

# *Overview of SCC Programs at SERCO, Ciemat and MHI*

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*NRC-EPRI Meeting*

*June 2011*

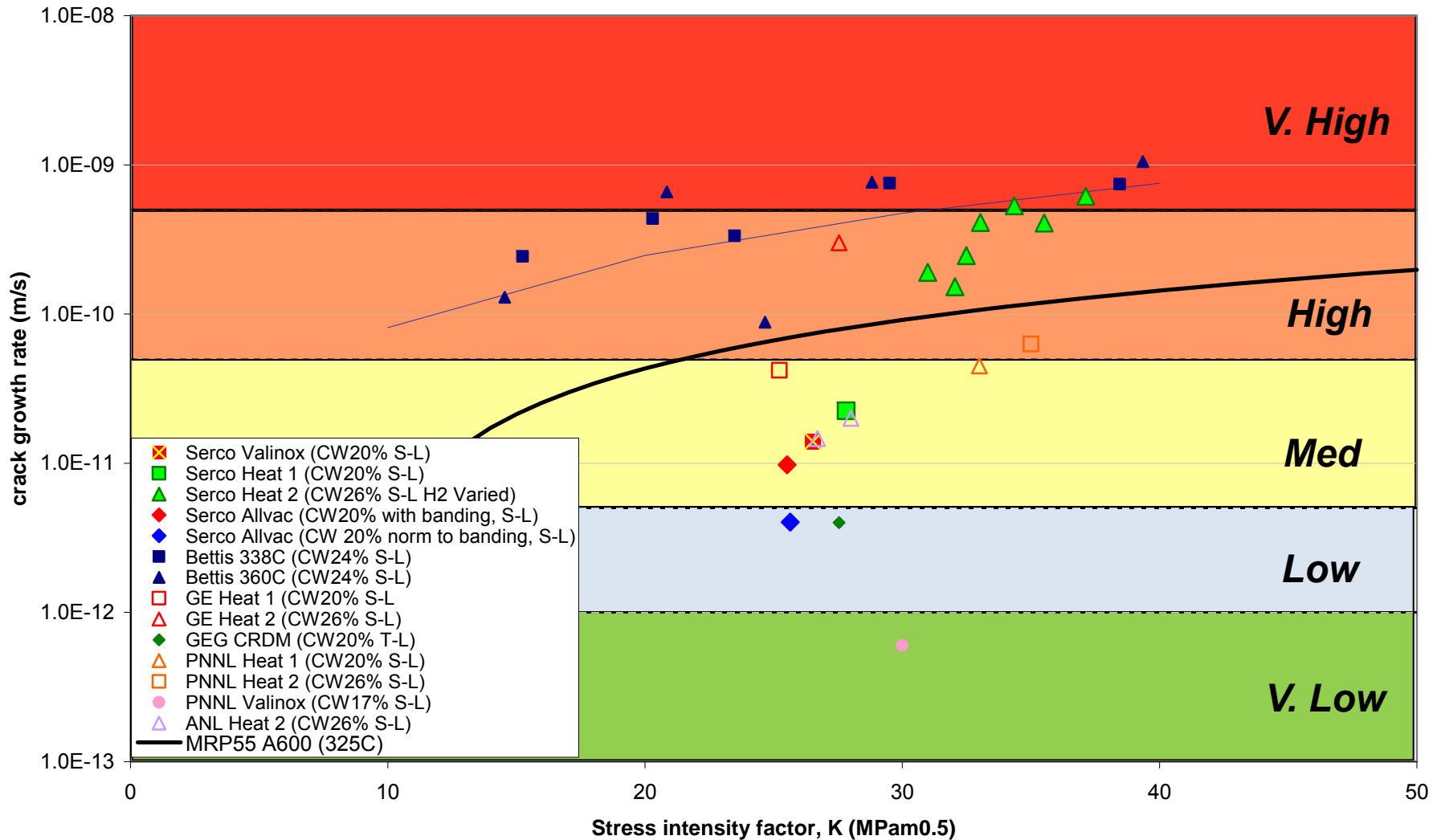


## Serco – Summary of Materials Tested

- **GE MATERIAL:** S-L, 20%CW (2 PASSES), CGR  $1.3-2.3 \times 10^{-8}$  mm/s
  - SUPPLIED IN CW FORM BY GE-GRC
- **ANL MATERIAL:** S-L, 26%CW (3 PASSES), CGR  $1.9-4 \times 10^{-7}$  mm/s
  - SUPPLIED IN CW FORM BY GE-GRC
- **ALLVAC BAR (// TO BANDING):** S-L, 20%CW (10 PASS),  $9.8 \times 10^{-9}$  mm/s
  - HEAVILY BANDED MATERIAL
- **ALLVAC BAR ( $\perp$  TO BANDING):** S-L, 20%CW (8 PASS),  $4 \times 10^{-9}$  mm/s
- **VALINOX CRDM:** S-L, 20%CW (4 PASS), CGR  $1.4 \times 10^{-8}$  mm/s

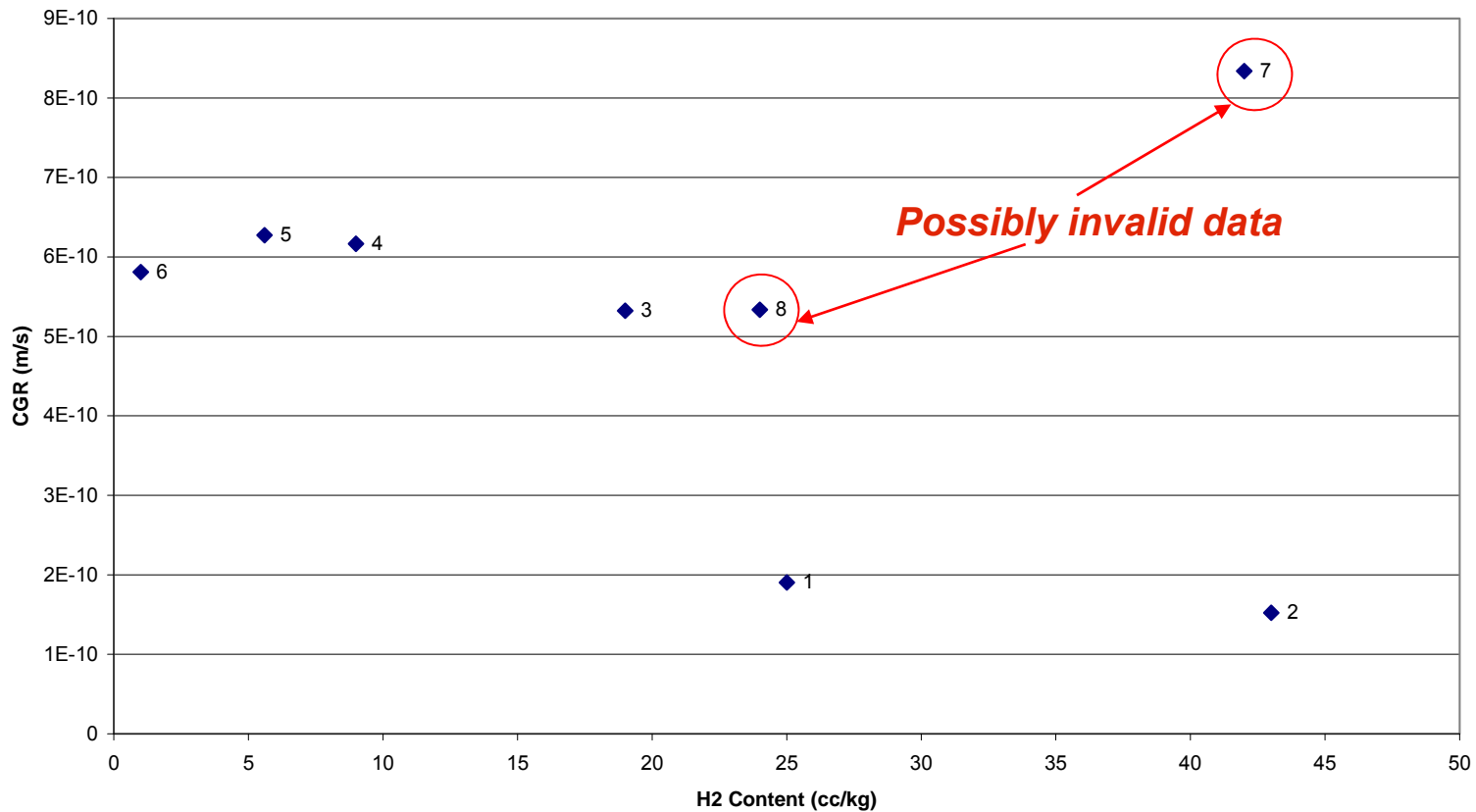
*Not listed, but most of Serco's 690 data is at 350C*

# Comparative rates + classification

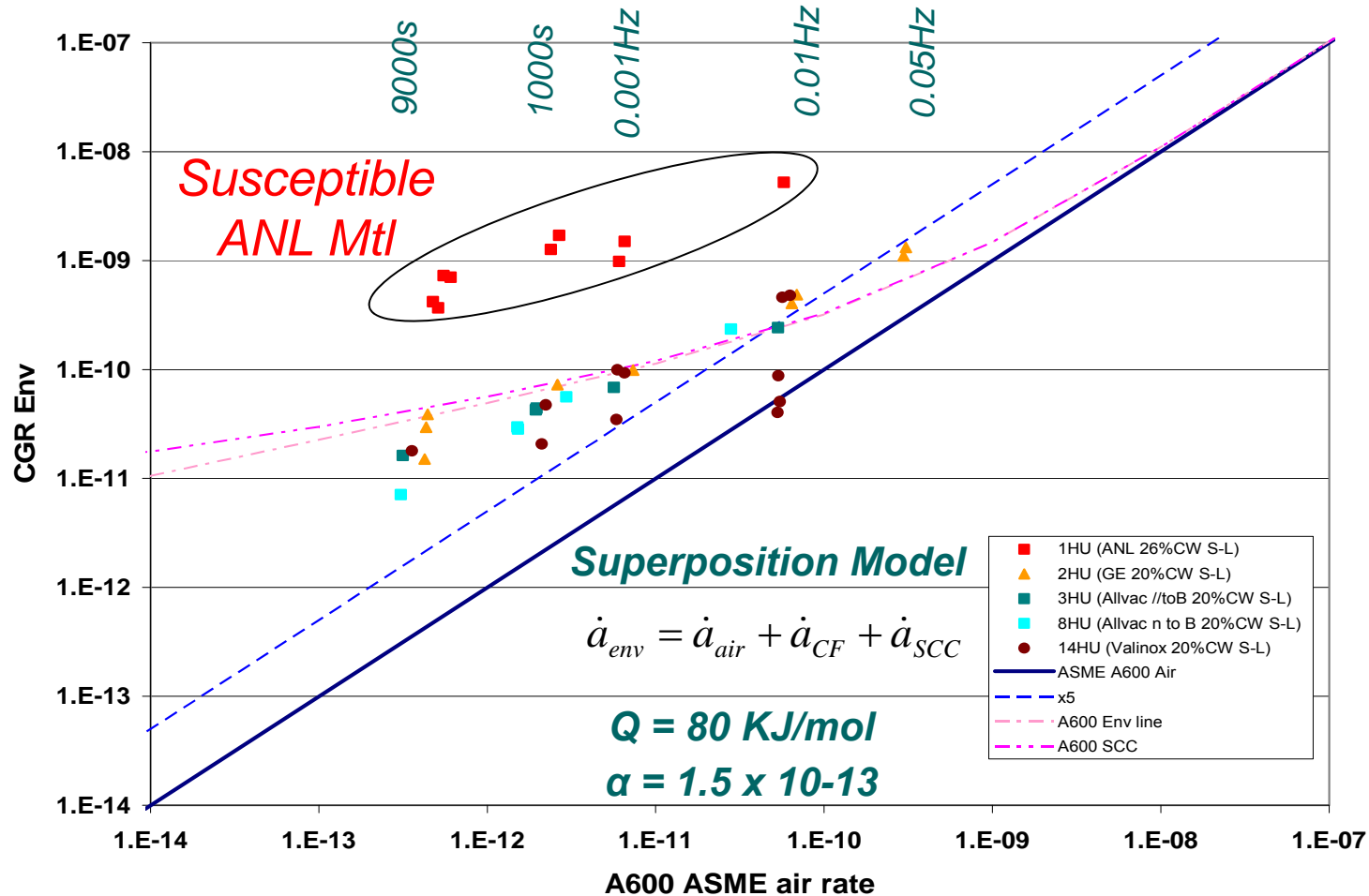


# Effect of Hydrogen Content on Crack Growth Rate ANL heat NX3297HK12

Apparent reduction in crack growth rate at high dissolved H<sub>2</sub> concentration



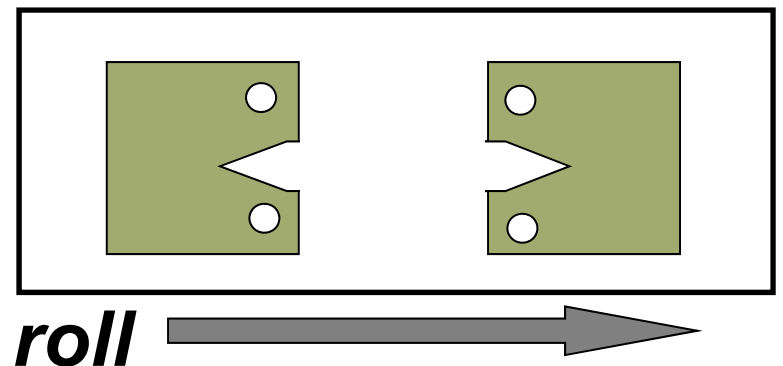
# Data Presentation on Corrosion Fatigue



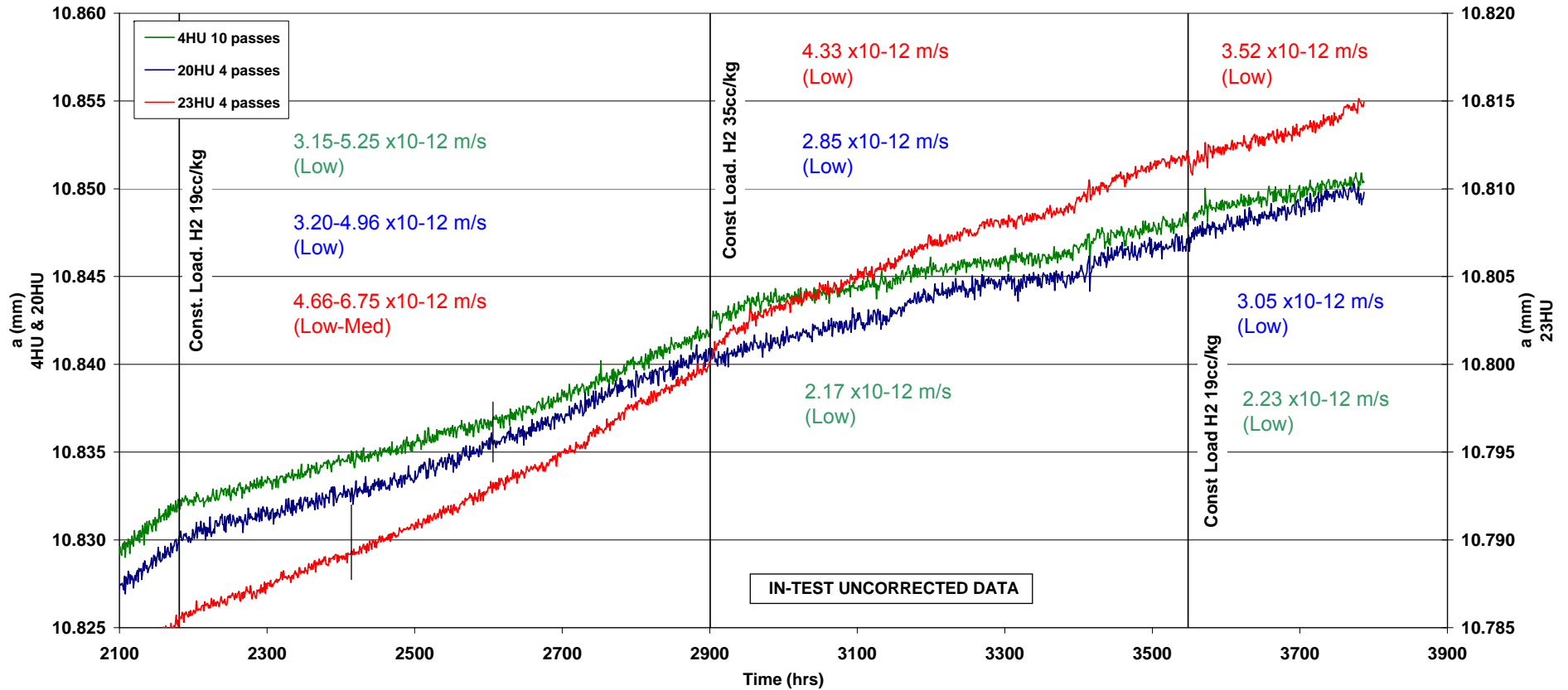
Most data bounded by non-CW Alloy 600 superposition model.  
 26% CW ANL heat showed significantly higher rates.

## Current test - 'Aggressive' rolling effects

- Investigate variability in rolling method to achieve equivalent reduction in thickness.
  - Usually dependent on machine limits.
- It has been suggested that the number of passes used when cold rolling may affect the susceptibility and CGR.
- Testing Allvac billet material (heavily banded).
  - Deliberate ploy to concentrate on one (Allvac) Material.
- 20%CW 1D rolled, with 4 or 10 passes (SL, banding along rolling direction)
- Also looking at CGR with respect to direction of rolling.
- 3 specimen test
  - 10 pass specimen as in previous test
  - two 4 pass specimens cut as shown

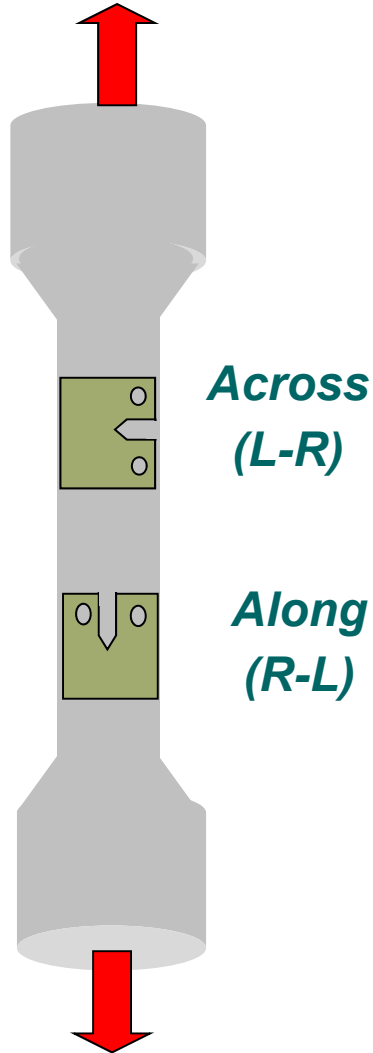


# In-test Data



*Very similar rates for 4 and 10 passes.*

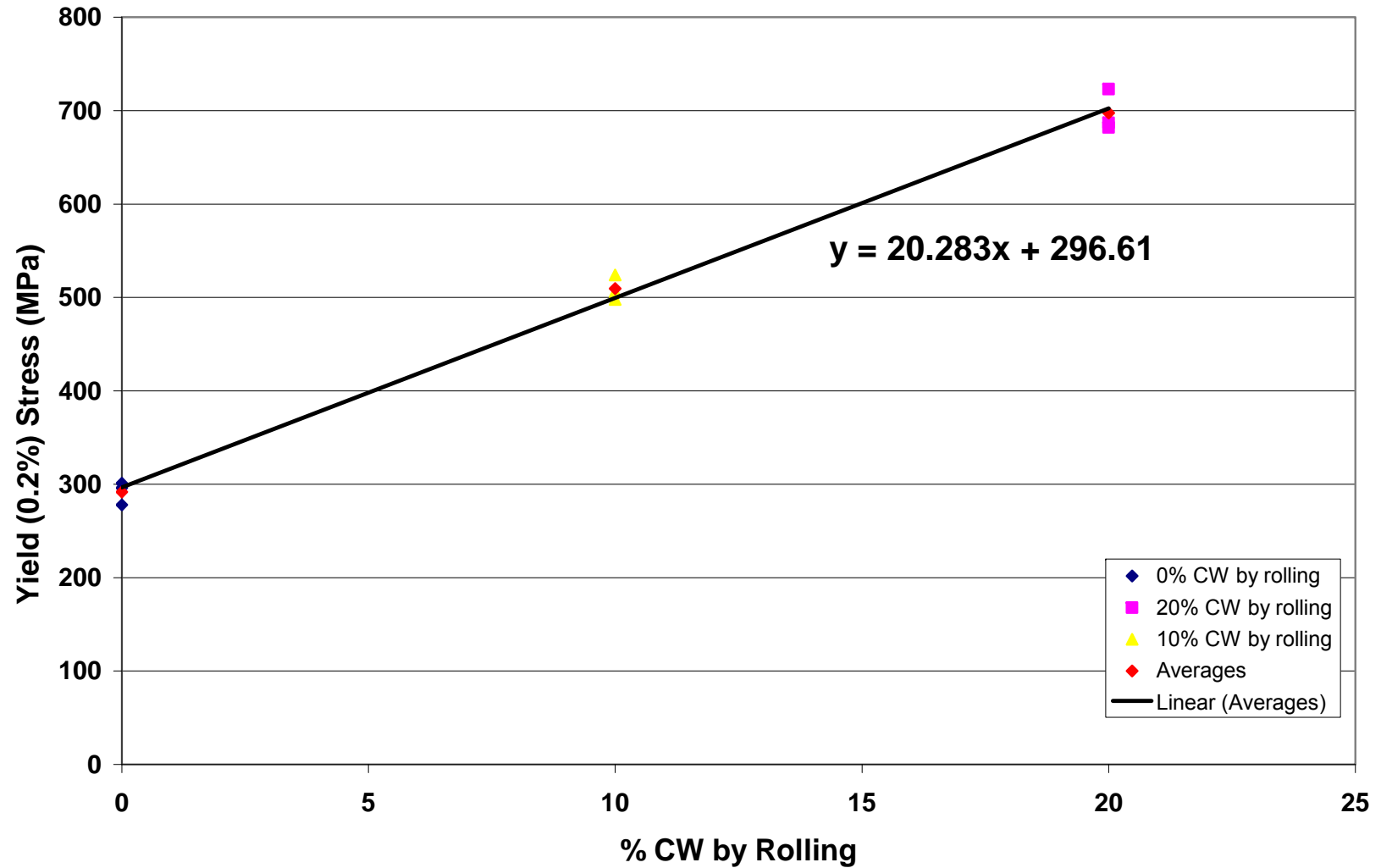
## Upcoming Test - Single Pull Allvac Material



- Compare CGRs with those of previous tests that performed CW by uni-directional cold rolling and forging.
- For direct relation to Cold Working by rolling, performed tensile tests on material CW by 0, 10 and 20% to measure the proof stress.
- Related proof stress to % thickness reduction to calculate true stress to be applied by single pull.
- SCC Tests in 'Along' (R-L) and 'Across' (L-R) orientations
- Compare hardness profiles and microstructures.
- More representative of weld shrinkage?



## Tensile Data of CW Material



## *Single Pull Method*

- Applied True Stress of 700MPa.
- Limited material available, making 8mm CTs.
  - Validity of specimens and available ligament compared with arguments in ASTM 647 and paper by P. Andresen at 11<sup>th</sup> Env Deg, K/size effects.
- Test alongside a CT from a 20% 1D cold rolled specimen (Allvac  $\perp$  to banding) as control for size effects with an already tested material.

## *SERCO Summary*

- SCC testing of four materials
  - GE Plate, ANL Plate, Allvac Bar and Valinox CRDM.
  - All Low-Med rates with exception of ANL heat (High).
  - Possible H<sub>2</sub> effect measured on susceptible ANL heat.
- Ongoing comparison of data between laboratories
- Concentrating on systematic evaluation of single material - Allvac bar
  - Alignment to banding
  - Hydrogen effects
  - Number of passes used for Cold Rolling
  - Single Pull, comparing application of cold work.

# Influence of rolling and tensile deformation on the Alloy 690 CGR UNESA-EPRI Project

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*L. Francia (UNESA)*

*K. Ahluwalia - EPRI manager*

***Dresden, Germany. May 9, 2011***

*Objective and scope*

- ✓ To compare the response of Alloy 690 TT deformed by rolling and tensile to SCC in primary water conditions.
- ✓ Material have been deformed up to 20%
- ✓ CGR tests have been performed in primary water conditions at 325°C
- ✓ CT specimens (12 mm thickness) with T-L orientation

CRDM Alloy 690TT tube  
supplied by Valinox (heat WP 787)



### Materials

✓ CRDM Alloy 690TT tube supplied by Valinox (heat WP 787)  
Electric Arc Furnace + ESR (Electroslag Remelt)  
Thermal treatment : 1050°C, 6 hours at 715°C

✓ Homogeneous microstructure, high grain boundary coverage with small carbides, no banding. Grain size ASTM No 3.5-4 (~ 90 μm)

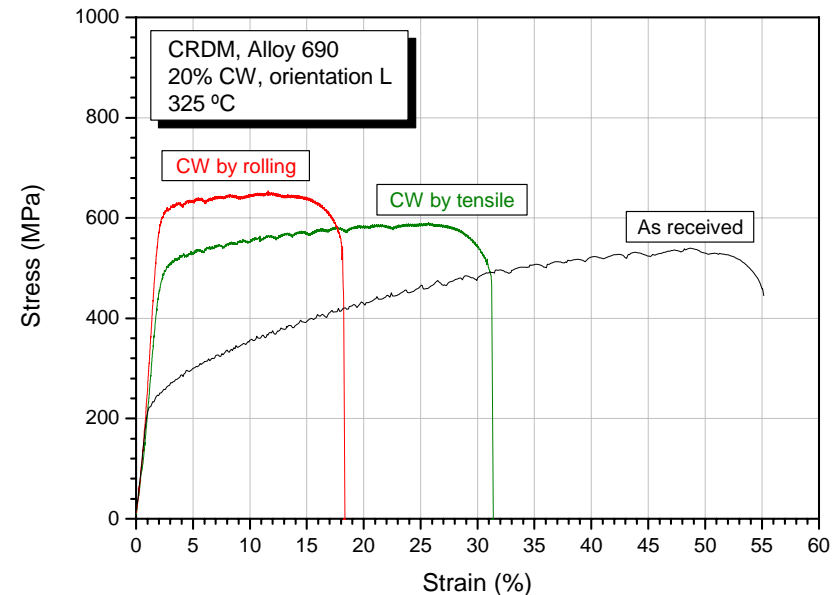
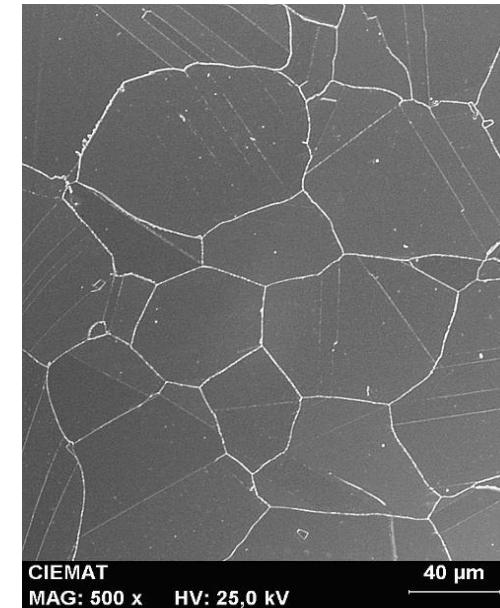
✓ Material has been deformed up to 20% by rolling in three passes and by straining tensile

#### ✓ Average Hardness

As received: 165HV1

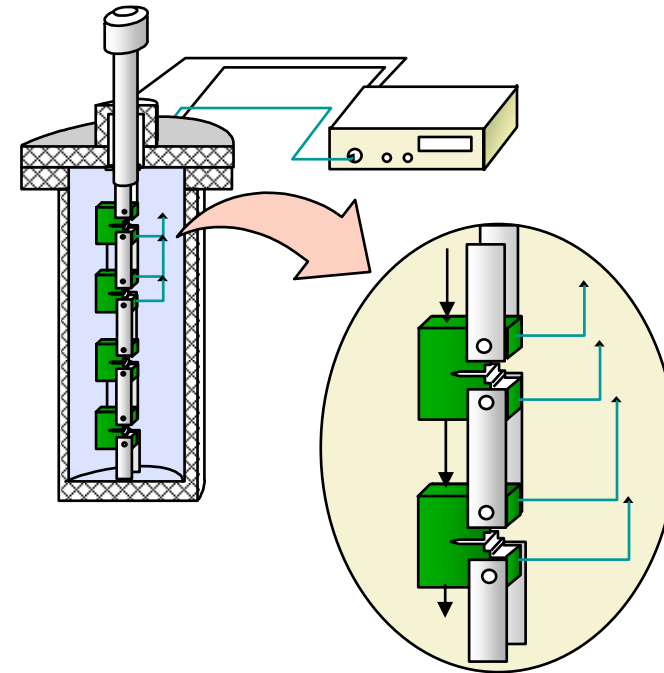
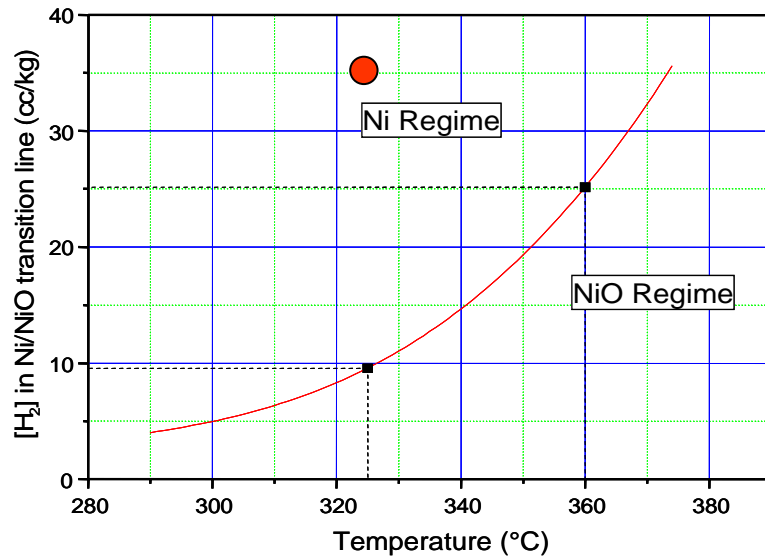
Tensile deformed : 279HV1

Rolling deformed : 298HV1



## Experimental conditions

900 ppm B, 2.5 ppm Li, pH=7.2 at 310°C  
 $H_2 = 35 \text{ cc/Kg } H_2O$  at 325°C (●)

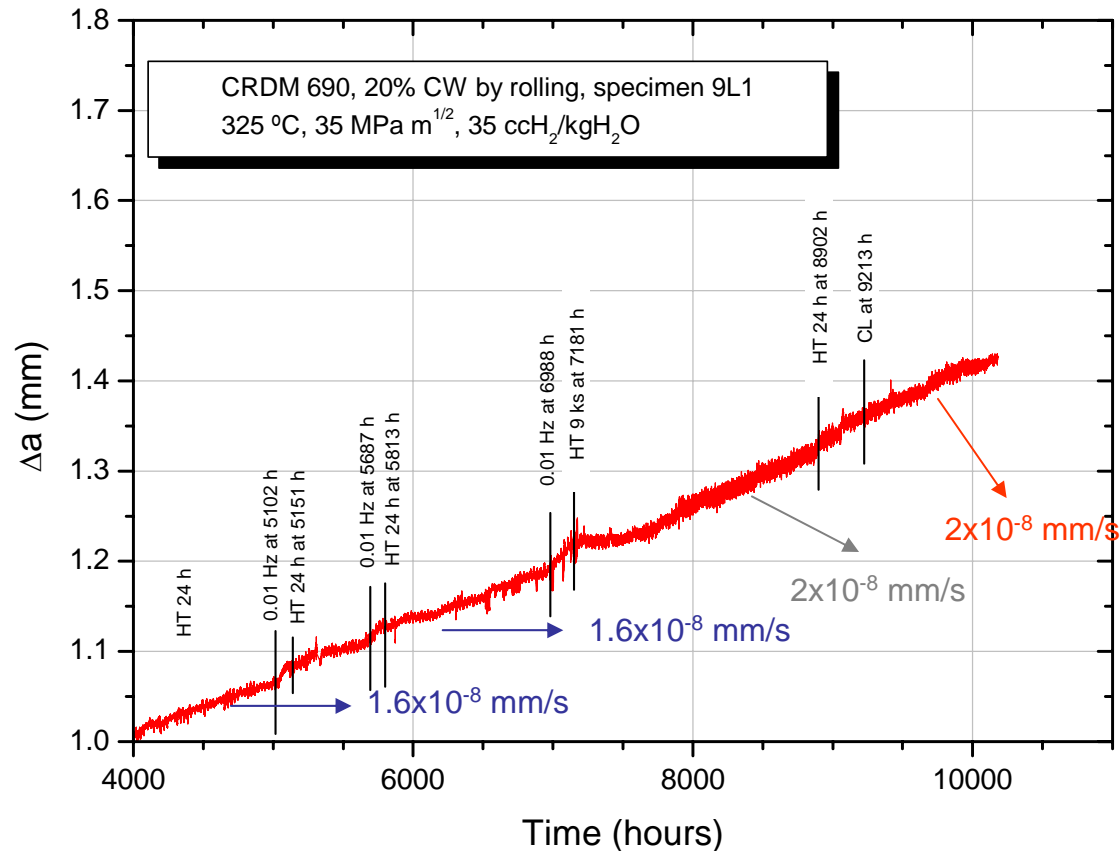


Four CT specimens in daisy chain. All of them instrumented by DCPD

- ✓ Pre-cracking in air:  $R = 0.3$  to  $0.7$  and  $1-2 \text{ Hz}$
- ✓ Fatigue pre-cracking in high T water:  $R = 0.7$  and  $0.05 - 0.01 - 0.001 \text{ Hz}$
- ✓ Constant Load (CL) with gentle unloading:  $0.001 \text{ Hz} + 9000 \text{ s}$  and  $24 \text{ h}$  hold time
- ✓ Pure constant load.

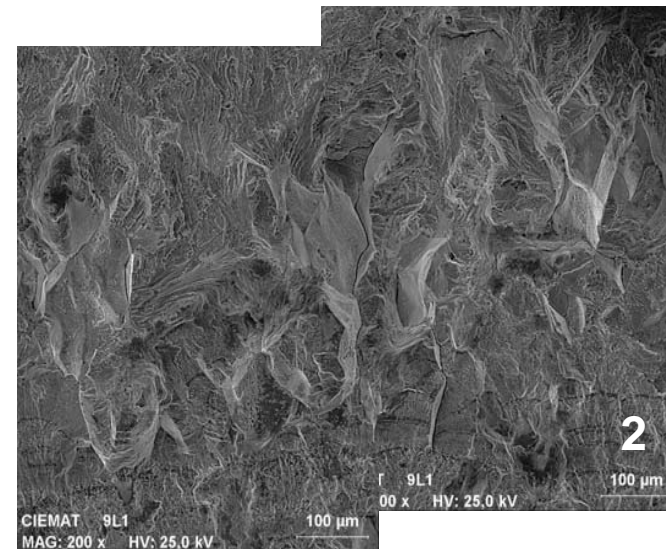
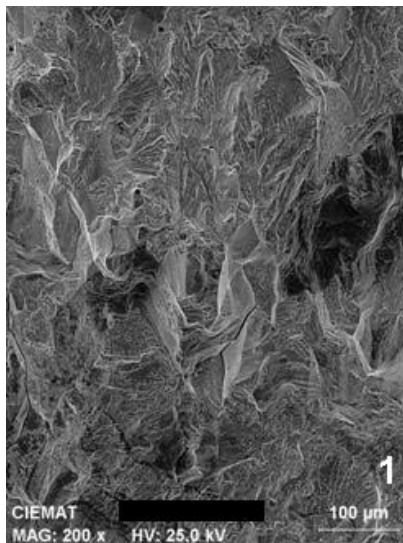
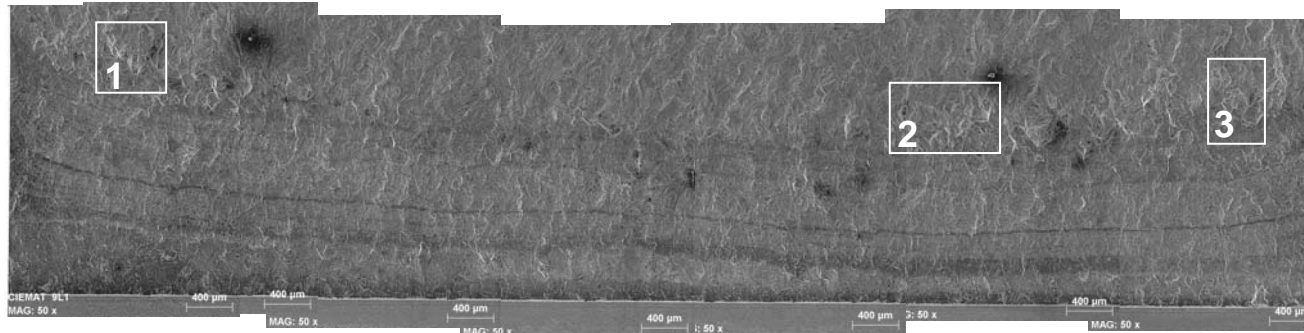
Specimen **9L1. 20% CW by rolling. CRDM tube WP787**  
 325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m

*DCPD data during hold time and constant load periods (corrected by fractography)*





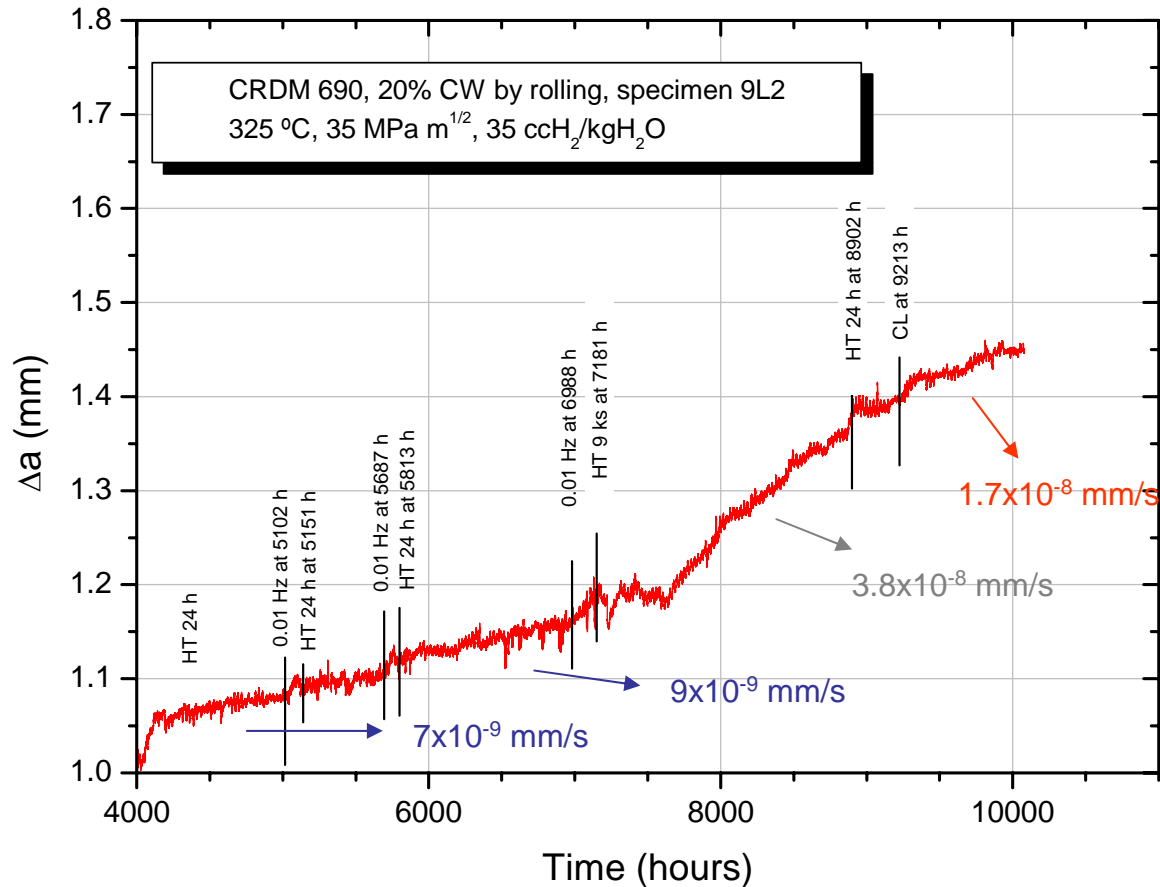
Specimen 9L1. 20% CW by rolling. CRDM tube WP787  
325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m



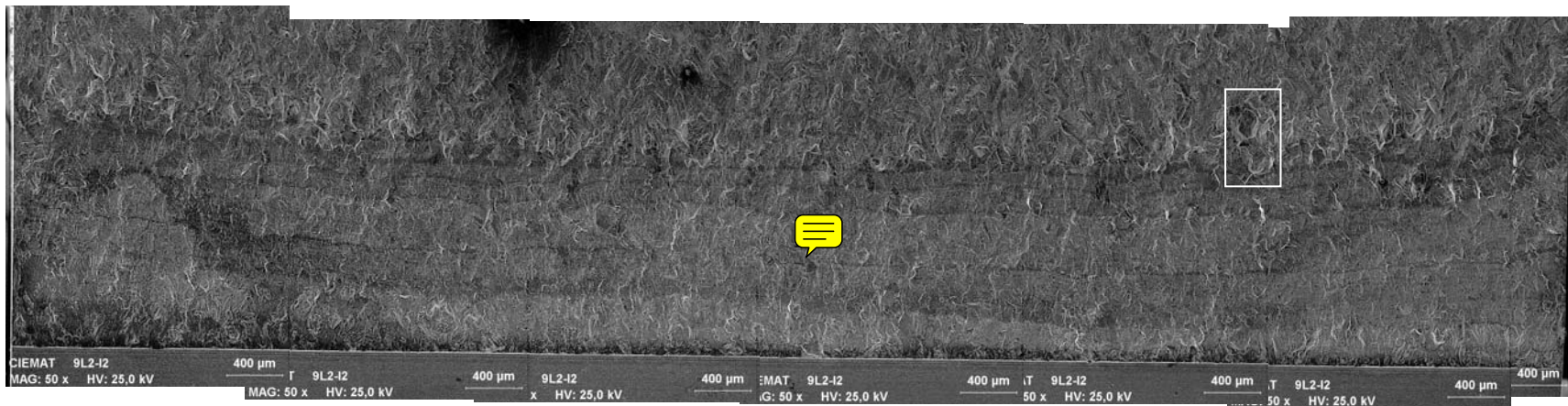
Intergranular areas at the crack tip

Specimen 9L2. 20% CW by rolling. CRDM tube WP787  
 325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m

DCPD data during hold time and constant load periods (corrected by fractography)



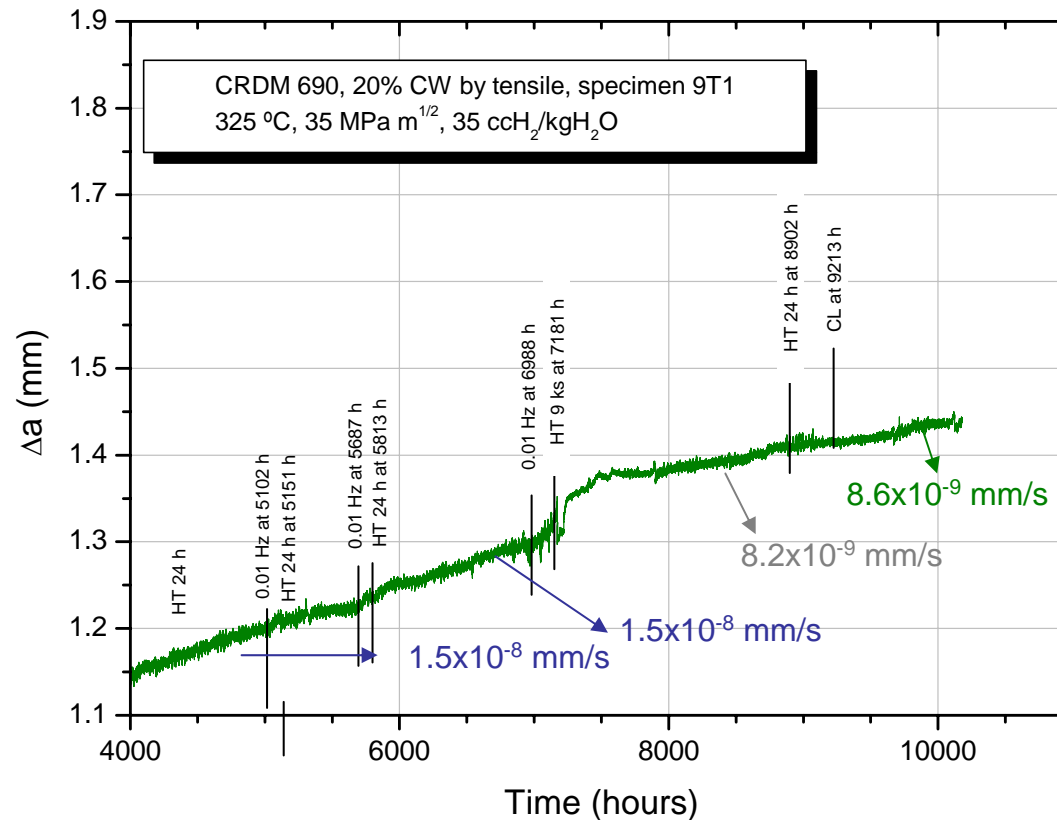
Specimen 9L2. 20% CW by rolling. CRDM tube WP787  
325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m



- ✓ The fracture appearance is very similar to the specimen 9L1
- ✓ Estimated final K<sub>I</sub> = 36.8 MPa√m

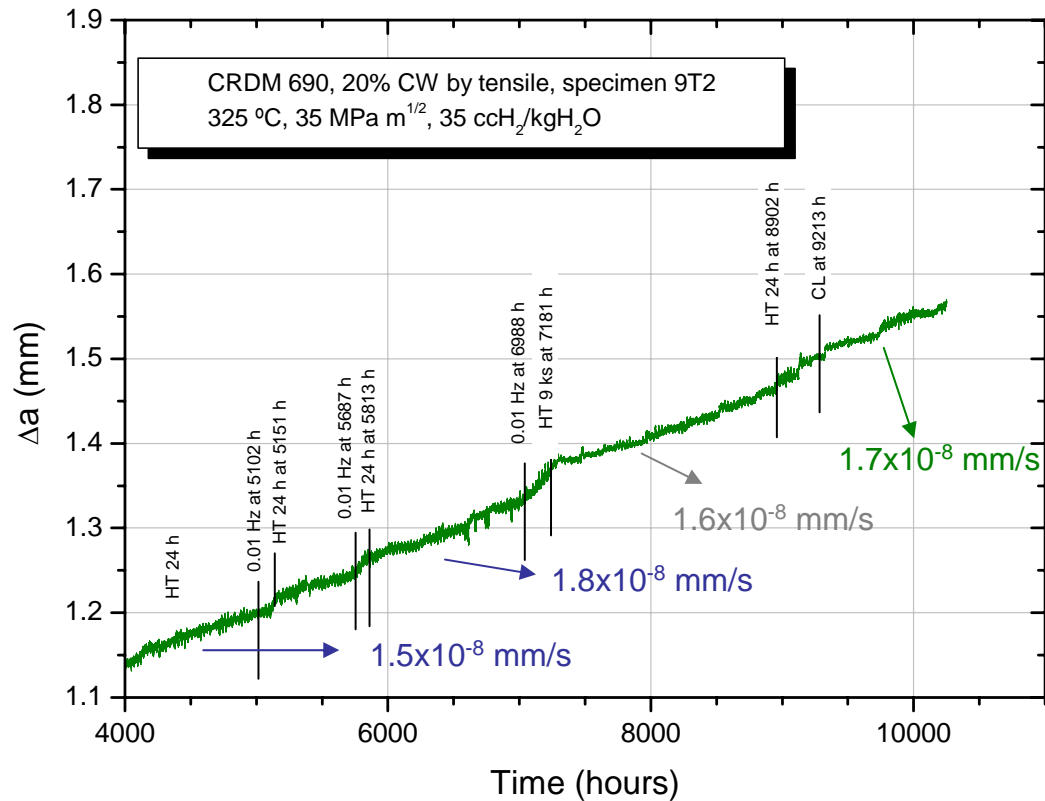
Specimen 9T1. 20% CW by tensile. CRDM tube WP787  
 325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m

DCPD data during hold time and constant load periods (corrected by fractography)

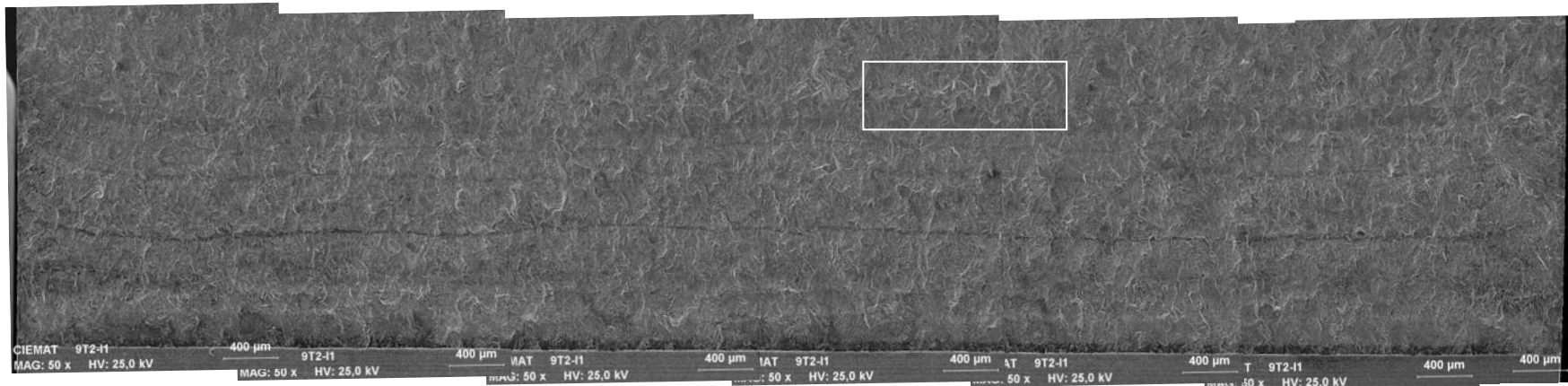


Specimen 9T2. 20% CW by tensile. CRDM tube WP787  
 325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m

DCPD data during hold time and constant load periods (corrected by fractography)



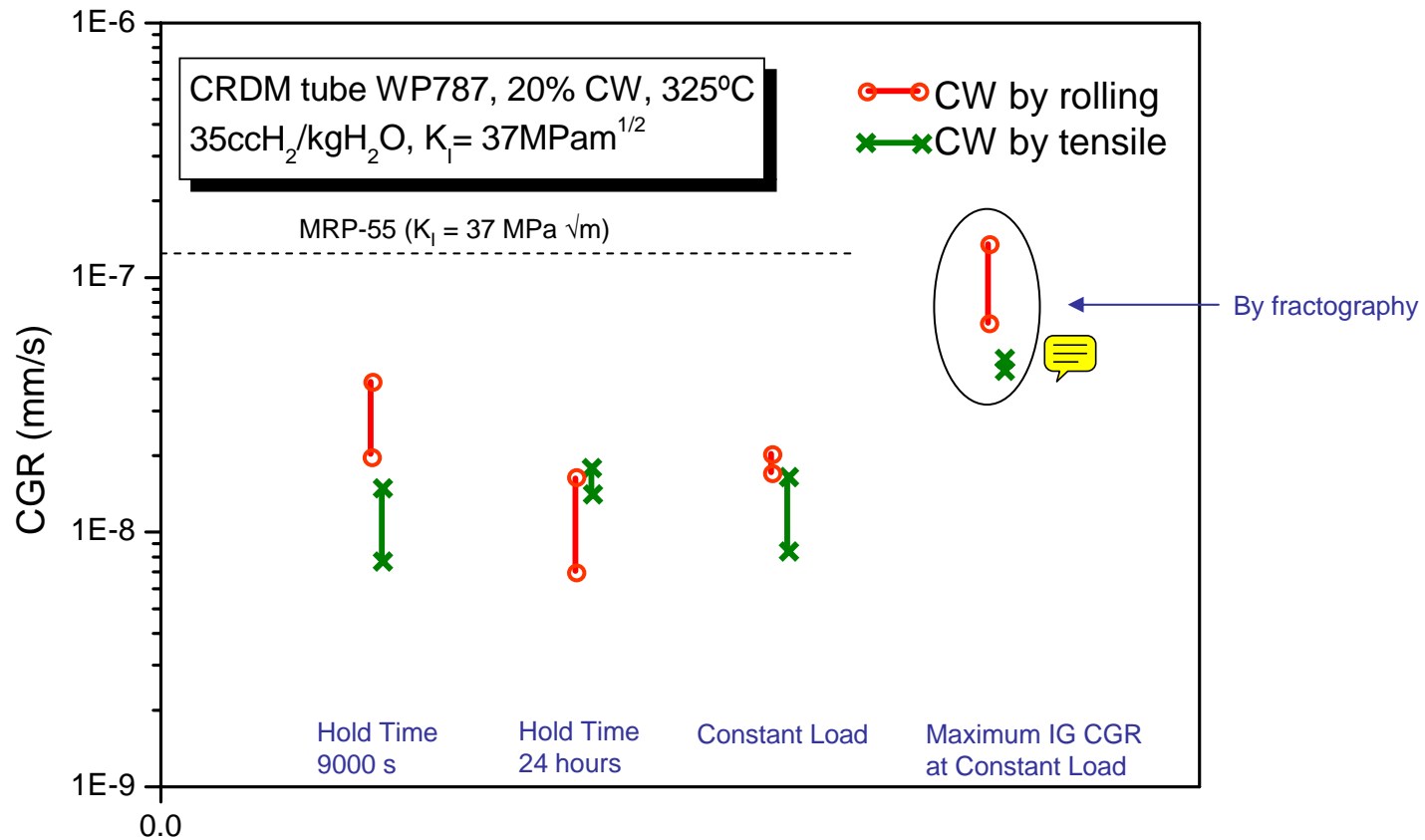
Specimen 9T2. 20% CW by tensile. CRDM tube WP787  
325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m



- ✓ Fracture surface is mainly TG. Only minor IG isolated areas have been found at the crack tip, but more than in the specimen 9T1
- ✓ Isolated grains are found in the commencement of Hold Time periods.
- ✓ Estimated final K<sub>I</sub> = 37.7 MPa√m

Summary of results. CRDM tube WP787. 20% CW  
 325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub> = 37 MPa√m

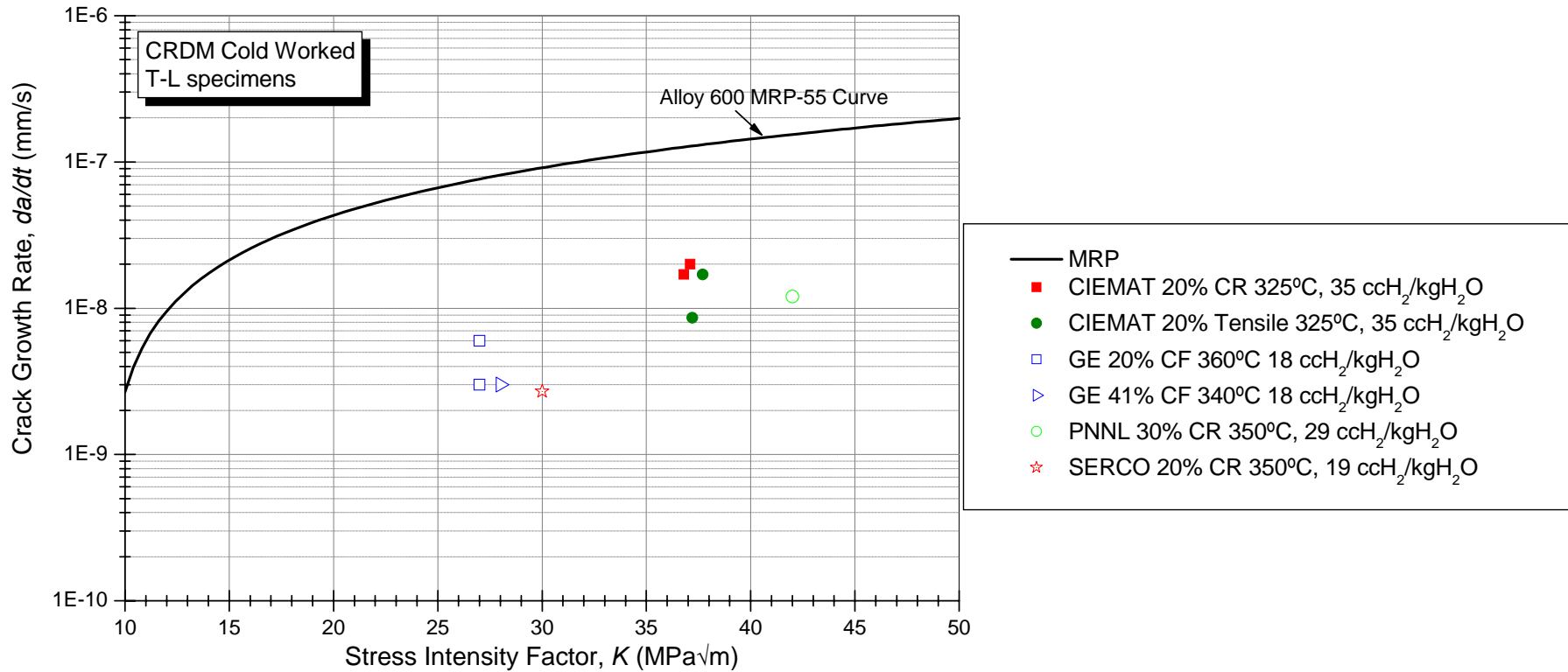
CGR by DCPD signal, corrected by fractography



CGR is higher in the CRDM Alloy 690TT cold worked by rolling than by tensile straining

Summary of results. CRDM tube WP787. 20% CW  
Comparison with other laboratories

CRDM 20-41 % Cold Worked  
 T-L orientation, Temp. 325-360°C  
 H<sub>2</sub>: 18-35 ccH<sub>2</sub>/kgH<sub>2</sub>O





Summary of results. CRDM tube WP787. 20% CW  
325°C, 35 ccH<sub>2</sub>/kgH<sub>2</sub>O, K<sub>I</sub>= 37 MPa√m

- ✓ Under Constant Load, CGRs for rolling material is higher than CGRs for tensile strained material
- ✓ The specimens CW by rolling show larger IG areas than the ones CW by tensile straining
- ✓ Small secondary cracks have been only observed in the specimens CW by rolling
- ✓ Mainly Transgranular morphology has been associated to the Hold Time periods.

CGRs for 20% CW Alloy 690 TT is higher for rolled material than for tensile strained material

	20% CW rolling	20% CW tensile
YS (MPa) at 325 °C	557	444
Hardness, HV1	298	279
Equivalent deformation, %	24.9	18.6
Average misorientation	0.95	0.72

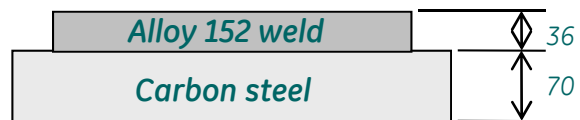
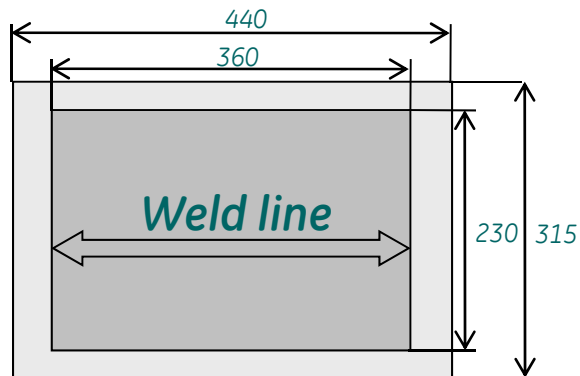
What is the point in comparing “different” materials?  
What should be the more adequate parameter for comparison?

## PWSCC CGR of Cold Rolled Alloy 152 Weld

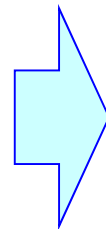
### Chemical Composition of Alloy 152 specimen

C	Si	Mn	P	S	Cu	Cr	Ni	Fe	Al	Ti	N	O	Nb+Ta
0.036	0.28	4.24	0.003	0.002	0.01	28.79	Bal.	8.91	0.09	0.08	0.018	0.06	1.46

### Geometry of Alloy 152 specimen

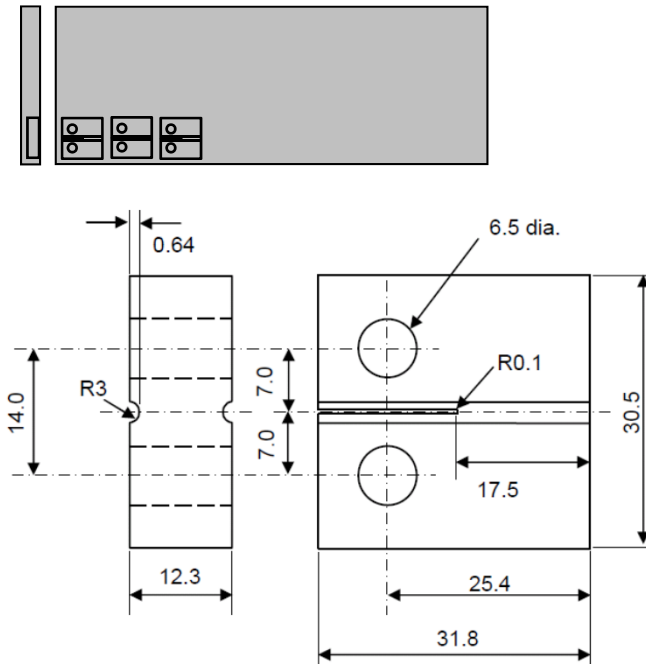


SMAW weld



1D Cold  
Rolling

(CR:15, 30 and 50%)



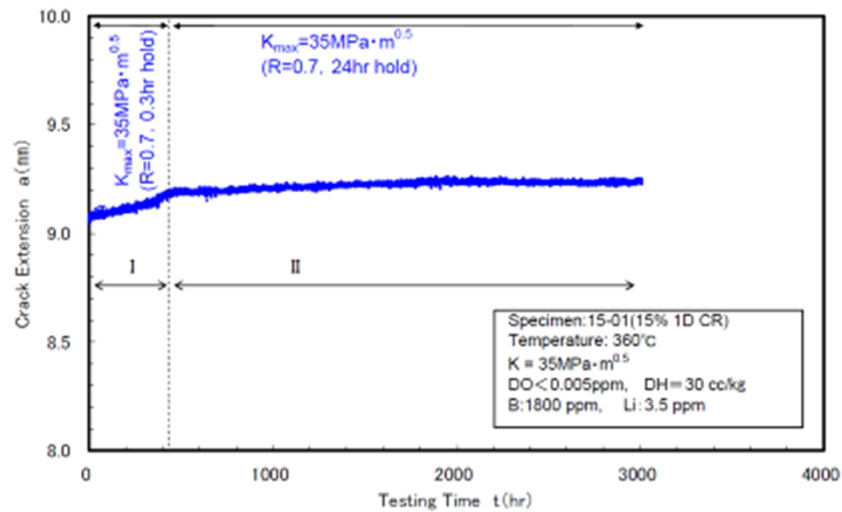
## Experimental Procedure for Crack Propagation Test

### PWSCC Crack Propagation Test Condition

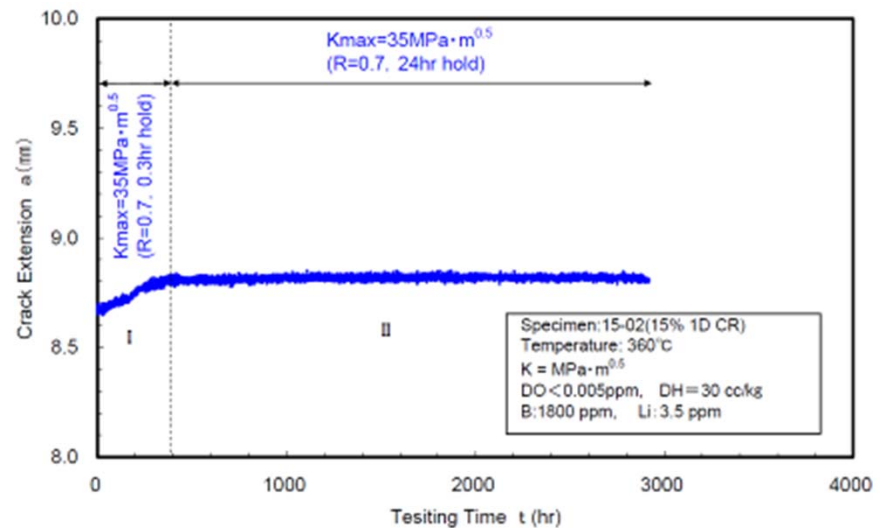
Test piece	0.5T CT
Loading	$K_{\max} = 35\text{MPa} \cdot \text{m}^{0.5}$ Initially $R=0.7$ , 0.3hr holding at $K_{\max}$ Then $R=0.7$ , 24hr holding at $K_{\max}$
Environment	360°C Simulated primary water DO<0.005ppm, DH:30 cc/kg B:1800 ppm Li: 3.5 ppm

# Crack Extension and Fracture Surface after the Test

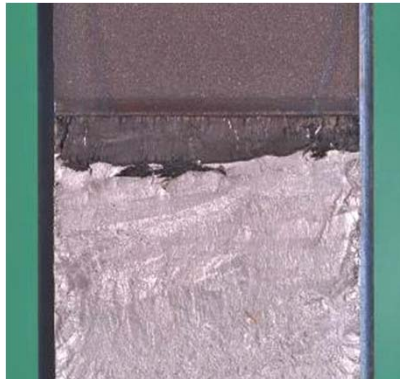
Specimen 15-01  
(15% 1D CR)



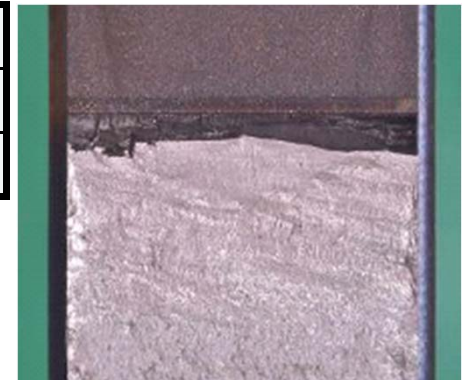
Specimen 15-02  
(15% 1D CR)



Period	CGR(mm/s)
I	$5.1 \times 10^{-8}$
II	$5.6 \times 10^{-9}$

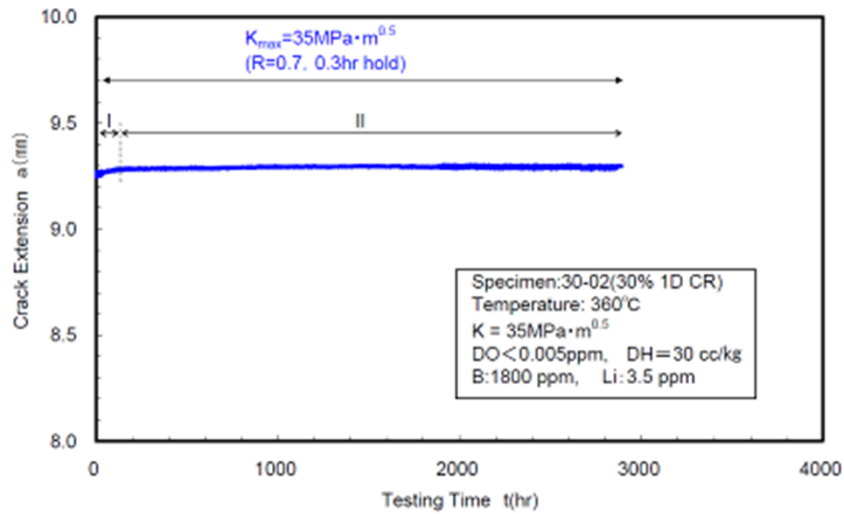


Period	CGR(mm/s)
I	$1.1 \times 10^{-7}$
II	$1.0 \times 10^{-9}$

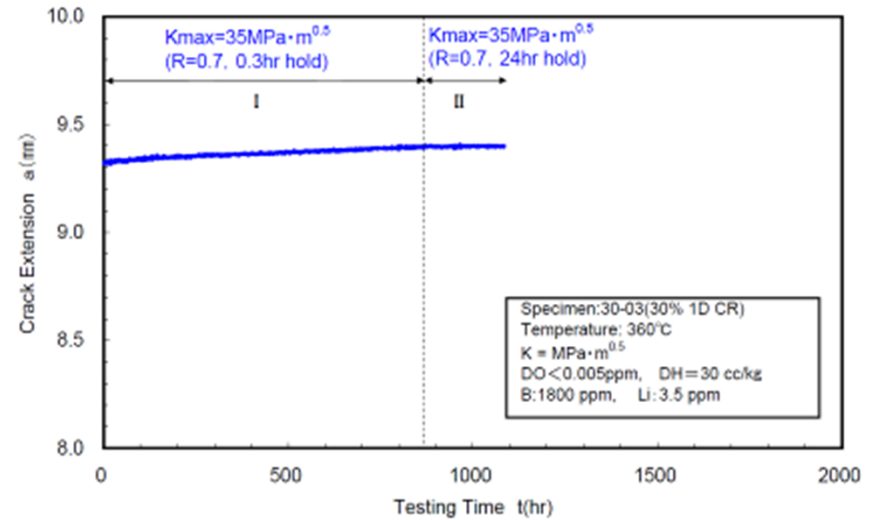


# Crack Extension and Fracture Surface after the Test

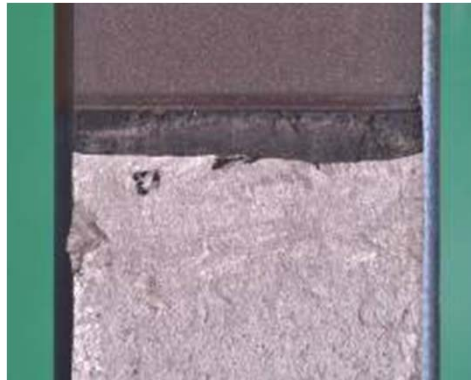
**Specimen 30-02**  
**(30% 1D CR)**



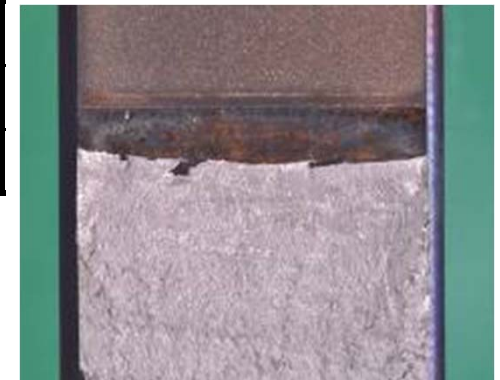
**Specimen 30-03**  
**(30% 1D CR)**



Period	CGR(mm/s)
I	$6.1 \times 10^{-8}$
II	$8.0 \times 10^{-10}$

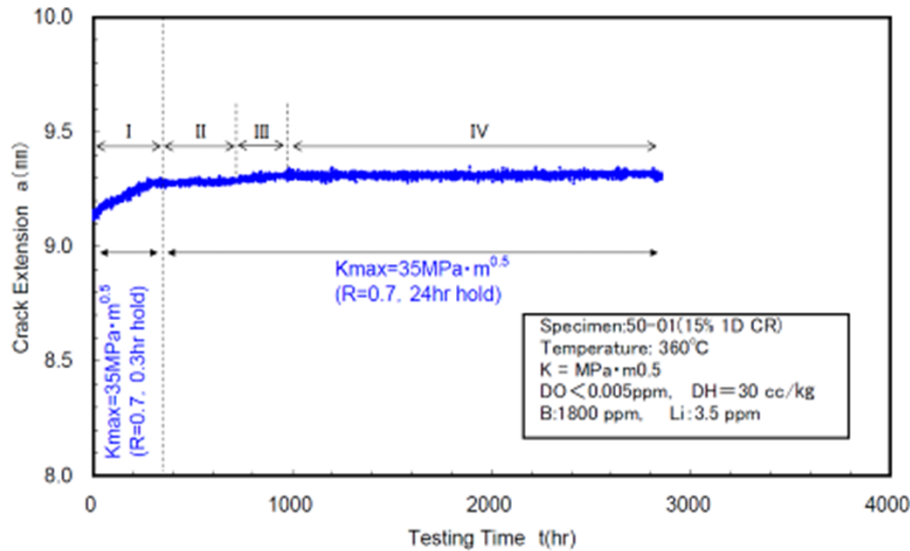


Period	CGR(mm/s)
I	$2.1 \times 10^{-8}$
II	$3.0 \times 10^{-9}$

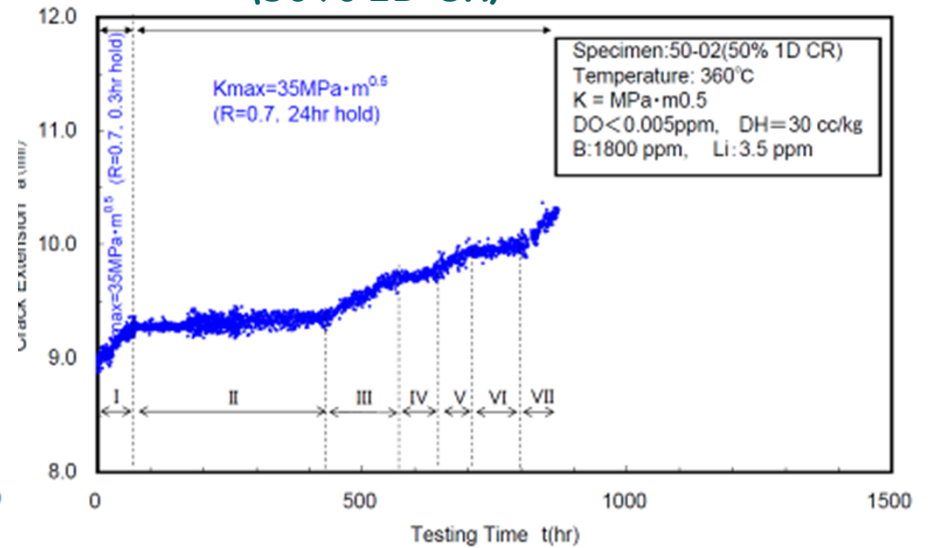


# Crack Extension and Fracture Surface after the Test

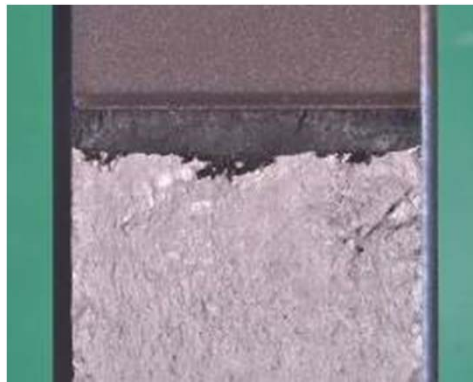
Specimen 50-01  
(50% 1D CR)



Specimen 50-02  
(50% 1D CR)



Period	CGR(mm/s)
I	$1.2 \times 10^{-7}$
II	$6.3 \times 10^{-9}$
III	$3.4 \times 10^{-8}$
IV	$1.0 \times 10^{-9}$



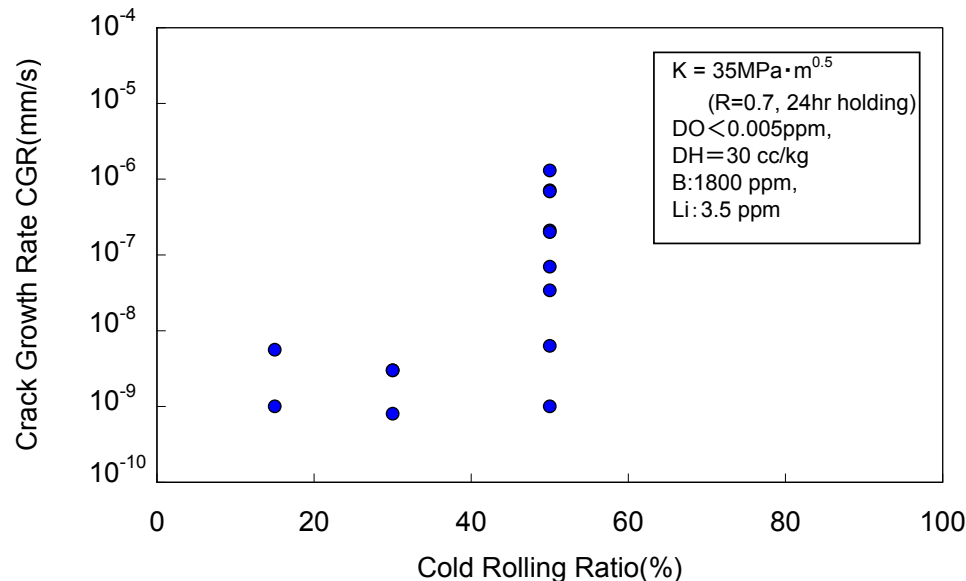
Period	CGR(mm/s)
I	$1.3 \times 10^{-6}$
II	$7.0 \times 10^{-8}$
III	$7.1 \times 10^{-7}$
IV	$2.1 \times 10^{-7}$
V	$6.9 \times 10^{-7}$
VI	$2.0 \times 10^{-7}$
VII	$1.3 \times 10^{-6}$



## Discussion

*CR ratio is one of the dominant factors in CGR.*

*However, CGR scatter is larger than change caused by CR ratio.*



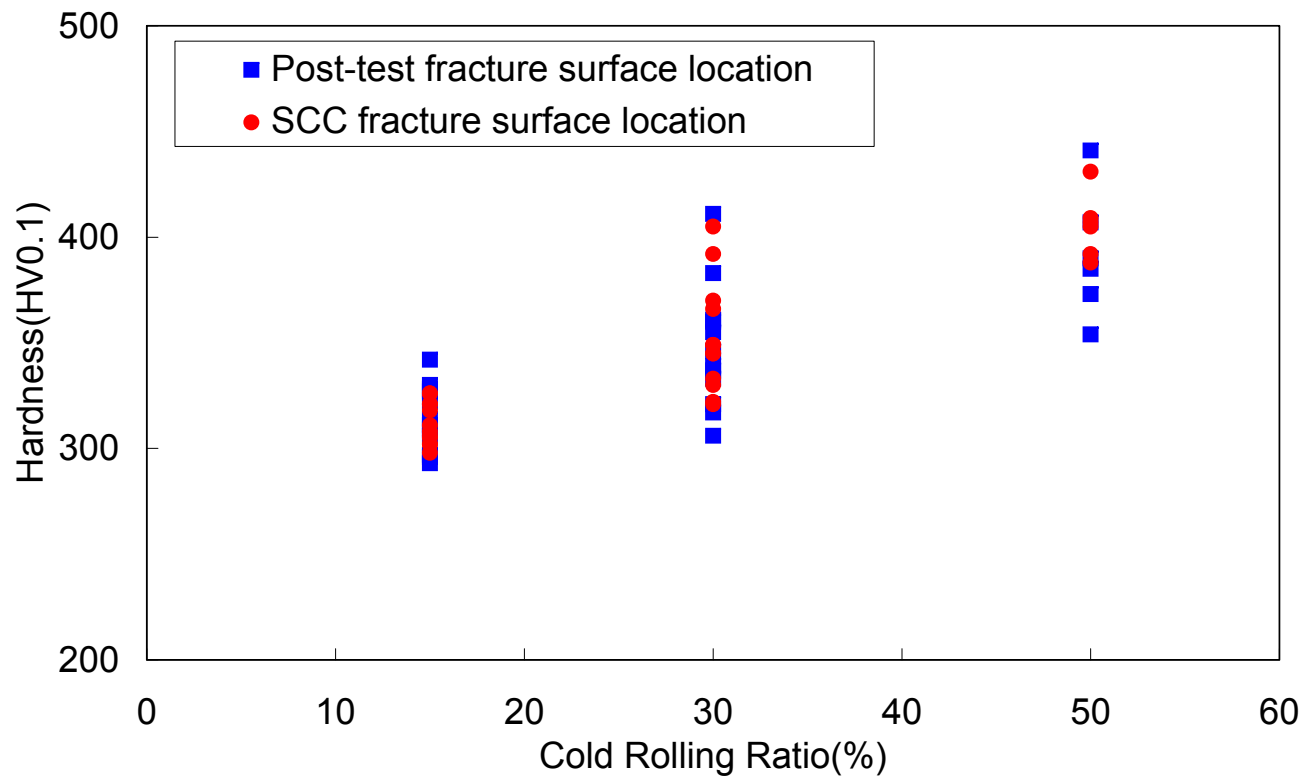
*Example of local crack extension*



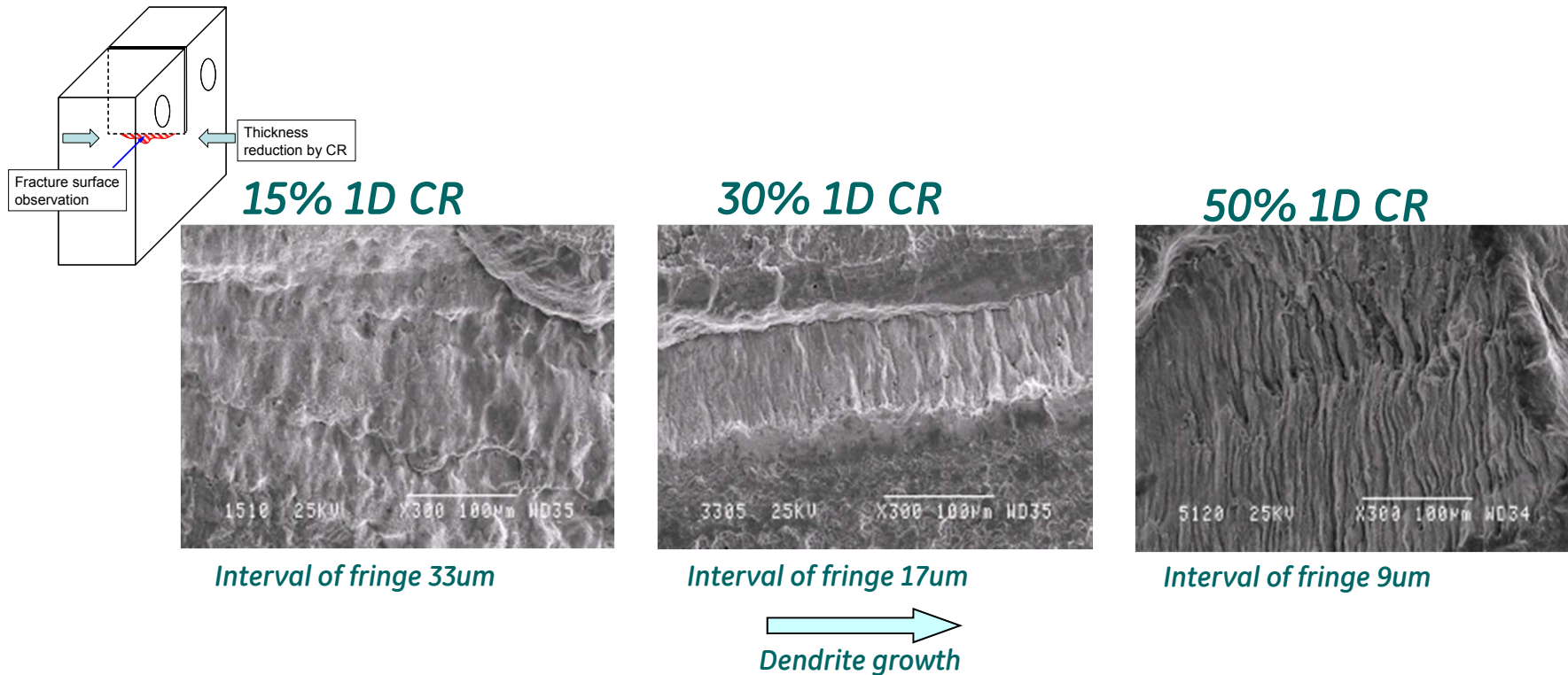
- In all specimens, crack extension is local.*
- Fracture surfaces were investigated to find cause of the local high CGR.*



*No significant difference between hardness of SCC fracture surface and that of post-fracture surface*



## Discussion: Fracture surface characteristics



- Interval of fracture surface fringe pattern is proportional to CR ratio.
- Therefore, the fringe reflects geometric change in microstructure caused by CR.
- The fringe is estimated to be surface of dendrite secondary arm.

## *MHI Summary*

- *PWSCC susceptibility of alloy 152 is strongly dependent of microstructure.*
  - *In the case of alloy 152 weld, CGR strongly depends on direction of dendrite growth.*
- *For alloy 690, CGR depends on grain boundary coherency.*
- *Cold working increases CGR of both alloy 152 & alloy 690.*