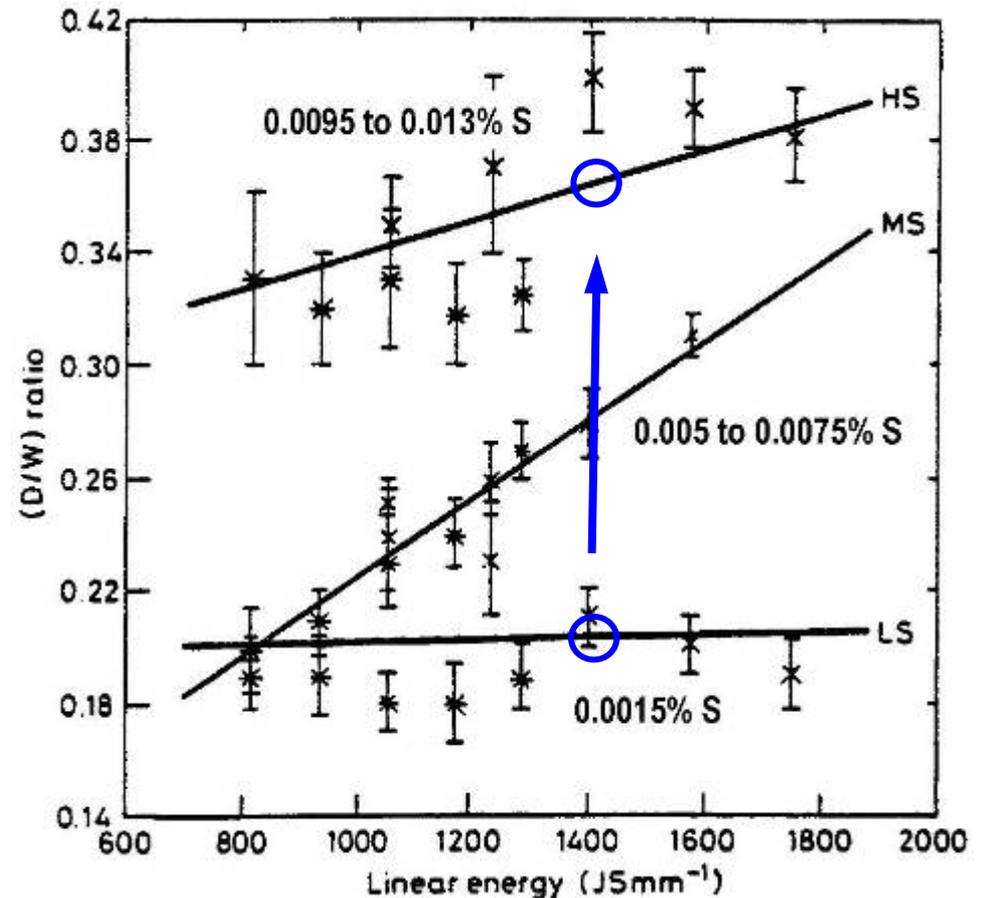


From Shirali – WJ July 1993

# Influence of Sulfur and Heat Input on D/W

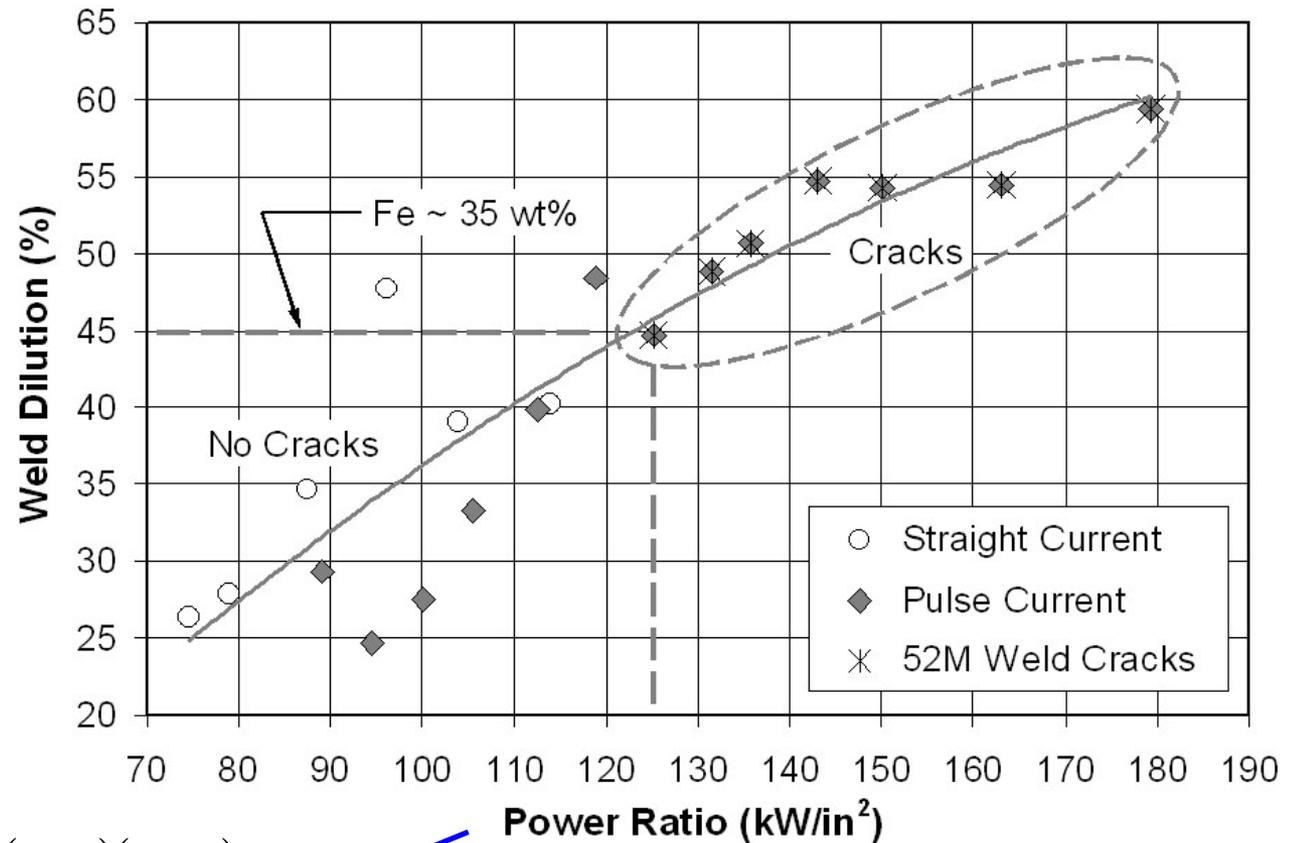
- Influence of heat input on austenitic stainless steels (304L & 316L) with low – med – high sulfur content
- Depth-to-width (D/W) ratio varies with heat input depending on sulfur content
- Heat input has little influence on D/W on low sulfur ( $\leq 0.0015\%$ )
- Heat input has strong influence on D/W on med sulfur (0.005 to 0.0075%) and high sulfur (0.0095 to 0.013%)

From Shirali – WJ July 1993



# Solidification Cracking by Dilution with Stainless Steel

- 52M solidification cracking on 304L plate (not high S material)
- Solidification cracking occurs at high dilution
- Studies show 52M diluted with > 35% Fe is susceptible to solidification cracking

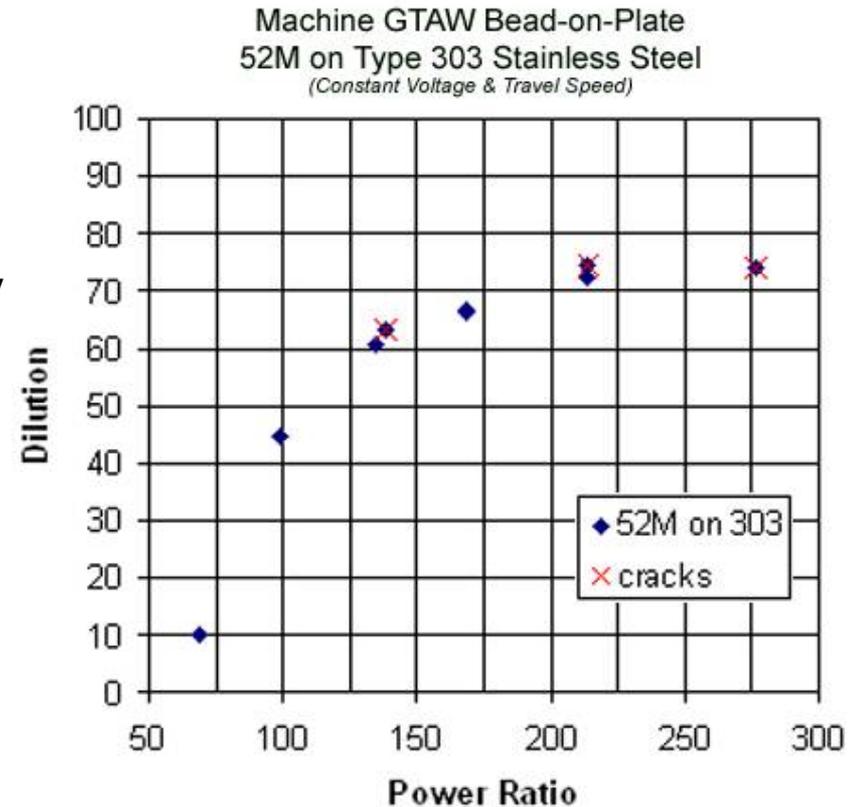
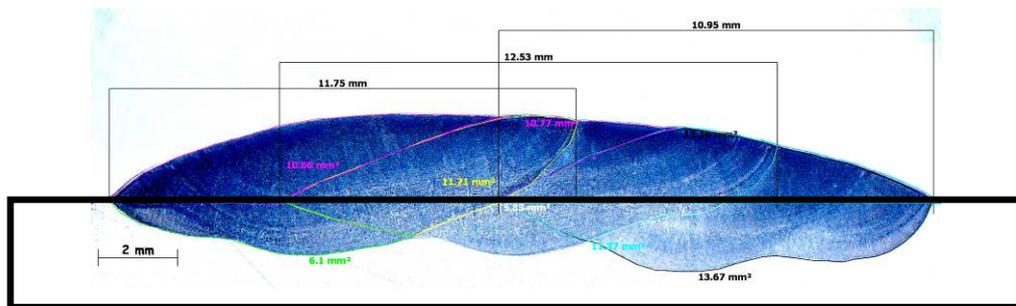


$$\frac{Power}{Weld\ Deposit\ Area} = \frac{(voltage)(amp)}{\left(\frac{WFS}{TS}\right)(A_{wire})}$$



# Bead on Plate – 52M on Type 303 Plate

- 52M bead on Type 303 plate testing
  - Measure dilution by cross section
  - Calculate composition from dilution
- Hot cracking occurred at 60% dilution
  - Expected extensive hot cracking – only found small cracks in cross section
  - Type 303 plate has 0.21% S
  - Hot cracking occurred in 60% to 80% dilution range
    - S - 0.132% to 0.156%
    - Si - 0.35% to 0.39%
    - Mn - 1.35% to 1.45%



*Why wasn't cracking more severe?  
Crack healing by backfilling?  
Or due to high Mn?*

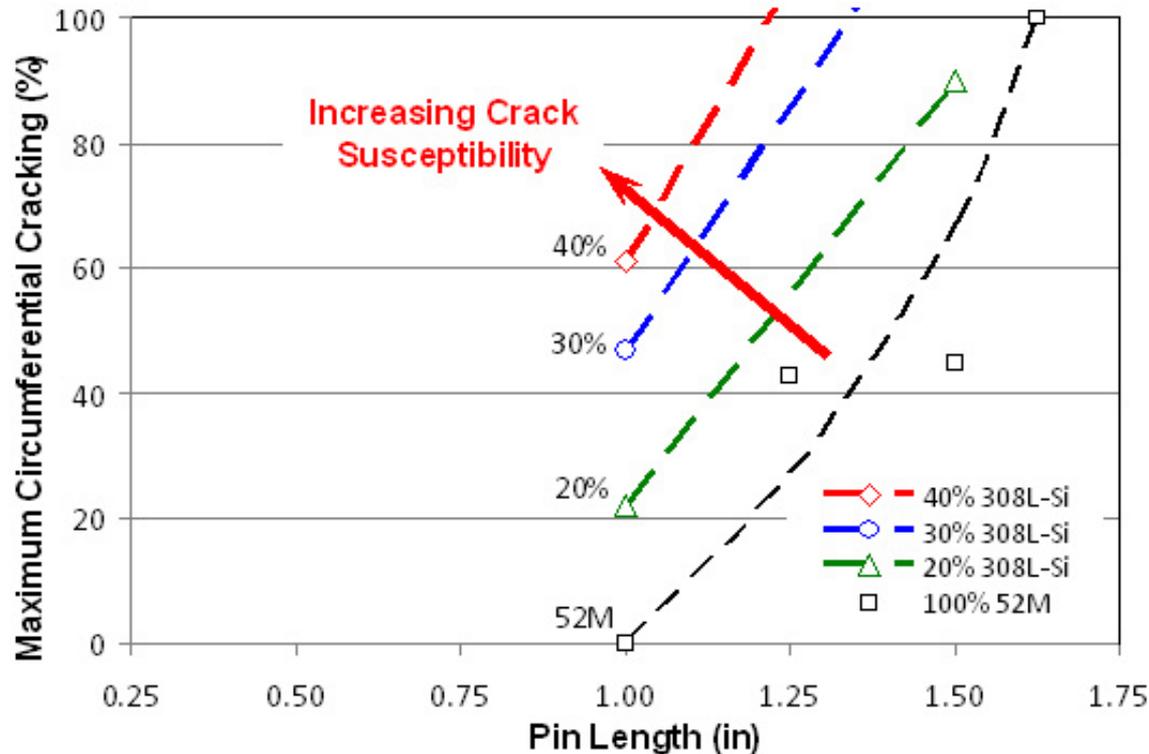
# Bead on Plate – 52M on 303 Plate Clad with ER308L-Si

- Added ER308L-Si cladding to Type 303 plate to increase Si level
  - 52M hot cracking occurred in all but extremely low dilution levels (~10% dilution)
  - Demonstrates synergy of S + Si and potential benefit of Mn in 52M filler wire
- Work in progress and all results are not yet analyzed



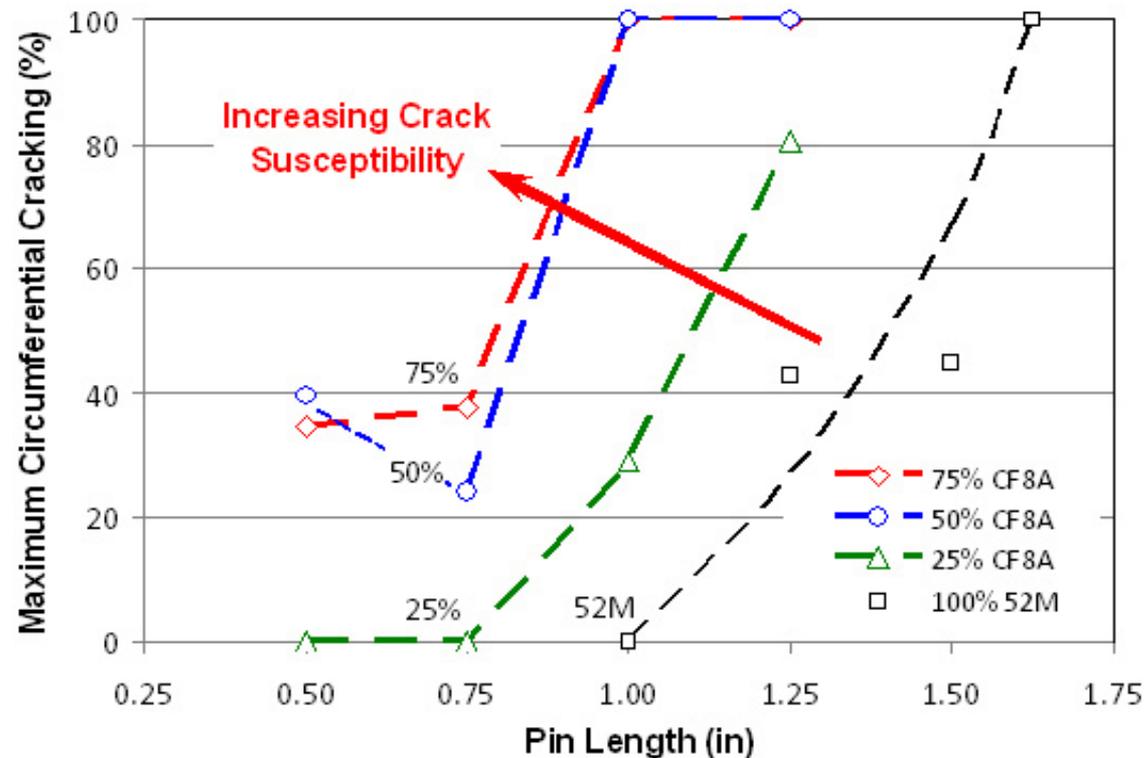
# 52M Dilution by 308L-Si (Preliminary CPTT Data)

	Cr	Ni	Fe	Mn	Si	Nb	Al	Ti	Mo	C	S	P
52M	30.1	59.2	8.9	0.72	0.11	0.87	0.09	0.16	0.05	0.020	0.0005	0.002
308L-Si	20.0	10.1	66.8	1.89	0.82	---	---	---	0.13	0.023	0.0118	0.0274
20%	28.1	49.4	20.5	0.95	0.25	0.70	0.07	0.13	0.07	0.021	0.003	0.007
30%	27.1	44.5	26.3	1.07	0.32	0.61	0.06	0.11	0.07	0.021	0.004	0.010
40%	26.1	39.6	32.1	1.19	0.39	0.52	0.05	0.10	0.08	0.021	0.005	0.012

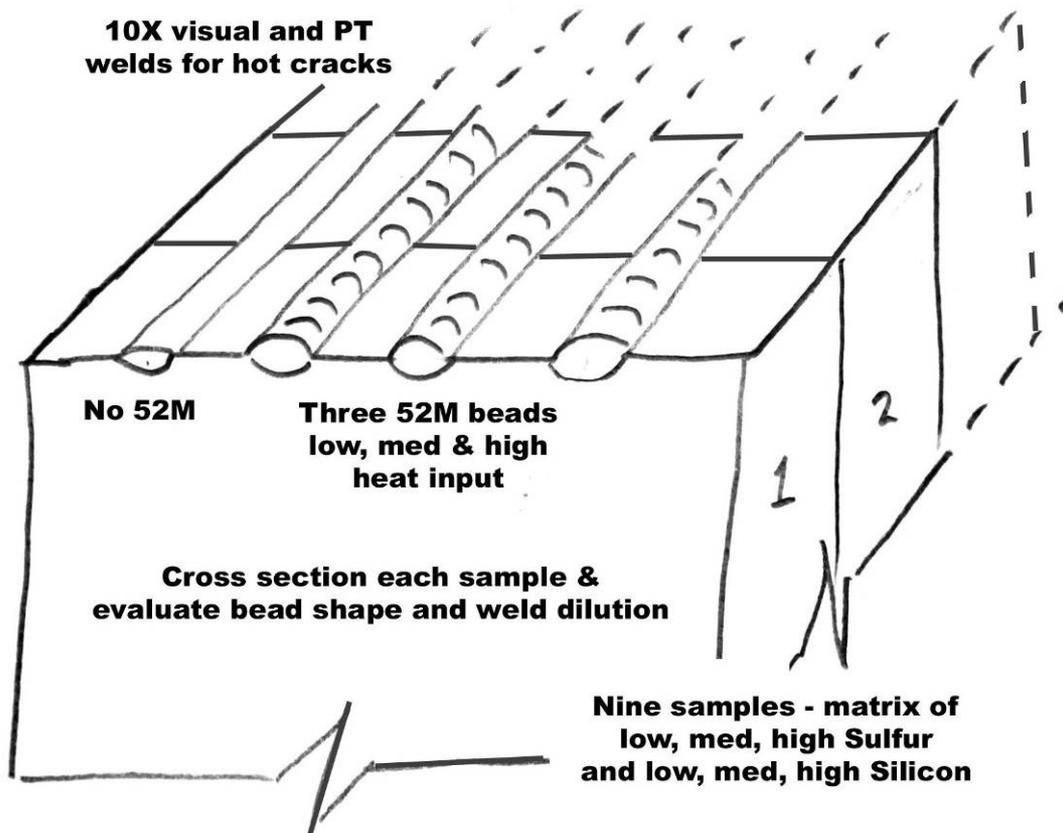


# 52M Dilution by CF8A (Preliminary CPTT Data)

	Cr	Ni	Fe	Mn	Si	Nb	Al	Ti	Mo	C	S	P
52M	29.75	58.93	8.75	0.74	0.11	0.93	0.13	0.19	0.08	0.020	< 0.001	< 0.01
CF8A	20.9	8.4	70.6	0.59	0.92	---	---	---	0.05	0.04	0.015	0.020
25%	27.5	46.3	24.2	0.70	0.31	0.70	0.10	0.14	0.07	0.025	0.005	0.0125
50%	25.3	33.7	39.7	0.67	0.52	0.47	0.07	0.10	0.07	0.03	0.008	0.015
75%	23.1	21.0	55.1	0.63	0.72	0.23	0.03	0.05	0.06	0.035	0.012	0.0175

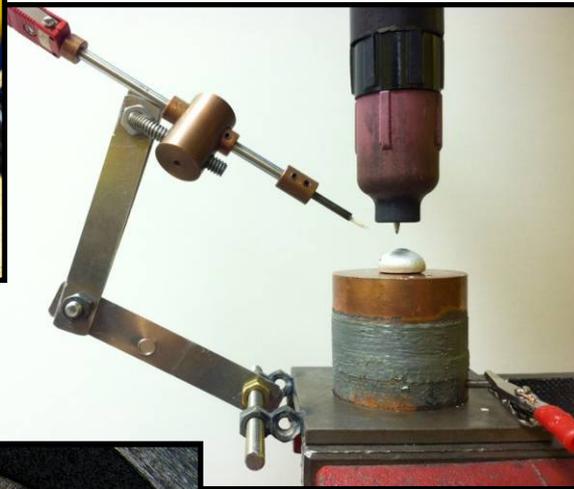


# Bead-on-Plate Tests on Controlled Composition Plates



- Testing on 9 controlled composition CASS samples (based on domestic PWR survey)
  - Sample matrix:
    - Sulfur Low - Med - High  
0.001 - 0.020 - 0.040
    - Silicon Low - Med - High  
0.05 - 0.90 - 1.80
- Determine hot cracking thresholds for different heats and specifications of high Cr nickel-base filler metals
- Evaluate synergy of S & Si on weld bead shape and dilution

# Button Melting Testing – Effects of Dilution on 52M



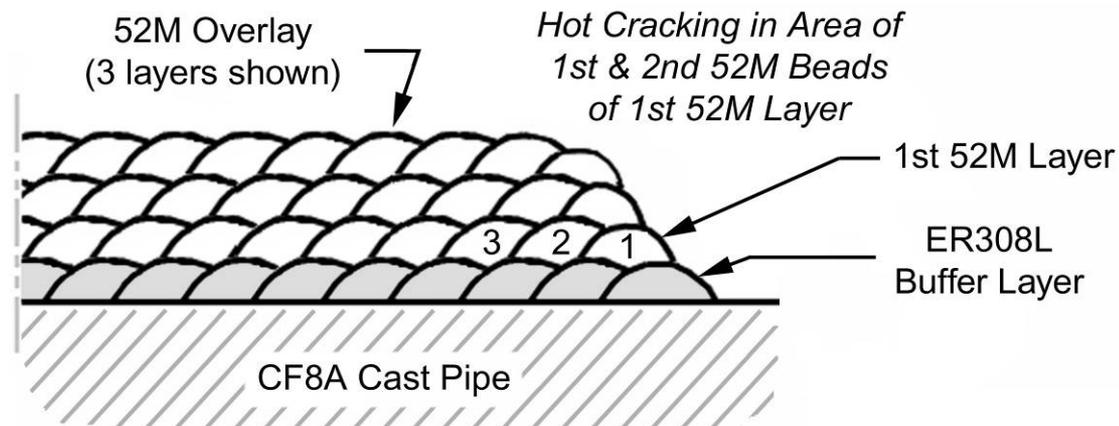
- Controlled 52M dilution compositions made by casting buttons
- Buttons are partially re-melted by GTAW
- Cooling curve is measured by plunging a Type-C thermal couple into the weld metal
- Solidification temperature range and eutectic start are measured by SS DTA technique
- Solidification grain boundaries are evaluated by SEM to determine low melting point constituents that coat the solidification grain boundary and cause hot cracking

# 52M Hot Cracking Controls

- Reduce susceptibility to hot cracking by controlling dilution:
  - Decrease power ratio (heat input)
    - Lowers S & P in diluted 52M weld metal
    - Lowers Fe in diluted 52M weld metal
  - Optimize and manage weld process parameters
  - Optimize and manage bead placement
  - Install hot crack resistant buffer layer
    - ER308L, ER309L, or crack resistant Alloy 82 buffer (barrier) layer
    - Improved buffer layer option may be low Fe 52MSS (ERNiCrFe-13)
      - High 30% Cr for PWSCC resistance
      - 52MSS is essentially immune to DDC

# Dilution Control by Buffer Layer

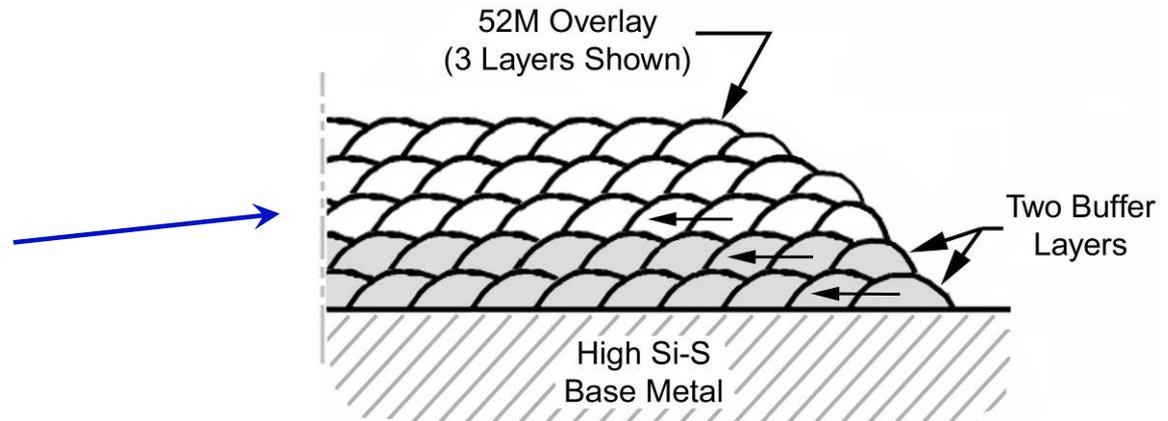
- Dilution of 52M with deleterious base metal is typical cause of hot cracking
  - ER308L layer installed to 'buffer' 52M from base metal (lowers dilution)
- Hot cracking may still occur in 1<sup>st</sup> 52M beads over buffer layer



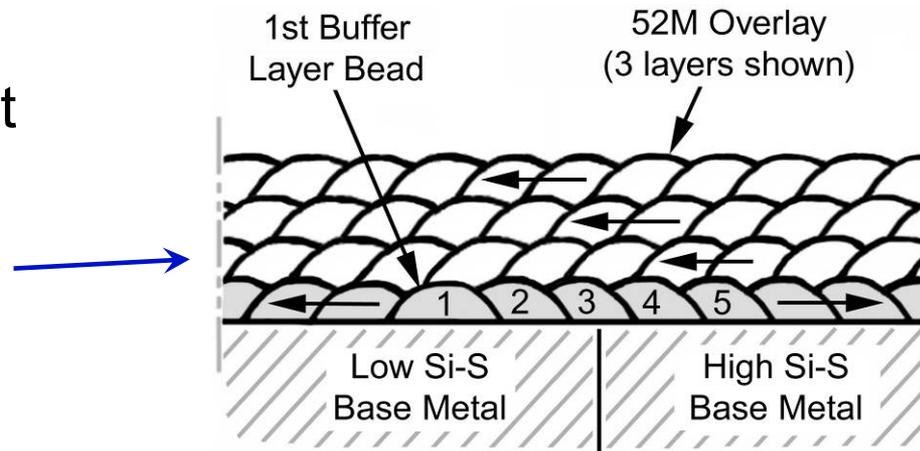
# Dilution Control by Bead Placement

- Other options:

- Two buffer layers (*doesn't solve high Fe problem*)



- Bead placement to minimize dilution with deleterious base material



# Successful 52M WOLs by Careful Control of Dilution

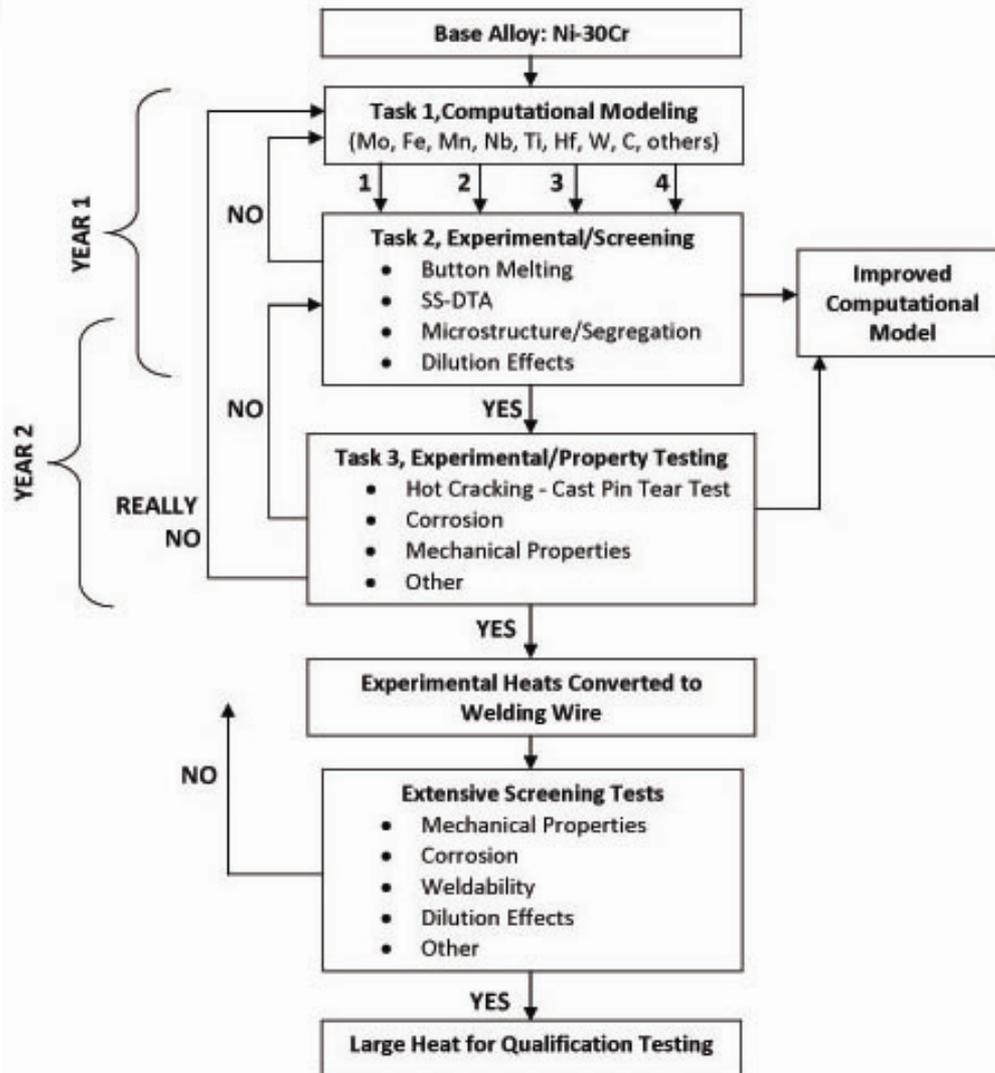
- Successful OWOL application on 4 RCP discharge nozzles
- Successful SWOL application on 4 RCP suction nozzles plus 5 other nozzles
- Rework was required to achieve acceptable quality on some WOLs
- Significant mock up testing was done to define parameters and techniques needed for successful welding



# Presentation Roadmap

- Alloy 52 / 52M Challenges
- Cracking Mechanisms
- **Weldability Testing**
  - Description of Testing and Ranking of Filler Metals
  - Dilution Issues and Testing
  - High Chromium Nickel-Base Alloy Development
- Magnetic Stir Welding
- **Welding Irradiated Material**
  - Problem and Issue
  - Laser Welding
  - Friction Stir Welding

# Development of a New High Cr Filler Metal



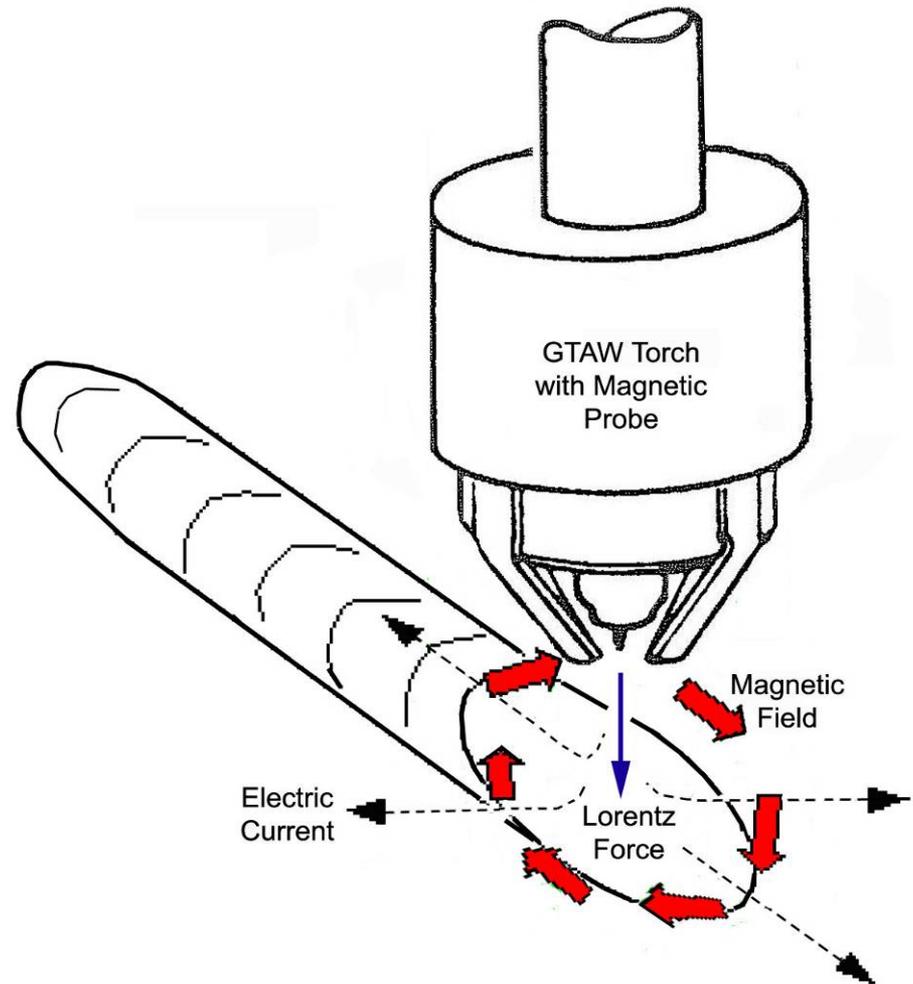
- EPRI project to develop a new filler metal was kicked off in fall of 2010
- Base composition is 30% Cr nickel-base
- Initial computational modeling at OSU to study solidification behavior and 2<sup>nd</sup> phases at the end of solidification is nearly complete
- Initial button melting experiments at OSU are in process
- New CPTT with induction melting capability and optimized mold design is nearly complete

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  - Problem and Issue
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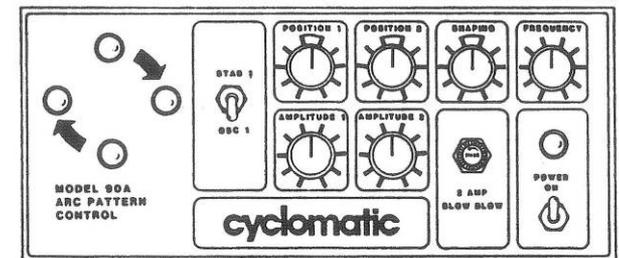
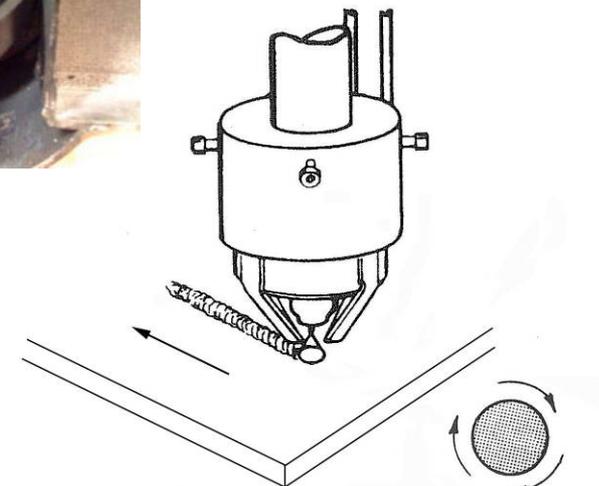
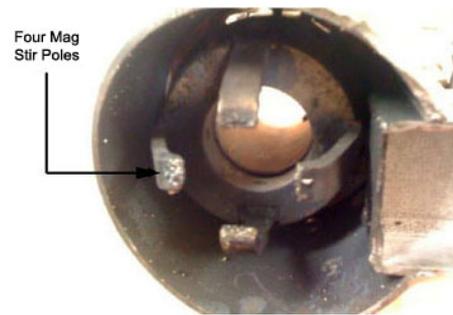
# Magnetic Stir Welding (GTAW)

- **Magnetic field induced to deflect and stir the arc**
  - Circular pattern used for this study
  - Stirring breaks up solidification pattern and produces a smaller grain size
  - Smaller weld metal grains are more conducive UT examination (lower attenuation)
  - Smaller grains also improve resistance to solidification cracking

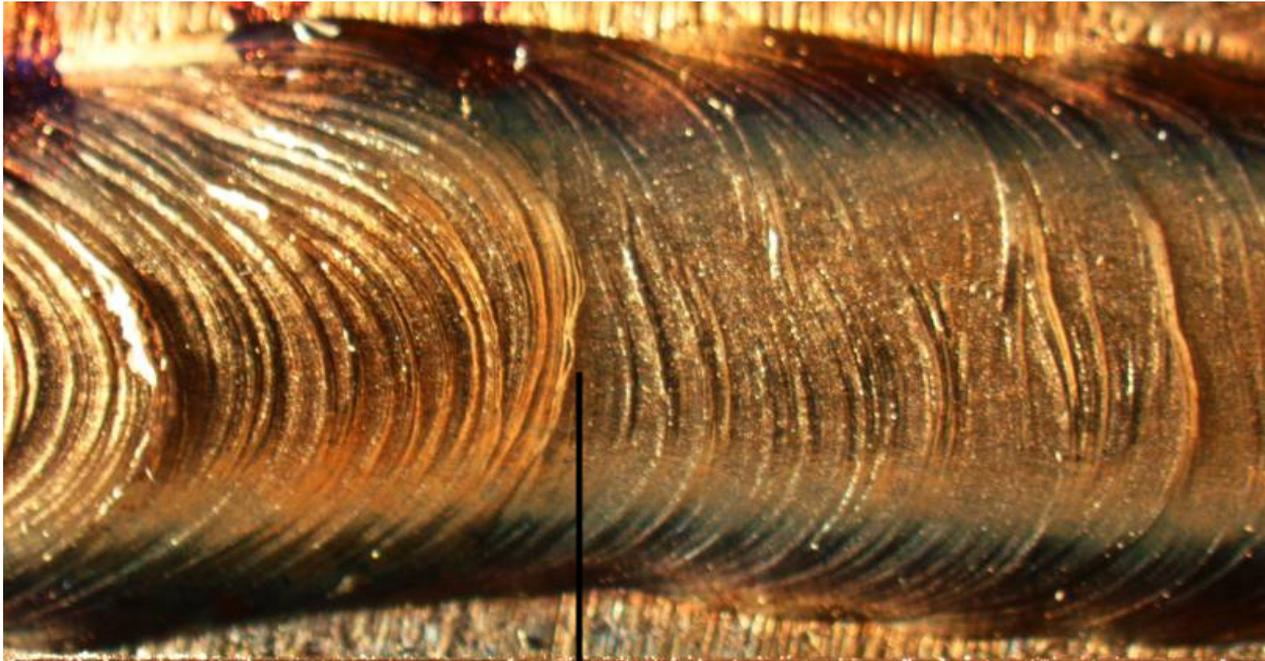


# Magnetic Stirring Equipment and Settings

- **Cyclomatic Model 90A used for feasibility testing**
  - Stirring set to circular arc stirring pattern
  - Testing included autogenous beads and 52M weld pads
  - Standard GTAW torch



# Magnetic Stir GTAW Bead

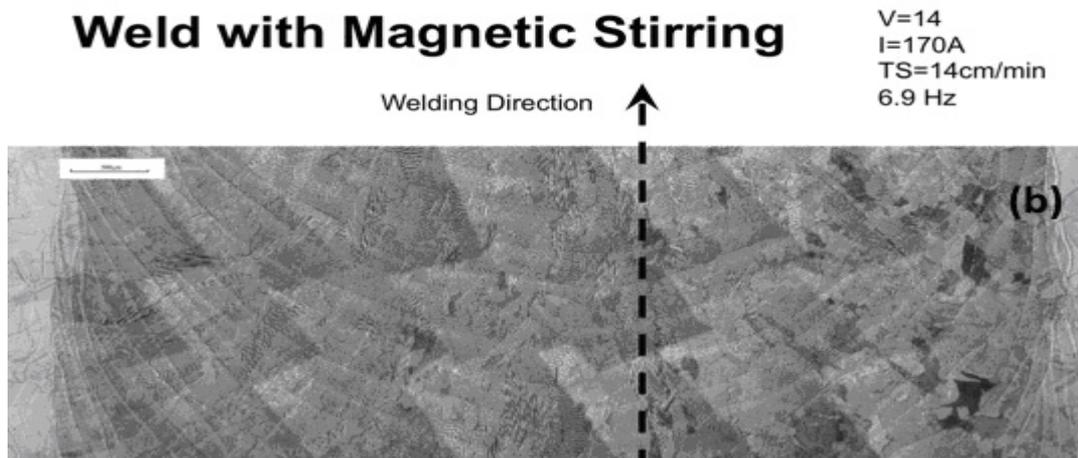
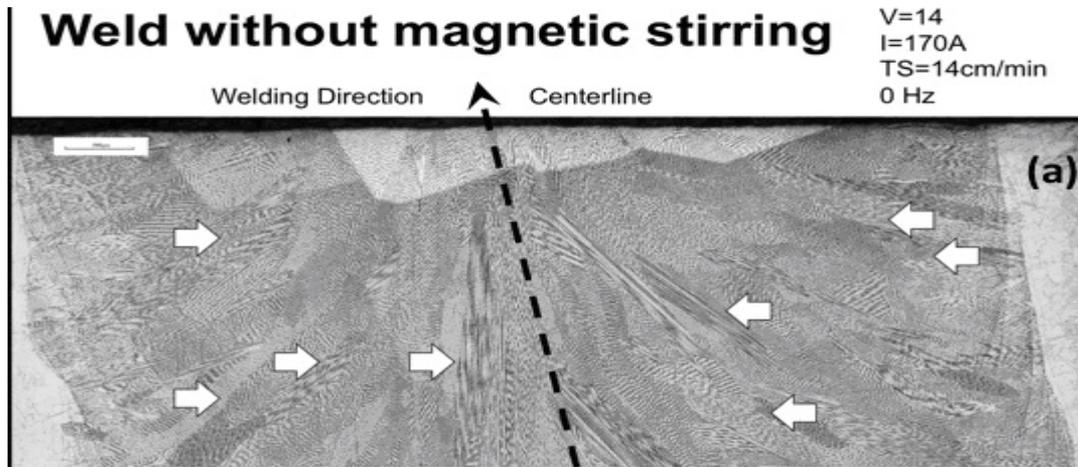


Magnetic Stirring  
(7 Hz)

No Magnetic Stirring

- 170 to 190 amp, 11 volt, 4 to 5.5 ipm travel, 40 to 50 ipm, 7 Hz stir frequency
- 0.035" 52M filler metal on Alloy 690 plate

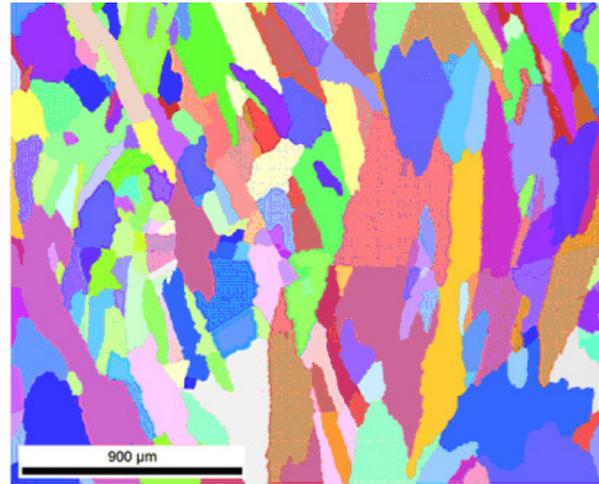
# Grain Size and Orientation with Mag Stir



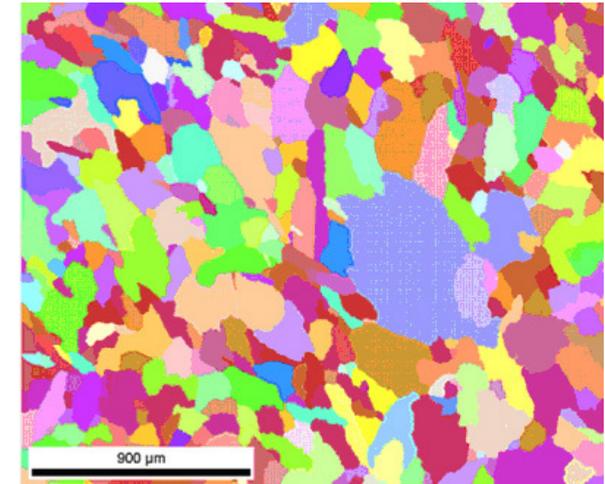
- Magnetic arc stirring breaks up long columnar grains
- 6.9 Hz circular stirring at 14 cm/min travel speed are most effective
- Reversal in weld metal solidification direction is what breaks up the weld metal grain growth

# Results - Electron Back Scattered Diffraction

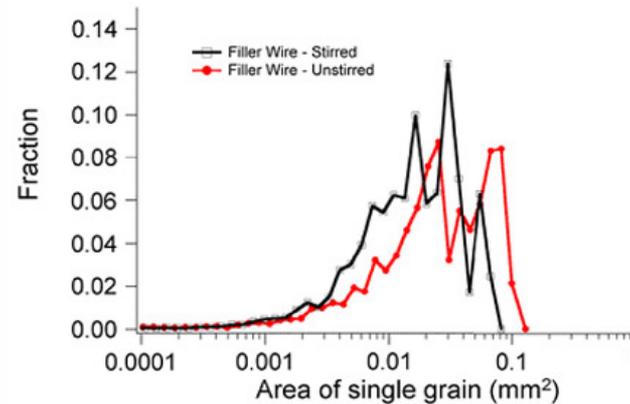
- Electron Back Scattered Diffraction
  - Method to look at grain size
  - Significant reduction in grain size with 6.9 Hz circular magnetic stirring



Unstirred (0 Hz)

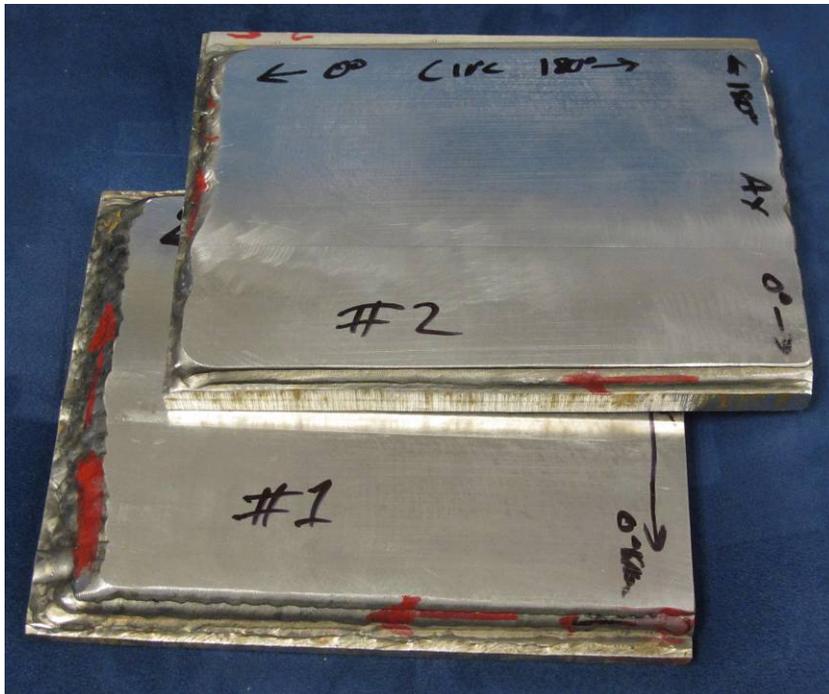


Stirred 6.9 Hz



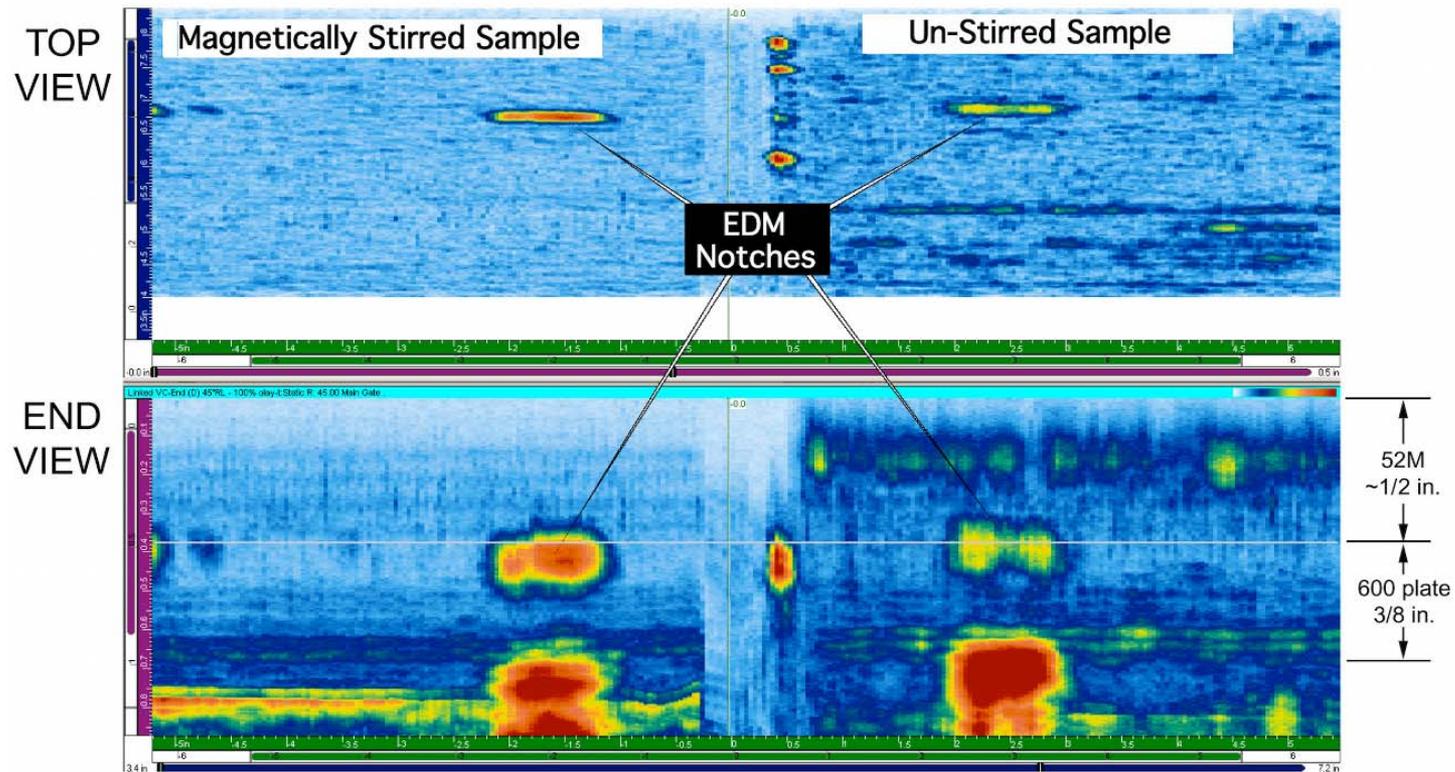
# Specimens for Ultrasonic Examination

- Two weld 52M pads on 690 plate
  - #1 non-pulse GTAW parameters
  - #2 with optimized magnetic stirring



- Weld pads prepared for UT examination
  - Machined flush with  $\sim 3/8$ " 52M thickness
  - Circ & axial EDM notches machined on plate back surface to 52M fusion line depth

# Improved Ultrasonic Response



- 45° RL Axial Scan

- 13:1 to 20:1 (+ & - scan direction) signal-to-noise ratio with stirring
- 5:1 & 8:1 (+ & - scan direction) signal-to-noise ratio without stirring

# Improved Resistance to Solidification Cracking

- 52M weld pad on Type 303 plate clad with ER308L-Si weld metal
- Surface micrographs with & without magnetic arc stirring
- Standard GTAW without stirring
  - 11.5 Volt
  - 240 Amp
  - 4 ipm travel speed
  - 58 ipm wire feed speed (0.045" dia.)
- GTAW with magnetic stirring
  - Parameters same as above with 7Hz stirring



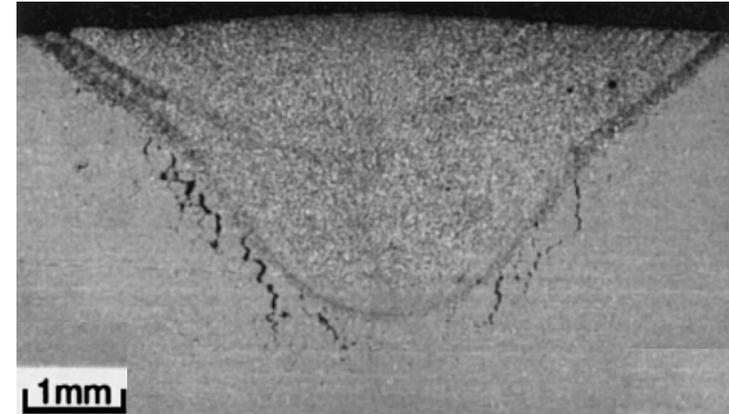
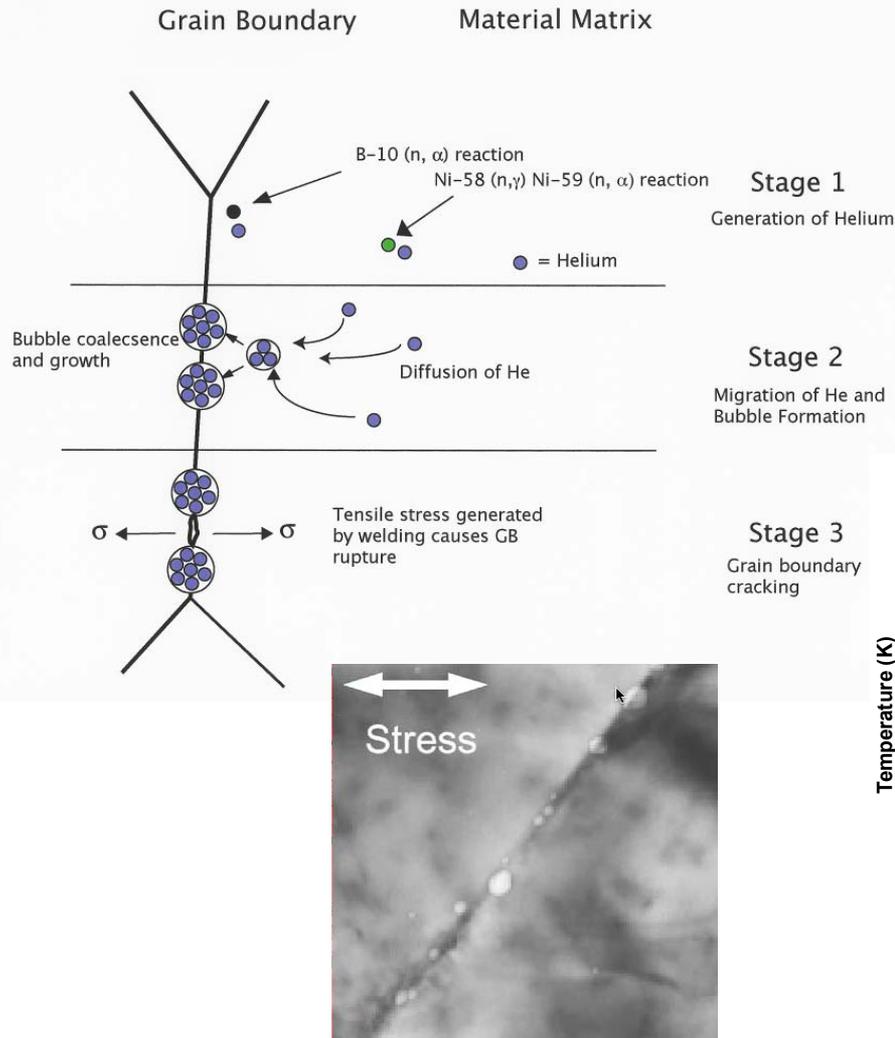
# Magnetic Stir GTAW – Future Work & Potential

- Testing indicates that GTAW with optimized magnetic stirring:
  - Interrupts the solidification pattern at the weld puddle fusion line
  - Breaks up large columnar grains typical of nickel-base welds
  - Ultrasonic examination response is improved by smaller grains (lower sound attenuation with smaller grains)
- Preliminary testing shows GTAW with magnetic arc stirring:
  - Improves resistance to solidification **cracking**
  - Can GTAW pulse parameters (or GTAW waveform controls) be used to duplicate the magnetic stirring effect?
    - Preliminary studies with pulse parameters and variable polarity are in process

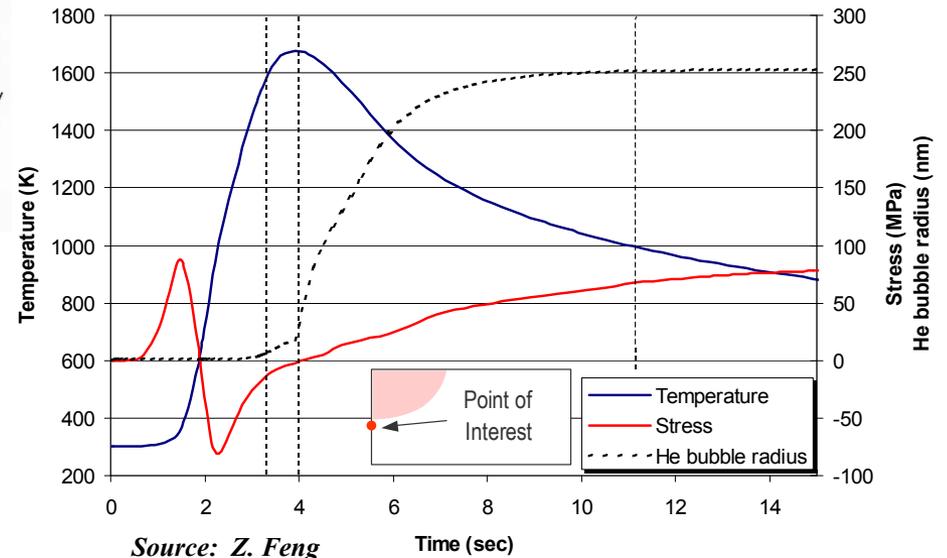
# Presentation Roadmap

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  - Friction Stir Welding

# Cracking Mechanism of Irradiated Materials



Asano et al. *J. Nucl. Mat.* 264 (1999)1-9



Source: Z. Feng

# Presentation Roadmap

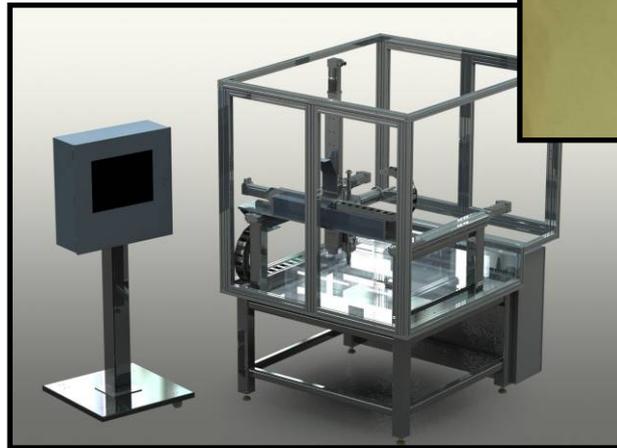
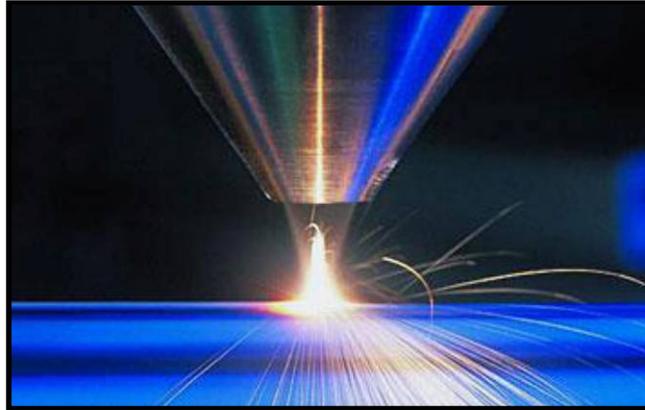
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# Laser Welding - Project Objectives and Scope

- Laser Beam Welding
  - Develop field deployable laser welding techniques for the repair or replacement of irradiated reactor materials
  - Install IPG fiber-laser and robotic manipulator
  - Develop low heat input laser beam welding parameters
  - Develop methodology for application of different weld types on irradiated material
    - WOL
    - Groove weld
    - Fillet weld
  - Develop BWRVIP guideline for laser beam welding

# WRTC Fiber Laser Welding System

- **WRTC Laser Welding Equipment**
  - **Fiber optic IPG Photonics system**
  - **Nominal 2000 watts**
  - **Laser Mechanisms weld head (three focal lengths - 150mm, 200mm, & 250mm)**
  - **Four axis (x, y, z, angular tilt) positioner designed and fabricated by Dynamic Design Solutions**

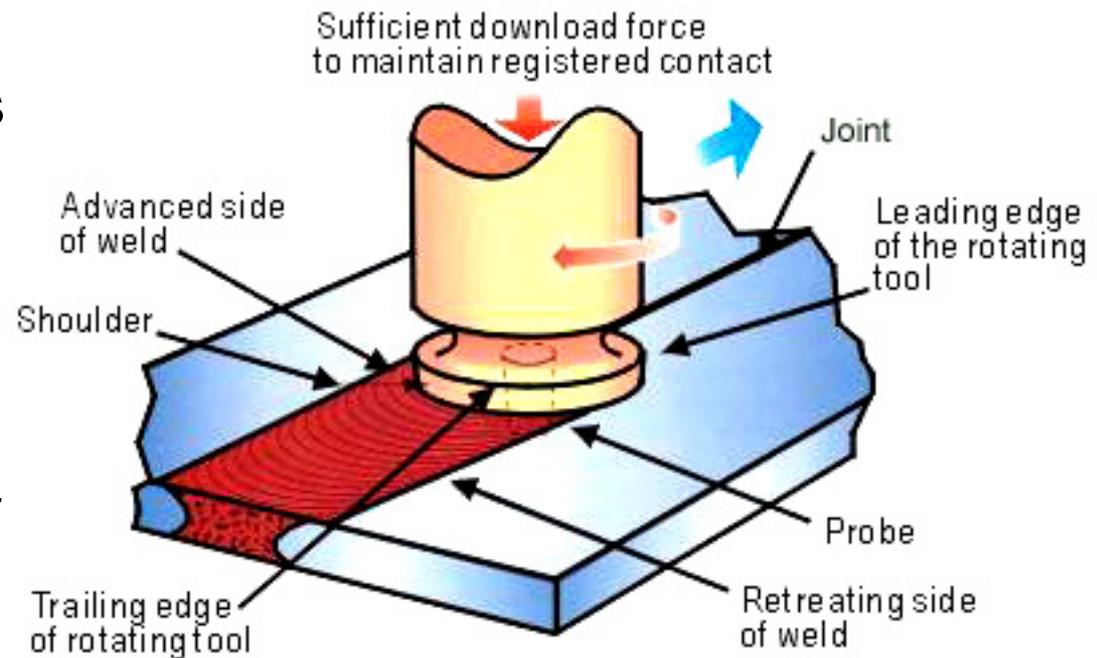


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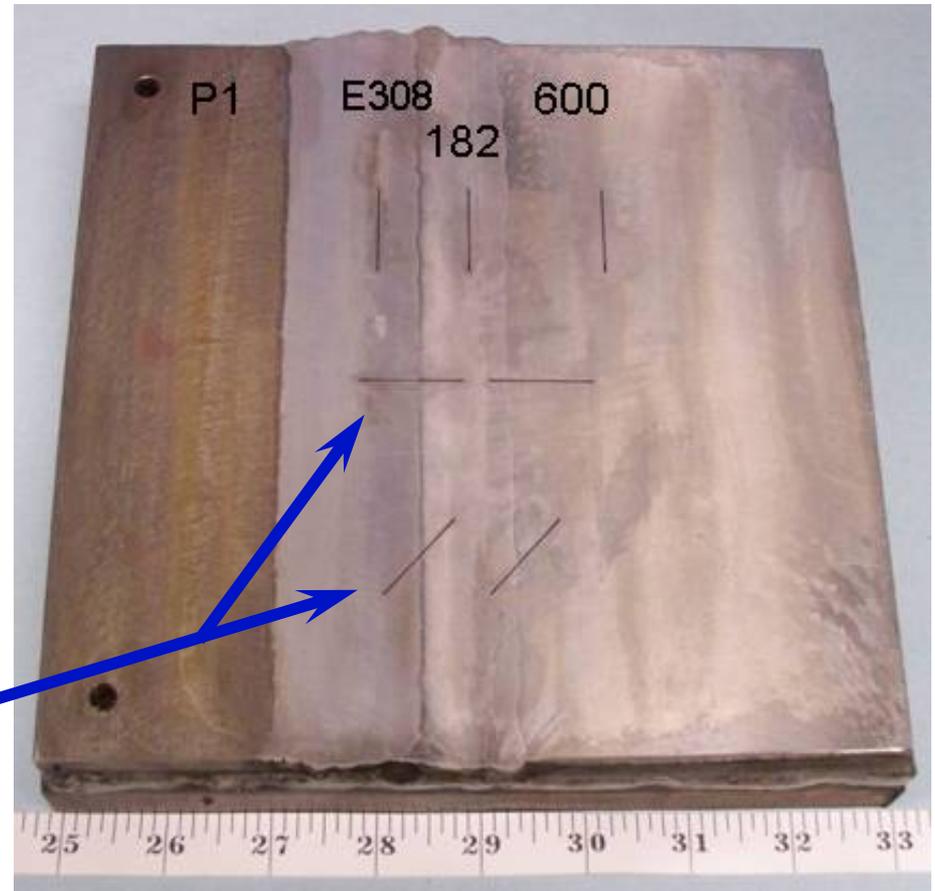
# Friction Stir Welding - Objectives and Scope

- Friction Stir Welding (FSW)
  - Assess underwater FSW welding process to seal IGSCC or other cracks in reactor internal components
- Develop FSW technology for reactor internal repairs
  - Installation of buffer pad (plate) to irradiated components
  - Seal IGSCC or other crack-like defects
  - Weld repairs on range of carbon steels, stainless steels, and nickel-base alloys

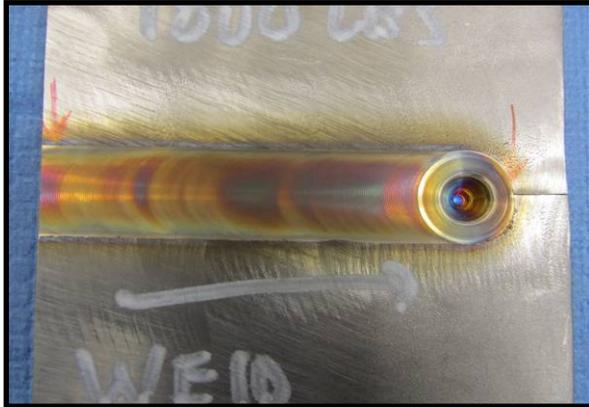


# Dissimilar Material Crack Sealing Test Plate

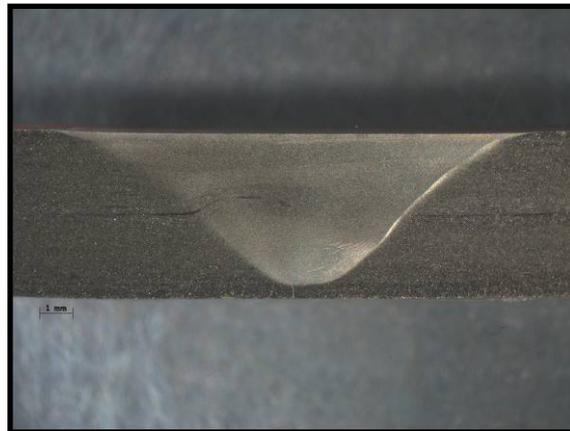
- Assess FSW crack sealing capability and welding on:
  - Wrought Type 304 plate
  - ER308L weld metal
  - Alloy 600 base metal
  - Alloy 182 weld metal
- Dissimilar metal weld test plate
- EDM notches simulate cracks along various orientations



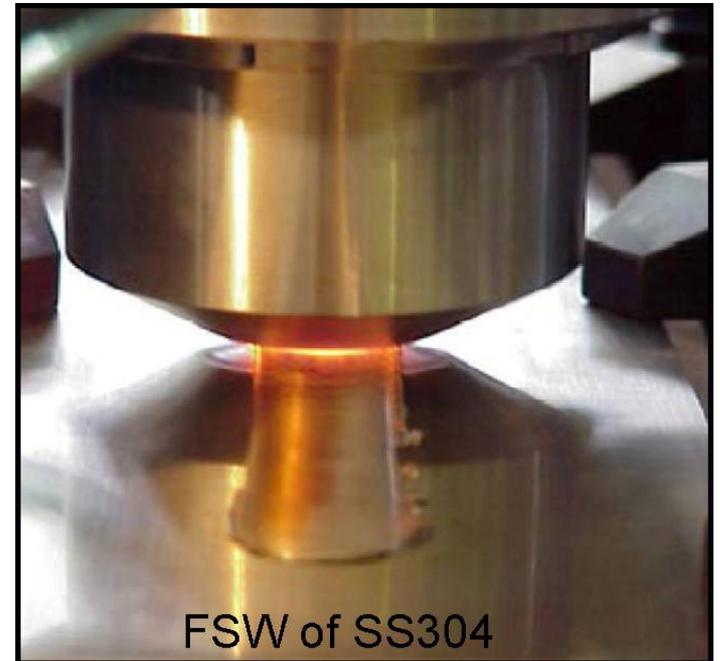
# Example of FSW on Type 304 SS



*Squared Groove Butt Weld on  
1/4 thick 304SS Plate*



*Cross Section of Butt Weld*

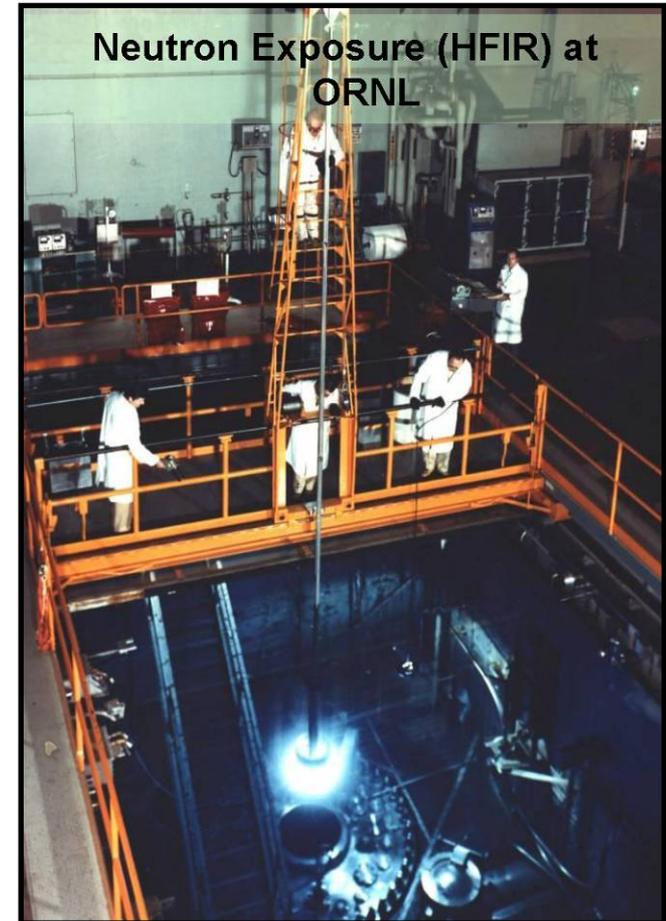


# LTO/LWRS Project Objectives and Scope

- Develop advanced welding technology required for reactor repair and upgrade to support reactor life extension beyond 60 years with an integrated approach between Industry EPRI/LTO and the DOE/LWRS-ORNL
  - Development of advanced welding technologies to weld highly irradiated material
  - Development modeling simulation to guide processes development and predictive application on irradiated materials
- Development of welding hot cell to deploy advance welding and coating processes
- Material degradation assessment development
  - Advanced weld simulation tool for lifetime prediction and weld performance assessment (future years)

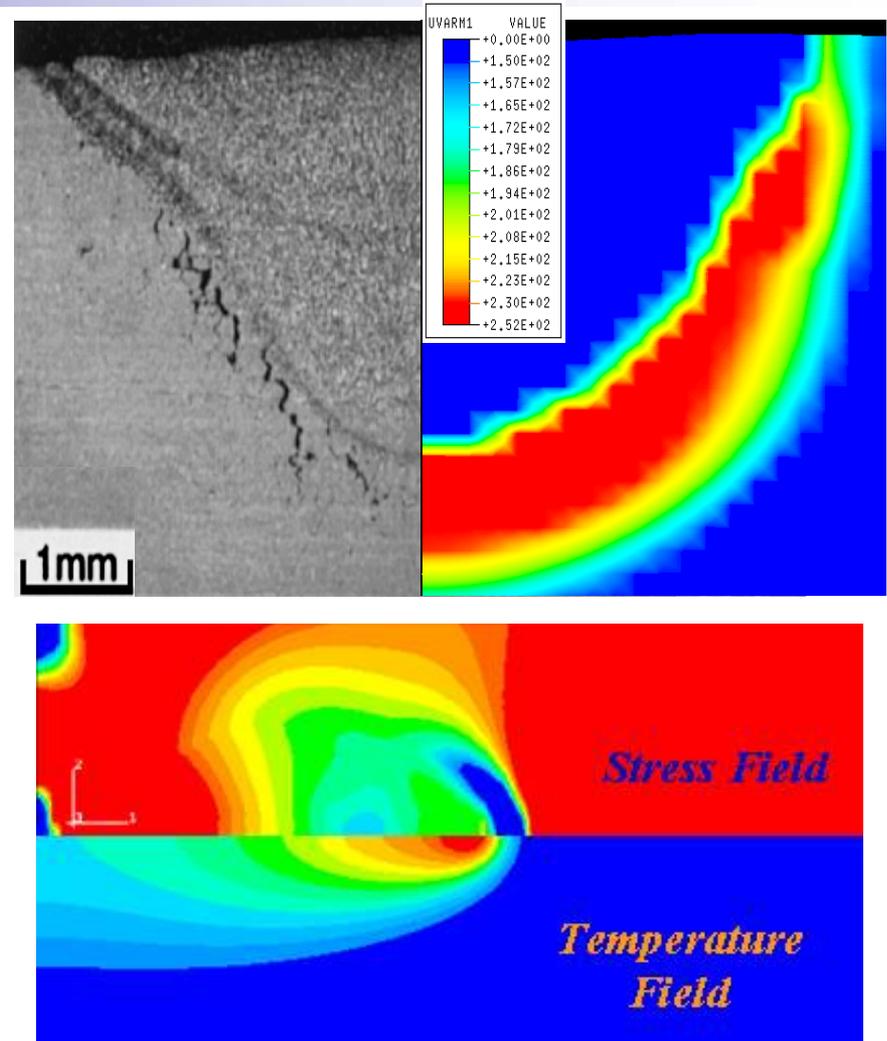
# LTO/LWRS Project Task 1 (2010/11)

- Fabricate irradiated sample set for welding experiments
  - Type 304, Type 316, and Wrought 182
  - Determine initial boron concentration to achieve desired helium level using HFIR irradiation details
    - Flux, energy spectrum, etc.....
    - Target boron levels for 50, 60 & 70 year reactor internal life
  - Determine the steel making practice samples
    - Powdered metallurgy
    - Conversional steel making practice (VIM)
    - Hot working
  - Detail calculations for HFIR exposure and sample holder design



# LTO/LWRS Project Task 2 (2010/11)

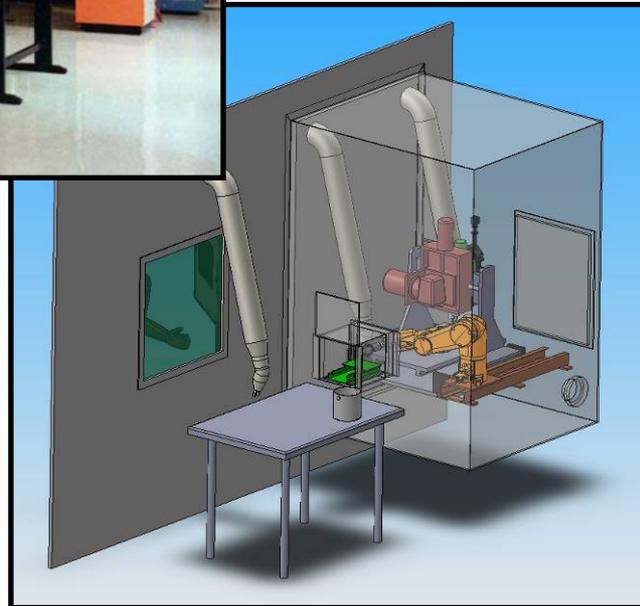
- Survey hybrid welding processes
- Develop computational model for hybrid processes
- Develop hybrid laser weld process model to optimize the weldability of irradiated material
  - Model based on welding process development
- Develop experiment methodology for direct measurement of transient high-temperature and stress history during welding



# LTO/LWRS Project Task 3 (2010/11/12)



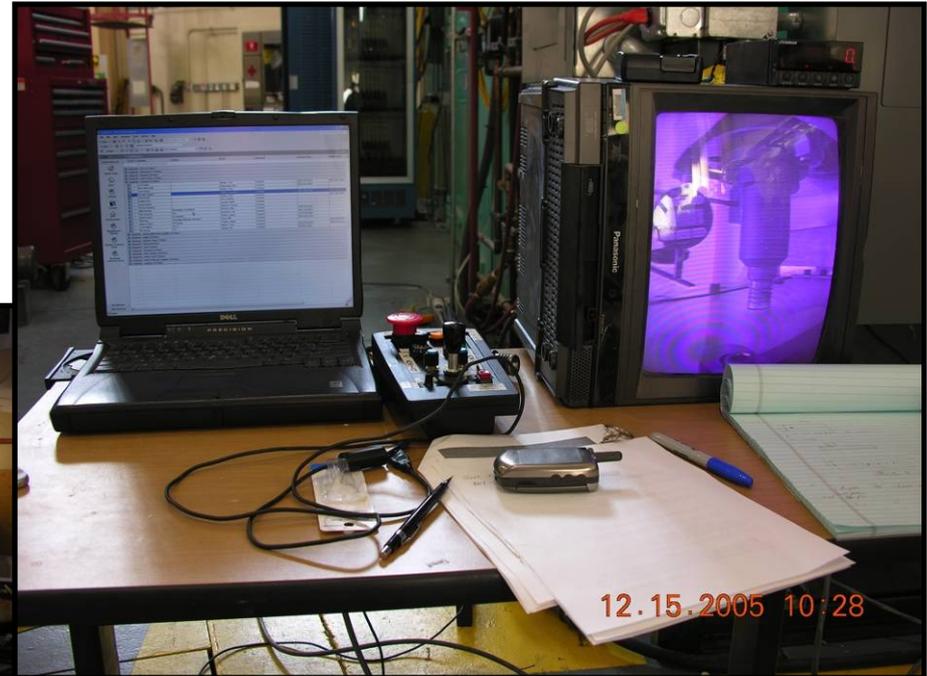
- EPRI is collaborating with DOE to design and develop a New welding hot cell at ORNL



- **Welding Capabilities**
  - Conventional and hybrid laser
  - Friction stir
  - Ultrasonic
  - Powder coating
  - Cold spray

# LTO/LWRS Project Task 3 (2010/11) Cont.

Motion control platform centered around FSW system



## LTO/LWRS Project Task 4 (2010/11)

- Advance modeling of hybrid welding process and optimization of stress state for welding irradiated materials
- Installation of laser welding cell at EPRI Charlotte facility
  - New fiber laser welding (2kW) system
  - Procurement of secondary heat sources
  - Procurement and installation of manipulator
- Welding experiments with real time stress measurement
  - Provide feedback for calibration of hybrid welding model

# Thank You – Questions or Comments?

