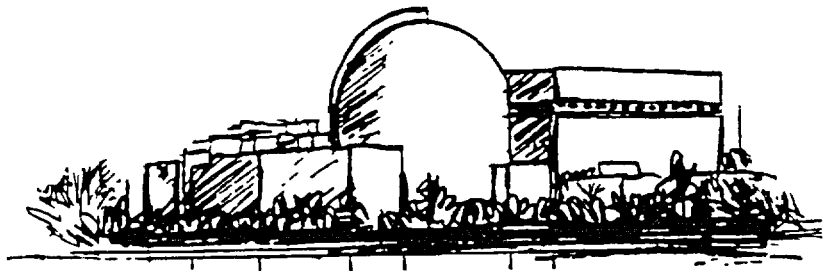


# Seabrook Steam Generator Tube Assessment Project

November 14, 2002

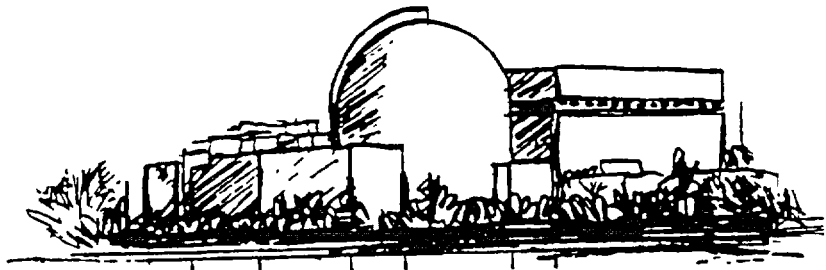
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# Meeting Objectives

- Present Root Cause Evaluation Results for Axial Indications Identified in May, 2002 Inspection
- Address NRC Staff Questions
- Reach Agreement on Extent of Condition

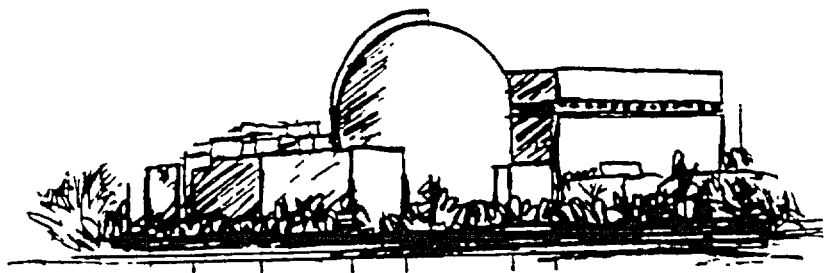
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# Regulatory/Industry Reporting

- Results Discussed With NRC In August 2002, and October 2002
- I&E Notice Issued By NRC
- INPO Event Report Issued
- Informed Plants With Fall 2002 Inspections Of Residual Stress Signal Characteristic
- Today's Public Meeting With NRC

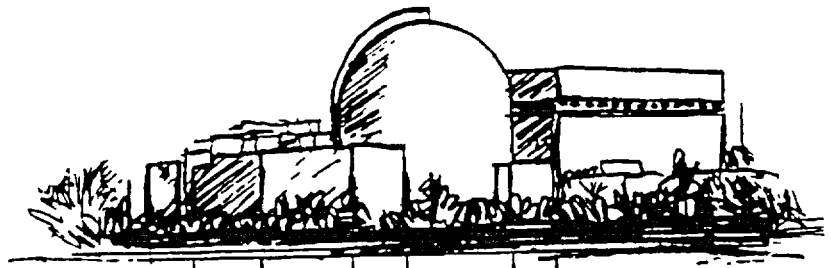
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# Investigation Process Summary

- Removed Two Degraded Tubes For Lab Testing
- Convened A Team Of Experts for Root Cause
- Identified The Root Cause
  - ❧ *High Residual Stress Caused By Manufacturing Process In A Small Subset Of Tubes*
- Identified Extent Of Condition At Seabrook
- Degradation Mechanism Is Not A New Generic Issue In Thermally Treated Tubing and is not an Active Damage Mechanism at Seabrook.

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# Seabrook Steam Generators

## ■ Model F

- œ Thermally Treated Alloy 600 Tubing

## ■ Recognized Industry Leader

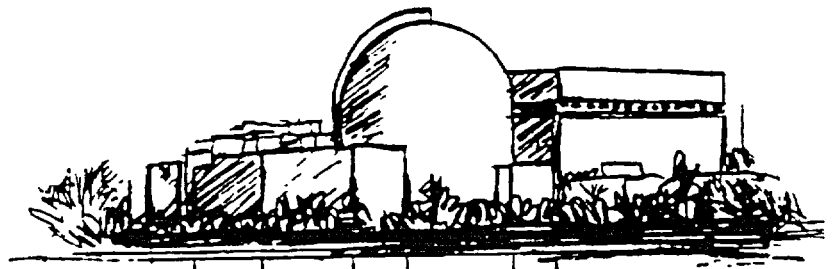
- œ Primary/Secondary Side Chemistry

- œ Life Cycle Management Strategies

## ■ Design Margin Strategies

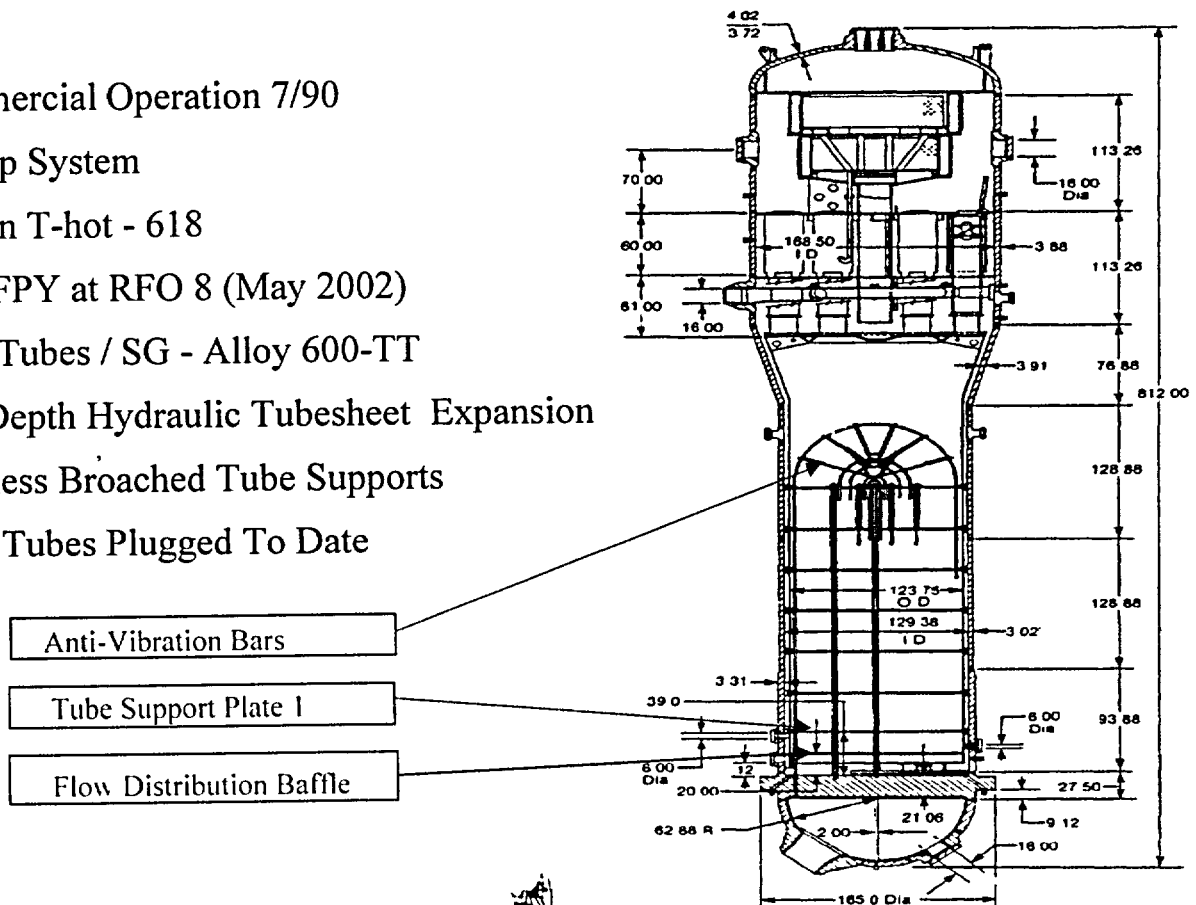
- œ 8% tube plugging vs. 0.5% plugged

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# Seabrook Model F Steam Generator

- Commercial Operation 7/90
- 4 Loop System
- Design T-hot - 618
- 9.7 EFPY at RFO 8 (May 2002)
- 5626 Tubes / SG - Alloy 600-TT
- Full Depth Hydraulic Tubesheet Expansion
- Stainless Broached Tube Supports
- 0.5% Tubes Plugged To Date

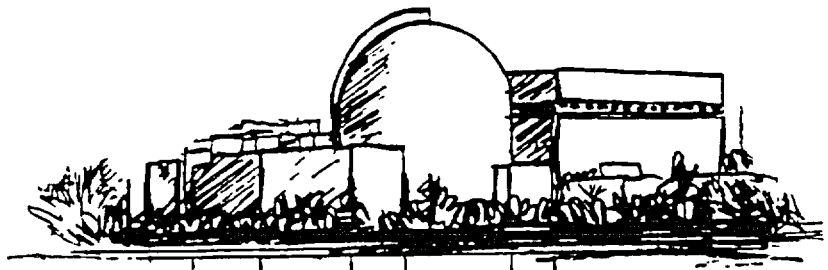


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# Manufacturing Process Timeframe

- Seabrook Tubes Manufactured between April 1980 and June 1980
- Seabrook Tubes Manufactured From 376 Heats
- Each Heat Contains Between 3,000 and 5,000 Tubes
- Eight Other Plants Had Tubes Manufactured In Same Timeframe

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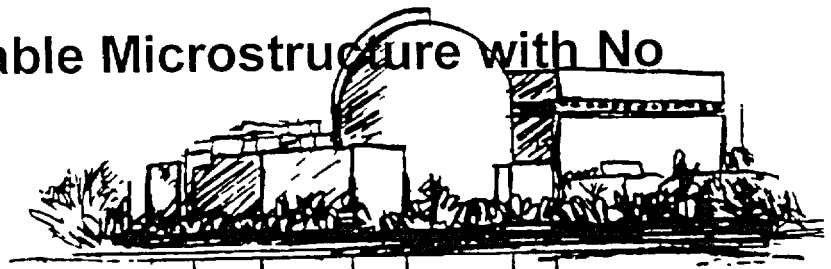
# Key Elements Of Manufacturing Process

■ Tube Manufactured To Specification (0.69" OD x .040" wall)

- |                               |                                 |
|-------------------------------|---------------------------------|
| ■ Final Mill Anneal           | 1950 F for 3 min                |
| ■ Straightening and Polishing |                                 |
| ■ Thermal Treatment           | 1320 F for 10 hrs               |
| ■ U-Bending                   | All Rows                        |
| ■ Stress Relief               | 1320 F for 2 hrs<br>(rows 1-10) |

■ Desired Result – Favorable Microstructure with No Residual Stress

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# Steam Generator Program Meets NEI 97-06

## ■ Primary Side Inspections

- ⌘ Starting In RFO 5- Inspect 100% Of Two Steam Generators At Each RFO

- \ "D" Generator Last Inspected RFO 6 May, 1999

## ■ Secondary Side Work

- ⌘ Sludge Lance / FOSAR at Each RFO

- ⌘ Extensive Hydraulic Cleaning

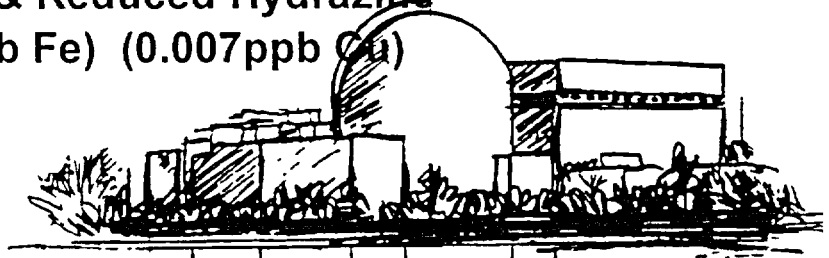
- \ UBIB, UBHB, CECIL, Pressure Pulse

## ■ Progressive Chemistry Program

- ⌘ Elevated pH, ETA, MPA & Reduced Hydrazine

- ⌘ Final Feedwater (1.4ppb Fe) (0.007ppb Cu)

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# OR08 Inspection Results

## ■ A Generator- 100 % E/C Inspection

\ AVB wear-normal mechanism in model F

## ■ D Generator- 100% E/C Inspection

\ 15 tubes with 42 indications - all at TSPs intersections

\ All indications confined to first ten rows

\ Indicative Of OD Stress Corrosion Cracking

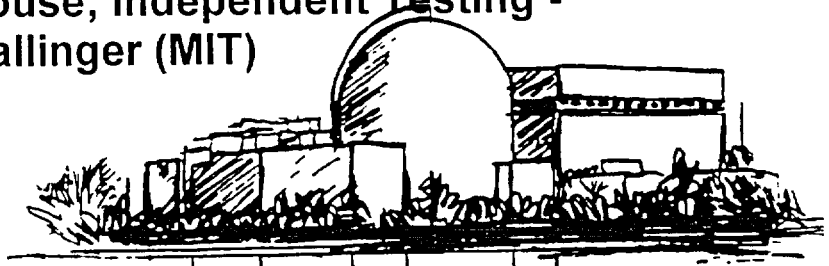
\ Confirmed by Plus Point & Ultrasonics

\ Two Tubes Removed - Fourteen Segments Available

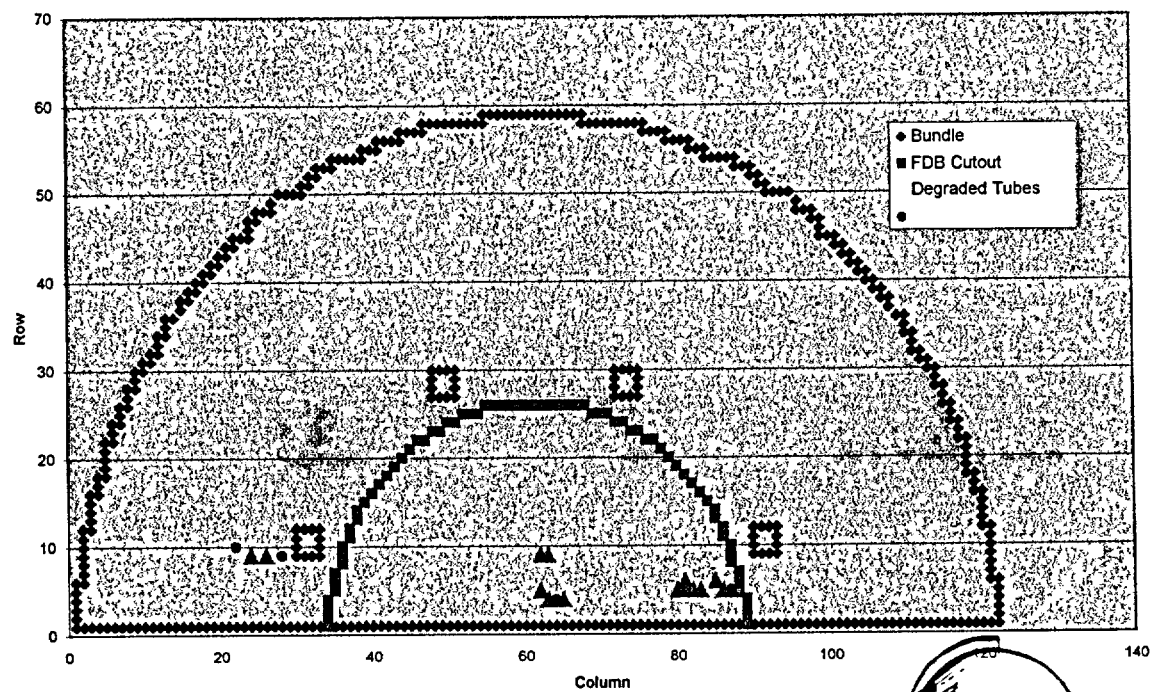
## ■ Root Cause Team Formed

∞ Lead Test Lab- Westinghouse; Independent Testing -  
Altran + Professor Ron Ballinger (MIT)

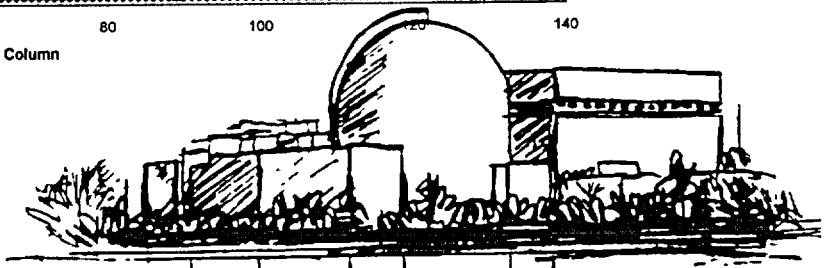
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## Location of Degraded Tubes



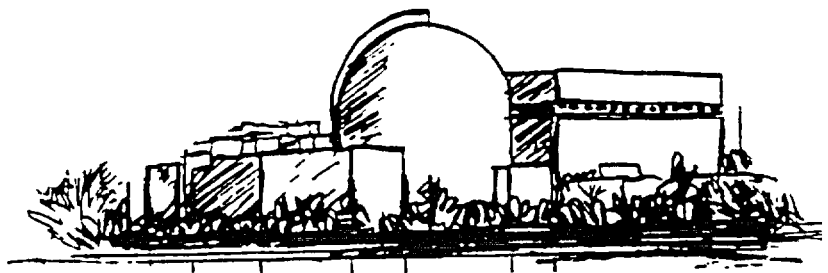
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# Pulled Tube Testing Summary

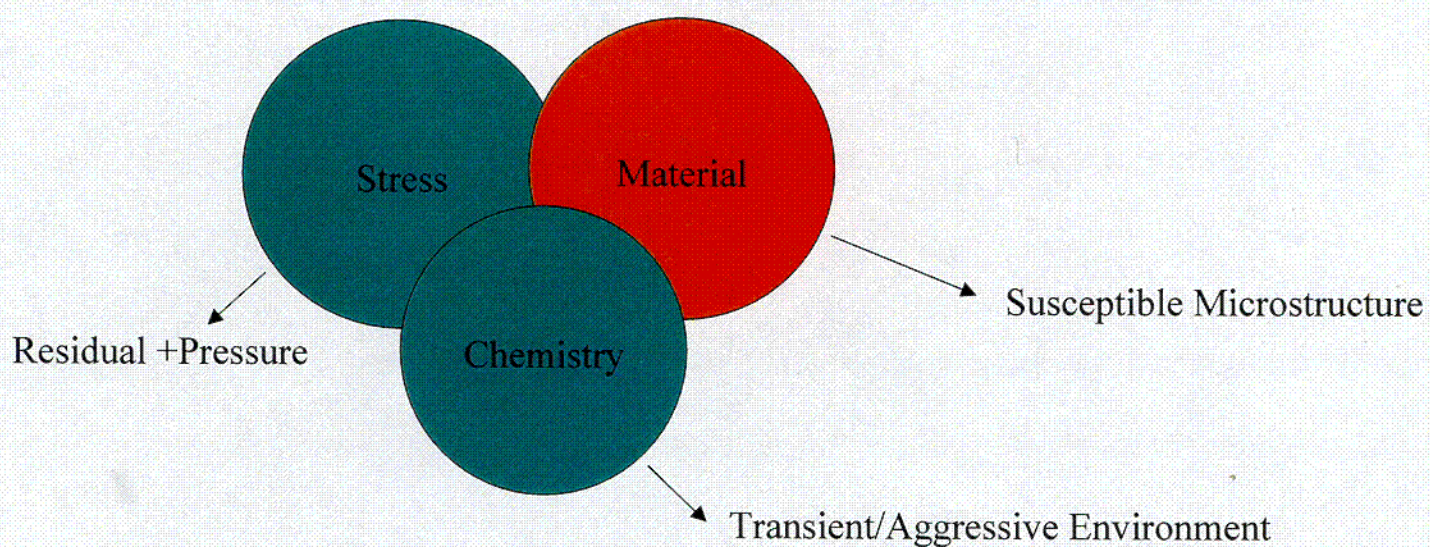
- Two Pulled Tubes Available For Testing (both from Heat 1374)
- Lab test confirmed field E/C data - Tubes have Micro Cracks
- Burst Test confirmed tube structural capability (i.e. Large Safety Margin)
  - ☞ Most Deeply Flawed Segment Tested To 7,000 psi- No leakage
  - ☞ Non Flawed Segments Burst Tested To 13,000 psi
- Tube Chemical and Mechanical Properties Consistent w/ spec
- Fractography Confirms Cracking Mechanism Is Stress Corrosion

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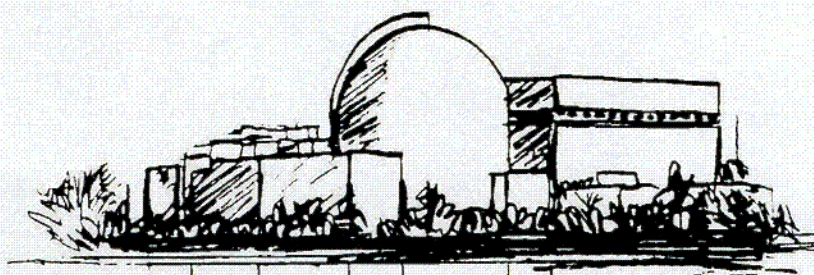




# Stress Corrosion Cracking



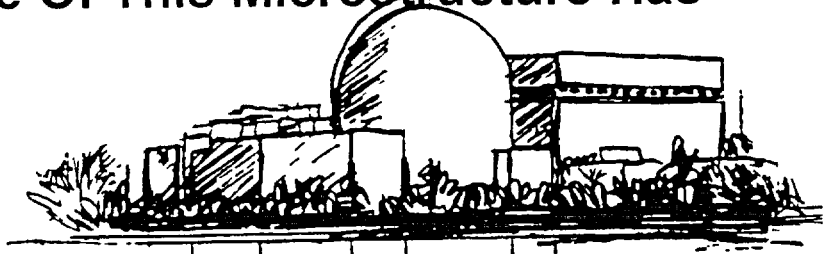
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## Material Susceptibility

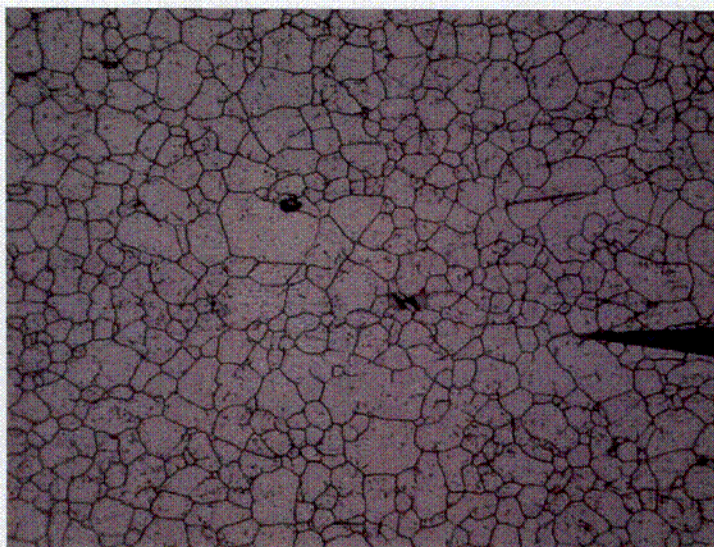
- Material Is Not Sensitized As Defined By Industry Standards
- 13 of 15 Tubes Are From Heat 1374 (.048% C)
- Two Tubes From Lower Carbon Heats (.032% C) - Heats 1456 & 1457
- Pulled Tube Micro Structure not typical of Optimum Thermally Treated Tubes
  - ⌘ Grain Structure
  - ⌘ Carbide Distribution
- In-Service Performance Of This Microstructure Has Been Excellent

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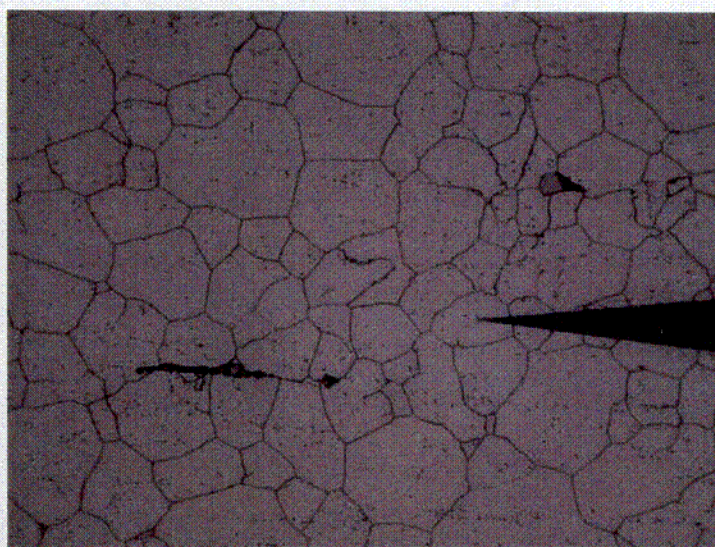
## Pulled Tube Heat 1374 Grain Boundary Etch



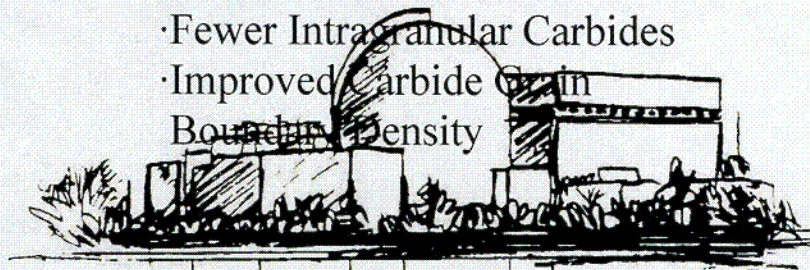
- ASTM Grain Size 10-12
- Intragranular Carbides
- Low Carbide Density On The Grain Boundaries

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## Archive Heat 1457 Grain Boundary Etch

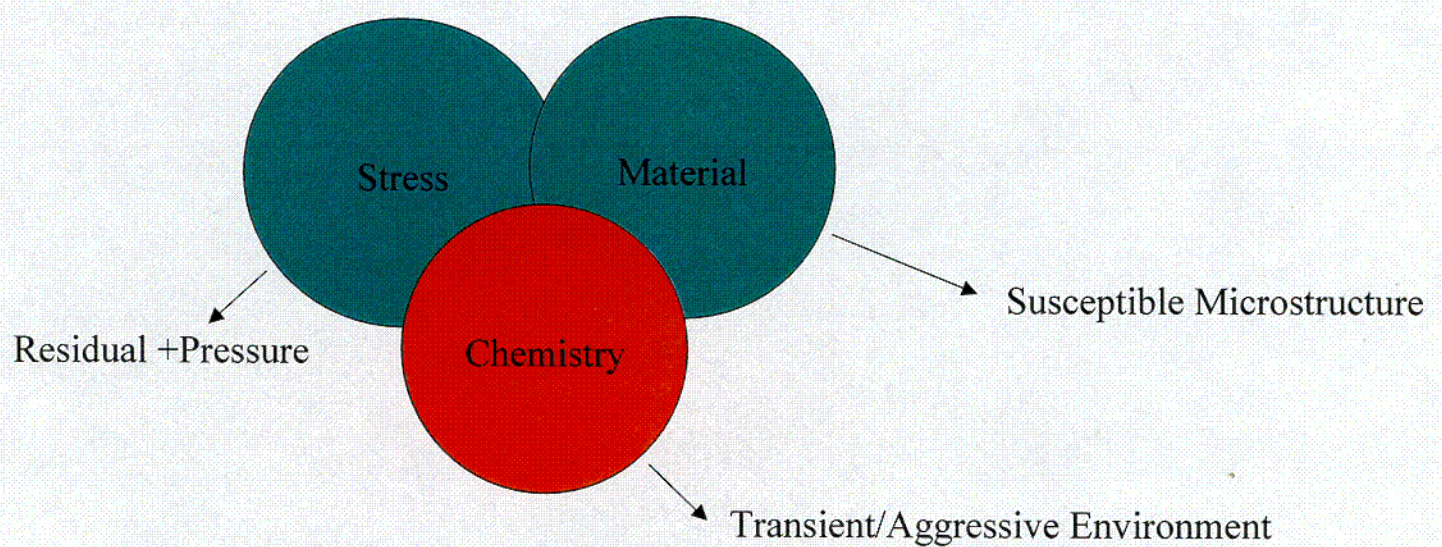


- ASTM Grain Size 5-7
- Fewer Intragranular Carbides
- Improved Carbide Grain Boundary Density

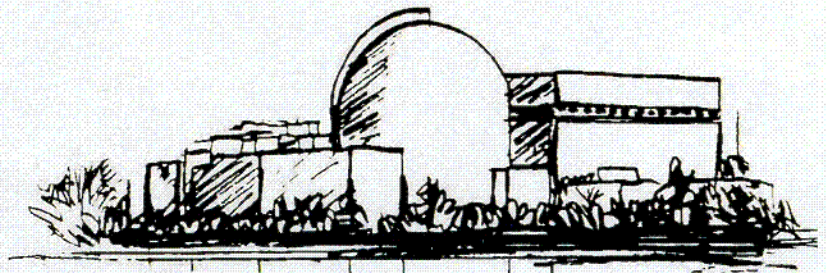




# Stress Corrosion Cracking



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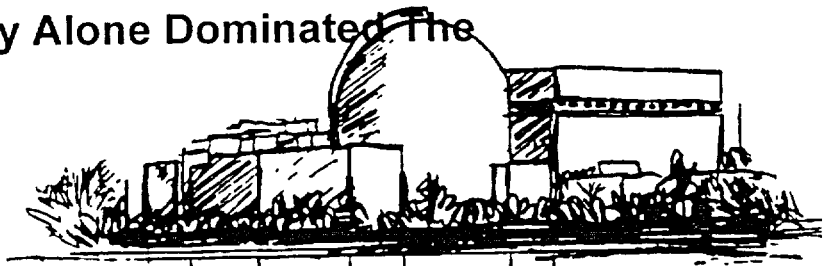




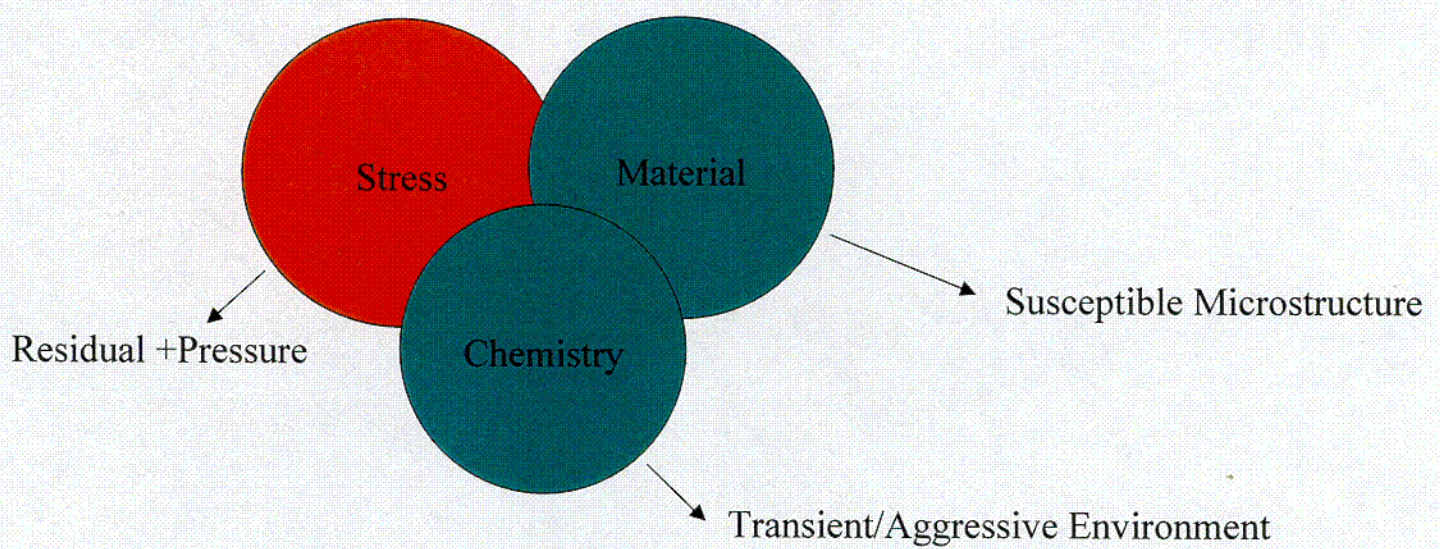
# Deposit Chemistry

- Surface Deposits From TSP Land Areas and Crack Tip Constituents Were Analyzed
- Deposits Contained Expected Constituents -
  - ⌘ Very Low Concentrations Of Copper Oxide and Lead Were Present In A Limited Number Of Samples
  - ⌘ Tight Crevice Between TSP Land and Tube Will Concentrate Contaminants
- No Evidence That Chemistry Alone Dominated The Cracking Process

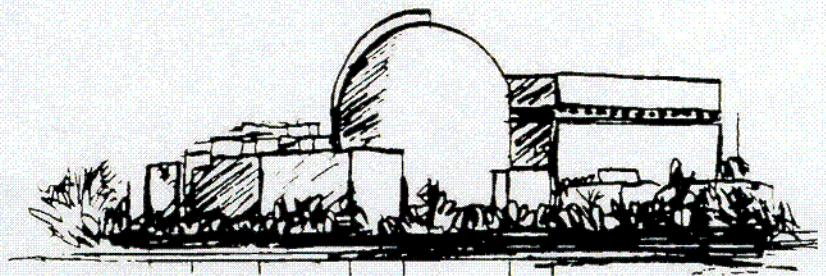
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# Stress Corrosion Cracking



**FPL** Energy Seabrook

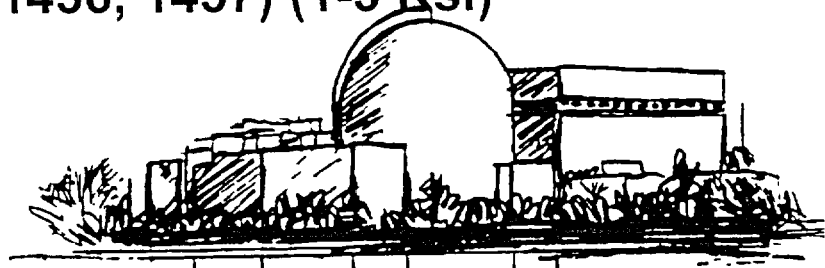




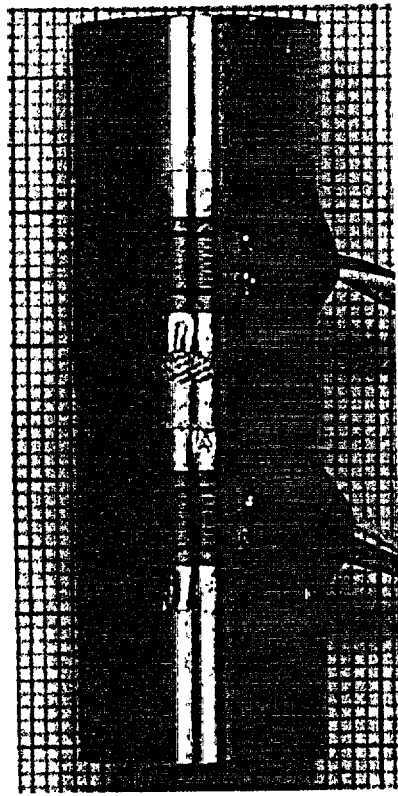
# Operating and Residual Stress

- Operating Stress - 10 Ksi (Pressure+Thermal)
- Both Pulled Tubes Have An Unexpectedly High Measured Residual Hoop Stress (16-26 Ksi vs 2 Ksi)
- OD Surface Residual Is Up To 2 times Higher
- Low Residual Stress In Archived Tubes From The Affected Heats (1374, 1456, 1457) (1-3 Ksi)

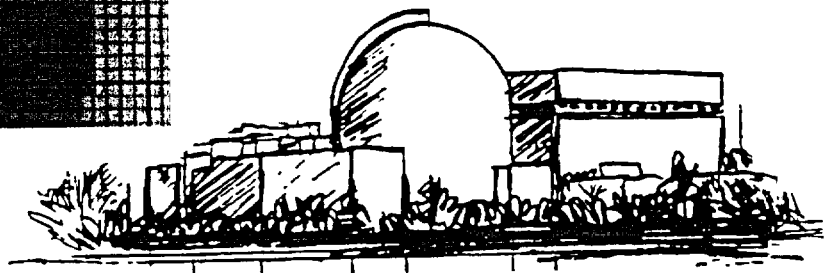
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# Residual Stress Split Ring Test Specimen



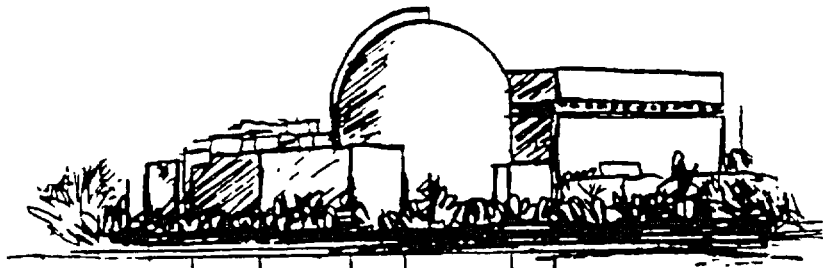
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# Residual Hoop Stress Results

Sample	Description	Residual Hoop Strain ( $\mu\epsilon$ )		Calculated Residual Hoop Stress (psi)		Average Residual Stress (psi)
		Gage 1	Gage 2	Gage 1	Gage 2	
Heat 1638	MA		-240		7,464	7,464
Heat 1638	MA	-378	-218	11,756	6,780	9,268
Heat 96845	MA	-358	-167	11,134	5,194	8,164
Heat 96845	TT	-56	-121	1,742	3,763	2,752
Pulled Tube R9C63	AR	-926	-762	28,799	23,698	26,248
Pulled Tube TR9C63	AR	-610	-496	18,971	15,426	17,198

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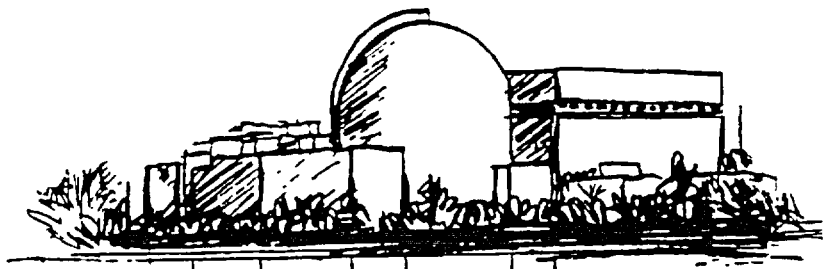


# Major Contributor to Cracking

## ■ Residual Hoop Stress

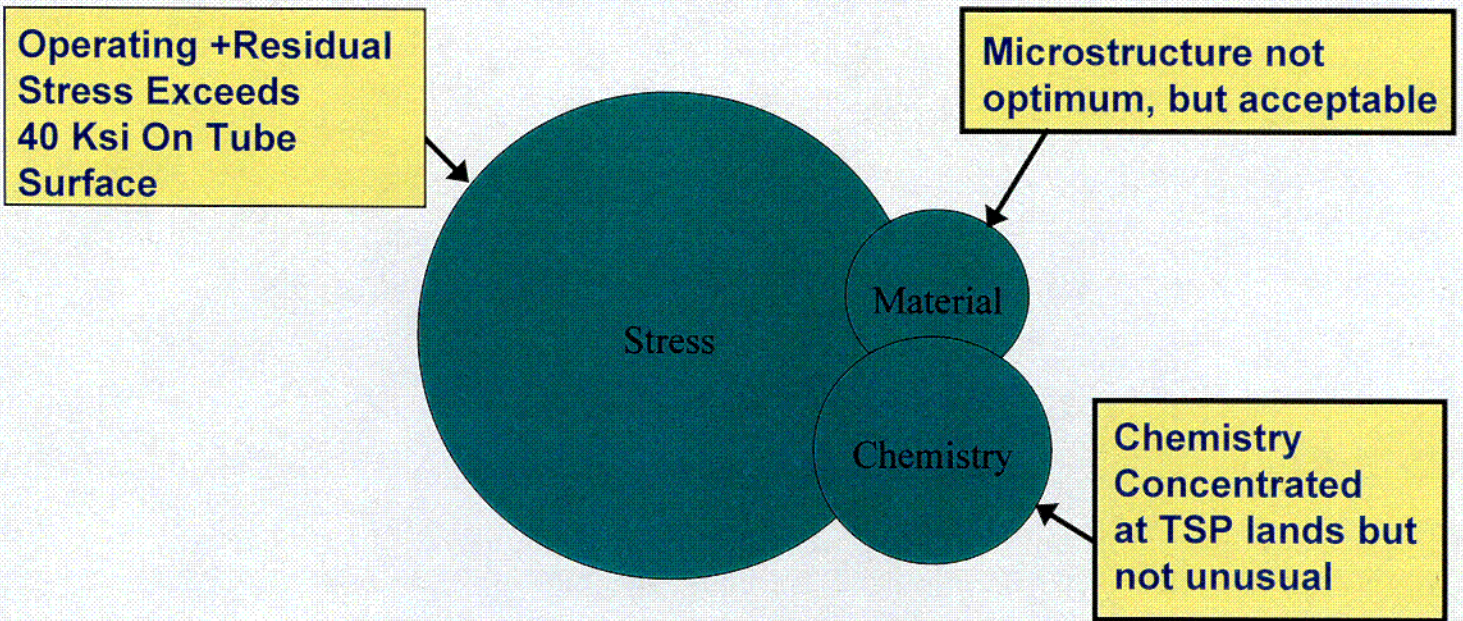
- ∞ Need > 40ksi To Initiate Cracking In TT Material
- ∞ High Residual + Operating Stress Exceeds 40 ksi on Tube Surface
- ∞ Potential Sources
  - \ Cold Work After Final TT
  - \ Improper Heat Treatment Including Stress Relief

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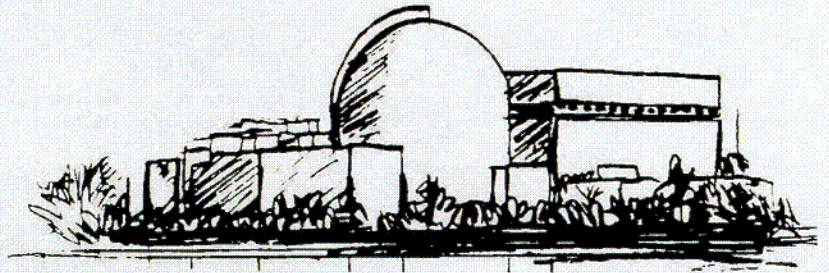




# Root Cause



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c05



# Assessment Of Extent Of Condition

## ■ Established EPRI Eddy Current To Stress Correlation Technique Used To Characterize Residual Stress In Rows 1-10

- œ Eddy Current Measures Change In Conductivity Induced By Cold Work In The Material

- œ Process Is Well Established For Rows 1-10

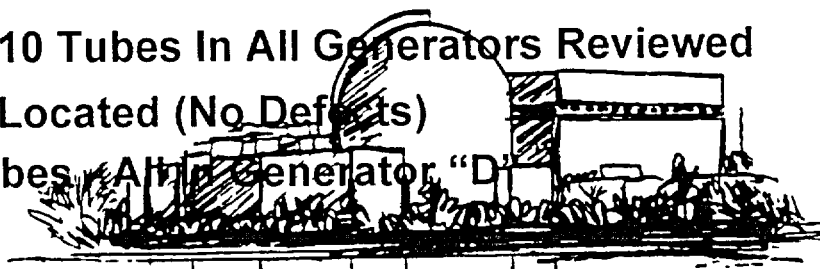
- œ Distinctive Eddy Current Offset Signal Exists In All 15 Degraded Tubes

- œ ECT Data for All Row 1-10 Tubes In All Generators Reviewed

  - \ 6 Additional Tubes Located (No Defects)

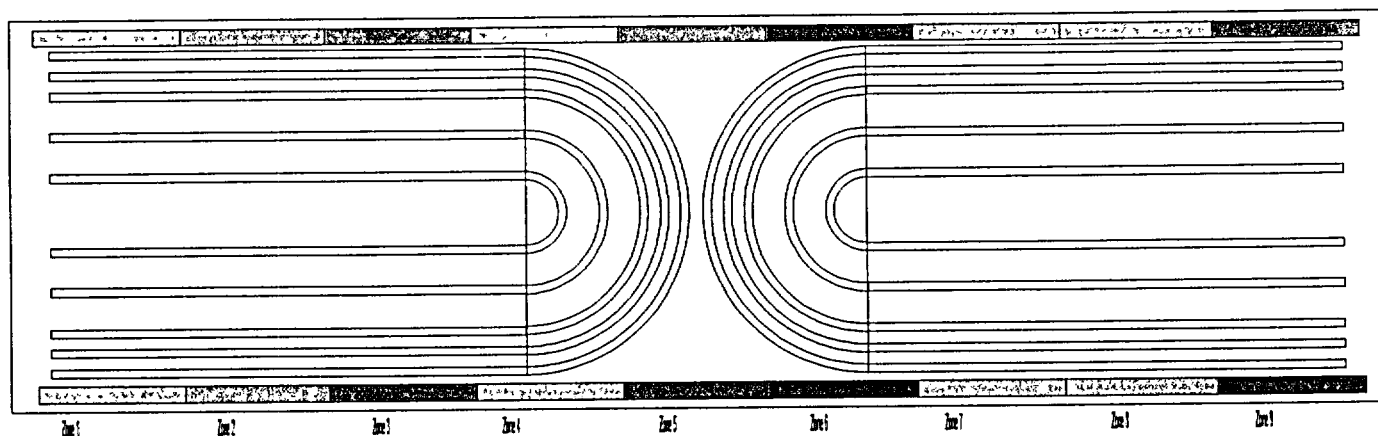
  - \ 21 Total Affected Tubes, All In Generator "D"

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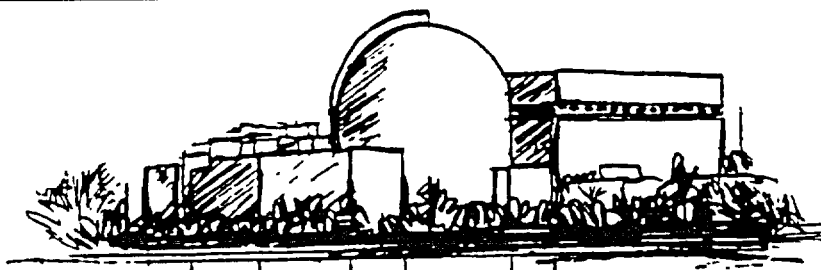


## U-bend Stress Relief Furnace Loading

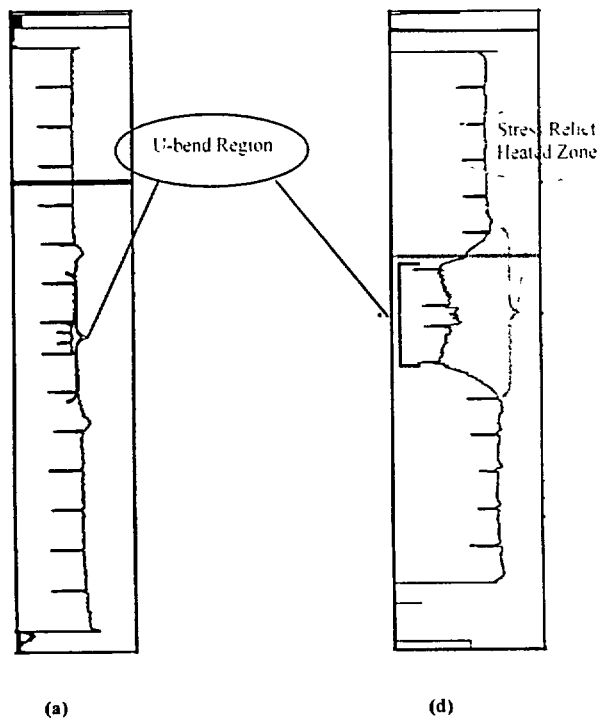


Zones 1 through 9 are heating zones in the furnace. Only zones 4, 5 and 6 were activated for stress relief of the u-bends.

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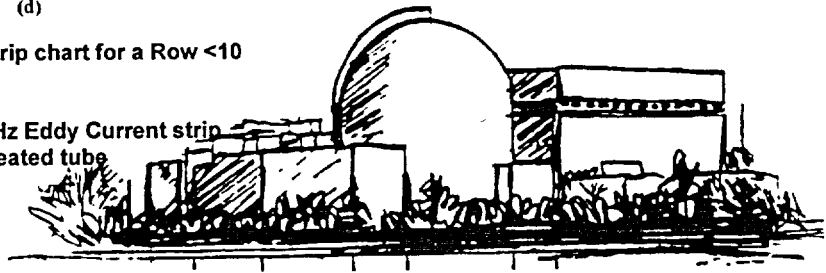


## Row 1-10 150 Hz Eddy Current Signal Traces



- (a) Normal 150 kHz Eddy Current strip chart for a Row <10 thermally treated tube
- (b) Seabrook degraded tube, 150 kHz Eddy Current strip chart for a Row <10 thermally treated tube

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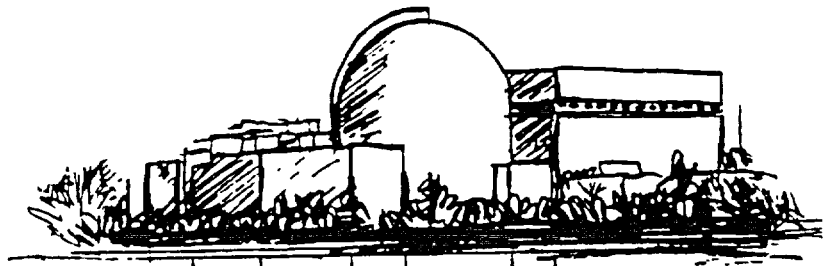
## Extent Of Condition In Rows 11- 59

### ■ Applying The Process To Rows $> 10$ Is Similar

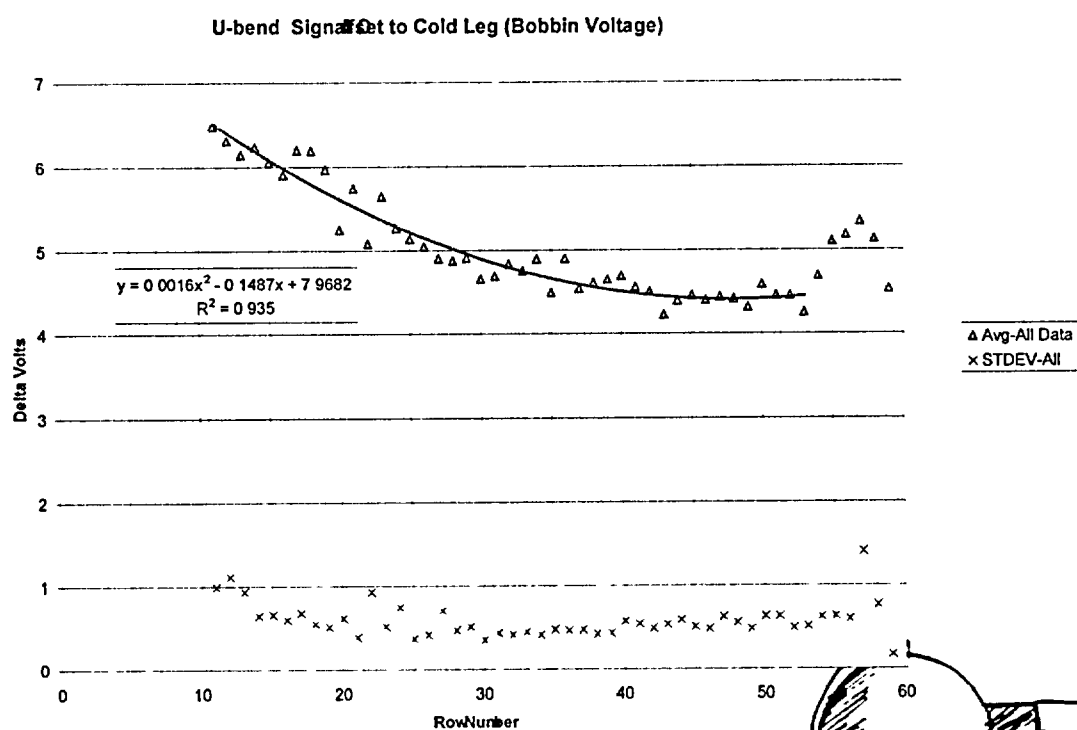
- ⌘ Eddy Current Offset Values Provide Excellent Correlation With Bend Radius
- ⌘ Tubes With High Residual Stress In The Straight Length Would Be Apparent
- ⌘ There Are No Outliers – No Signal Reversals
- ⌘ Correlation Between Degraded Tubes And Offset Signal Is 100% In Rows  $< 10$

### ■ Absence Of Observed Corrosion In Outer Rows Indicates That No Tubes Are Susceptible

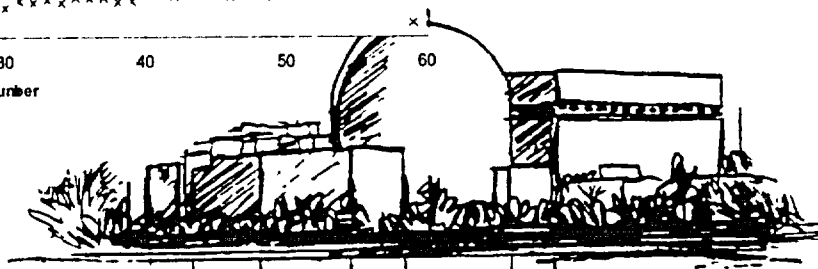
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## U-bend Bobbin Signal Offset -CL



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## Root Cause Summary

- The Root Cause Of The Tube Cracking In Seabrook Steam Generator "D" At RFO8 Is High Residual Stress In A Subset Of Tubes Caused By Manufacturing Process
- The Extent Of Condition Is Defined
- Degradation Mechanism Is Not A New Generic Issue In Thermally Treated Tubing and is not an Active Damage Mechanism at Seabrook.

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