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SCC CGRs of Alloys 690 and 152 Weld in PWR Water

Bogdan Alexandreanu, Omesh Chopra, and Bill Shack Nuclear Engineering Division Argonne National Laboratory

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Program Objectives and Presentation Topics

ANL NRC Program Objectives

- SCC CGRs of Alloys 690/52/152
- SCC CGRs of service Alloys 600/82/182 (previous work included materials from Davis-Besse and V.C. Summer plants)

Presentation Topics

CGR testing in simulated primary water

- experimental approach

Crack Growth Rates for Alloy 690

- cold-worked and as-received conditions
- Crack Growth Rates for Alloy 152 Weld



Stress Corrosion Cracking Testing Facilities

- Four (of eight) hydraulic testing frames:
 - upgraded with Instron digital controls
 - 1-6L autoclaves rated to 320-350°C
 - instrumented for CGR measurements
 - independent water loops
- Crack initiation facility:
 - two 8-L autoclaves rated to 350°C
 - independent water loops
 - accommodate reverse U-bend and tubular specimens





Experiment

- Temperature: 320°C
- PWR Water (<10 ppb DO, 1000 ppm B, 2 ppm Li, ≈23 cc/kg hydrogen)
- Flow Rate: ≈55 mL/min
- Conductivity: ≈20 µS/cm
- Loading sequence to facilitate the transitioning from transgranular fatigue cracking to intergranular SCC cracking



Transitioning to SCC

ANL approach very similar to the widely-used method (increasing R and decreasing frequency) but with a few differences:

- Precracking is always conducted in the environment¹

- Use of slow/fast sawtooth with increasing rise times
 - Loading takes into account possible ΔK and ΔK_{th} effects²
- Monitor specimen response
 - Calculate cyclic CGRs based on rise time³
 - Monitor cyclic rates for environmental enhancement ⁴
- Attempt to transition only from those conditions that show environmental enhancement
- ¹ Provides (essential) baseline cyclic data
- ² $\Delta K / \Delta K_{th}$ effects have been observed/formulated for Alloy 690
- ³ Well-characterized cyclic CGR also enables the calculation of the SCC component for periods with hold times or constant load with periodic unloading
- ⁴ Accomplished by plotting the measured in the environment vs. the rates that would be expected in air under the same loading conditions



The Analysis of Cyclic CGR Data*



* Example shown is for Ni-alloy welds, the approach for Alloy 690 is similar



Cyclic CGR Data for Alloy 690 in PWR and HP Water at 320°C



Earlier ANL and B&W data in good agreement; no environmental enhancement

ΔK/ΔK_{th} was observed/formulated for Alloy 690; the effect is taken into account in current tests



Cold-Rolled Alloy 690 Specimens

- Alloy 690 in plate form (MIL-DTL-24802), Heat NX3297HK12
 - Cold-rolled in three passes to achieve approx. 26% reduction in thickness
 - Deformation is inhomeogeneous: near-surface harder than mid-plane by 12%
 - 1/2T-CT specimens were cut in SL and ST orientations (mid-plane)





Alloy 690TT Specimens

- Alloy 690TT CRDM tubing, 2541 Heat WP142
 - 1/2T-CT specimens were cut from tubing (in as-received condition) in CR and LR orientations







Cold-Rolled Alloy 690 Specimen A690WC-SL-1



Well-behaved crack advance under constant load



Cold-Rolled Alloy 690 Specimen A690WC-ST-1



Well-behaved crack advance under constant load



Fracture surfaces of A690WC-SL-1 and A690WC-ST-1



Straight fracture surfaces, uniform SCC engagement





Cold-Rolled Alloy 690 Specimen A690WC-ST-2*



- No effect of temperature (good agreement with Bettis and GE observations)
- * Test in progress



Alloy 690 Specimen C690-CR-1* (as-received condition)



CGR of 5×10⁻¹² m/s maintained for approx. 1100 h under constant load

* Test in progress



Cyclic CGR data for Alloy 690



- Alloy 690 cold-rolled shows significant environmental enhancement, comparable to Alloy 600 from Davis-Besse (Nozzle #3)
- Cyclic CGR rates for as-received alloy lower than those for cold-rolled alloy



SCC CGRs vs. K for Alloy 690



- Good agreement with Bettis data at the same hydrogen level (23 cc/kg)
- ANL data: CGR rates for as-received* alloy lower than those for cold-rolled alloy
- * C690-CR-1 testing in progress



Alloy 152 Specimens (A152-TS-2 and A152-TS-4)



- Electrode heats: NX168IJK (52, 3/32), WC96D8 (152, 1/8),
 WC43E9 (152, 3/32), WC04F6 (152, 1/8)
- 1-T CT specimens cut in TS orientation (T= long transverse, S = short transverse). Crack growth is along the direction of dendrites



Alloy 152 specimen A152-TS-2



Well-behaved crack advance under constant load



Fracture surface of A152-TS-2







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Fracture surface of A152-TS-2







Alloy 152 specimen A152-TS-4



 Well-behaved crack advance under constant load with periodic unloading (this type of loading was necessary to break the ligaments)





Fracture surface of A152-TS-4





Specimen was transitioned to SCC twice maintained grow

- Specimen was transitioned to SCC twice, maintained growth under constant load conditions only in 2 of 7 attempts
- Observed behavior suggested the formation of ligaments
- SCC CGRs measured during cyclic loading with hold times or at constant load with periodic unloading



Fracture surface of A152-TS-4 (Examples of Ligaments)





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Cyclic CGR Data for Alloy 152 Weld



- Select conditions show environmental enhancement
- R=0.7 and rise 1000s shows no enhancement; the behavior seems to suggest a ∆K effect (and is consistent with other Alloy 690/152 observations)



SCC CGRs vs. K for Alloy 152



25th Percentile of Ni-weld Curve bounds the data





Summary of Testing Results

- Fracture surfaces were uniform for both Alloy 690 and 152 specimens; the experimental approach facilitated transitioning to SCC
- The cyclic CGRs of cold-rolled Alloy 690 show significant environmental enhancement
- As-received Alloy 690 appears to show less environmental enhancement than the cold-rolled alloy
- The environmental enhancement of cyclic CGRs is minimal for Alloy 152
- The SCC CGRs in simulated PWR water at 320°C are in the 10⁻¹¹ m/s range for cold-rolled Alloy 690 and as-welded Alloy 152

